

Properties of Barium-Magnesium Titanate Dielectrics

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Dielectrics having compositions in the system $\text{BaTiO}_3\text{-MgO-TiO}_2\text{-TiO}_2$ were matured (less than 0.1% of absorption) at 1,275° to 1,425° C. Data are given for the dielectric constant K at a frequency of 1 mc/s and various temperatures from -60° to +85° C, and for Q , the reciprocal of the power factor, at 25° C and frequencies of 50, 1,000, and 20,000 kc/s and 3,000 mc/s. Values of K (1 mc/s and 25° C) ranged from 12 to 1,550 and those of Q from 9 to 10,000. Values of K decreased, and those of Q increased for several weeks after specimens were matured, when the content of BaO was greater than 30 percent and that of TiO_2 less than 50 percent. Partial restoration of the original values of K and Q resulted from heating these specimens at various temperatures for brief periods. Linear thermal expansion (25° to 700° C) ranged from 0.46 to 0.71 percent. A few specimens of barium-strontium titanate were tested for the effects of thermal history on the properties.

I. Introduction

This is the second paper pertaining to ceramic dielectrics composed of titanium dioxide and the oxides of the alkaline earth elements. Previous work, by the present investigators, on barium-strontium titanate dielectrics [1]¹ indicated the usefulness of these materials in the fields of electrical communications and instrumentation.

Some of the properties of dielectrics having compositions in limited portions of the system BaO-MgO-TiO_2 have been determined by other investigators. Wainer [2] found that the addition of magnesia to barium titanate resulted in high electrical losses. Low losses, however, were observed by Rieke and Ungewiss [3] on bodies with compositions in a portion of the system MgO-TiO_2 (9 to 57 percent of TiO_2). Thus it might be expected that portions of the ternary system would represent compositions of bodies with low losses, and other portions would indicate compositions of bodies with high losses. It was anticipated also that many of these bodies would have a positive or neutral temperature coefficient of dielectric constant because of the low temperature coefficient of magnesium titanate reported by Rosenthal [4].

¹ Figures in brackets indicate the literature references at the end of this paper.

II. Preparation of Specimens and Methods of Test

In the preparation of specimens having the computed compositions shown in figure 1, chemically pure magnesium carbonate was the source of MgO . The titania, grade TMO, and barium carbonate were from the same stocks used in the production of barium-strontium titanates [1].

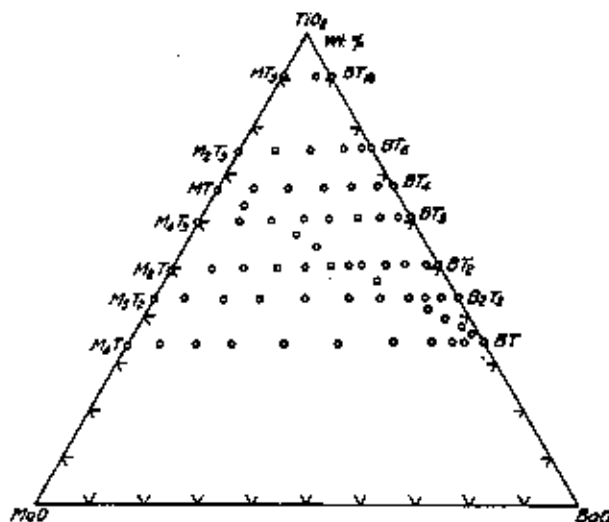


FIGURE 1. Ternary diagram for system BaO-MgO-TiO_2 showing compositions studied.

B = BaO; M = MgO; T = TiO_2 .

In the first paper of this series, details are given of the method for producing mature specimens as indicated by less than 0.1 percent of absorption (water basis). The properties of the dielectrics were determined by the methods and equipment, with one exception, previously used. A crystal-controlled oscillator was installed in the apparatus for determining the effects of variation in tem-

perature of the specimens upon the dielectric constant.

III. Results and Discussion

In table 1, data are given for the composition, heat treatment, absorption, shrinkage, dielectric constant (K), and Q -value (reciprocal of the power factor) of mature specimens.

TABLE 1. Composition, heat treatment, absorption, shrinkage, dielectric constant, K , and Q , of bodies in the system BaO-MgO-TiO₂

Specimen designation	Proportion of end members of join		Composition weight			Heat treatment			Absorption	Shrinkage	Dielectric constant, K , at 25° C and—				Reciprocal, Q , of power factor at 25° C and—			
	MgO: 5TiO ₂	BaO: 13TiO ₂	BaO	MgO	TiO ₂	No. 1 for 1 hr at—	No. 2				50 kc/s	1,000 kc/s	20,000 kc/s	3×10 ⁶ kc/s	50 kc/s	1,000 kc/s	20,000 kc/s	3×10 ⁶ kc/s
							Temperature	Time										
MT5.....	100.0	0.0	9.2	90.8	1,100	1,285	1	0.06	16.7	47	47	47	48	400	710	1,130	340	
18BM6.....	39.6	67.4	6.0	2.9	90.6	1,100	1,275	1	.01	13.0	31	73	74	102	42	93	
BT18.....	0.0	100.0	2.0	90.4	1,100	1,275	1	.00	15.0	75	74	74	200	1,000	7,000	
	2MgO: 3TiO ₂	BaO: 8TiO ₂																
M2T3.....	100.0	0.0	25.2	74.8	1,100	*1,290	3	.00	10.7	18	19	18	18	34	380	2,000	1,900	
6BM2.....	74.0	26.0	6.5	18.2	74.9	1,100	1,250	1	.01	14.0	26	25	23	136	50	20	590	
6BM5.....	47.3	52.7	13.0	11.9	75.1	1,100	1,275	1	.00	14.1	48	48	44	42	15	12	490	
6BM8.....	20.0	80.0	19.5	5.1	75.4	1,100	1,275	1	.00	14.7	70	64	46	11	11	8	
6BM9.....	4.9	95.1	23.0	1.5	75.5	1,100	1,275	1	.00	14.9	53	53	48	97	23	25	
BT6.....	0.0	100.0	24.3	75.7	1,100	1,275	1	.01	14.0	46	45	44	35	400	2,300	
	MgO:TiO ₂	BaO:4TiO ₂																
MT.....	100	0	33.5	66.5	1,100	*1,300	3	.00	12.9	18	17	17	17	200	4,000	>8,000	2,400	
4BM2.....	80	20	6.5	26.5	67.0	1,100	1,275	1	.00	15.4	26	27	25	125	28	19	110	
4BM4.....	60	40	13.0	20.0	67.0	1,100	1,300	1	.01	17.4	41	39	30	52	11	9	130	
4BM6.....	40	60	19.5	13.4	67.1	1,100	1,300	1	.01	14.8	55	49	37	44	9	8	
4BM7.....	25	75	24.3	8.4	67.3	1,100	1,300	1	.00	15.0	60	53	40	27	9	9	
4BM9.....	10	90	29.2	3.2	67.5	1,100	1,300	1	.00	13.8	46	44	39	260	22	18	
BT4.....	0	100	32.4	67.6	1,100	1,300	1	.03	15.9	34	34	35	33	>1,600	2,000	3,700	2,600
	4MgO: 3TiO ₂	BaO:3TiO ₂																
M4T3.....	100	0	40.2	59.8	1,100	1,385	1	.00	15.0	16	16	16	16	300	4,000	2,300	3,000	
3BM2.....	80	20	7.8	32.2	60.0	1,100	1,315	1	.00	15.7	18	18	17	1,600	3,000	3,600	1,100	
3BM5.....	60	40	19.5	20.1	60.4	1,100	1,300	1	.03	15.5	29	27	24	54	19	27	130	
3BM6.....	40	60	23.4	16.1	60.5	1,100	1,300	1	.00	17.1	34	31	26	34	37	13	18	110
3BM7.....	25	75	29.3	10.0	60.7	1,100	1,300	1	.01	17.3	45	39	32	23	27	8	14	320
3BM8.....	15	85	33.2	6.0	60.8	1,100	1,260	1	.00	17.5	20	47	37	62	5	10	
3BM9.....	5	95	37.1	2.0	60.9	1,100	1,260	1	.02	18.5	38	37	34	160	30	38	
BT3.....	0	100	39.0	61.0	1,100	1,260	1	.03	17.4	44	44	43	42	650	720	800	450

* Heat treated previously (see table 2).

TABLE 1. Composition, heat treatment, absorption, shrinkage, dielectric constant, *K*, and *Q*, of bodies in the system BaO-MgO-TiO₂—Continued

Specimen designation	Proportion of end members of join		Composition weight			Heat treatment			Absorption	Shrinkage	Dielectric constant, <i>K</i> , at 25° C and—				Reciprocal, <i>Q</i> , of power factor at 25° C and—			
	2MgO:TiO ₂	BaO:2TiO ₂	BaO	MgO	TiO ₂	No. 1 for 1 hr at—		No. 3			50 kc/s	1,000 kc/s	20,000 kc/s	3×10 ⁴ kc/s	50 kc/s	1,000 kc/s	20,000 kc/s	3×10 ⁴ kc/s
						Temp-erature	Time											
M2T	100	0		50.2	49.8	1,100	1,280	1	.00	13.9	14	14	14	2,200	10,000	10,000	3,000	
2BM1	85	15	7.3	42.7	50.0	1,100	1,310	1	.00	17.0	18	16	16	4,900	10,000	10,000	1,000	
2BM2	73	27	12.2	38.5	50.2	1,100	1,310	1	.01	16.0	15	15	17	1,800	6,000	8,000	180	
2BM4	60	40	19.0	30.1	50.3	1,100	1,286	6	.01	18.7	24	24	24	750	800	700	
2BM5	50	50	24.5	25.1	50.4	1,100	1,260	3	.00	17.9	32	32	31	>1,200	1,100	700	47	
2BM6	40	60	29.4	20.1	50.5	1,100	1,260	3	.02	12.7	31	31	32	220	410	270	
2BM7	30	70	34.3	15.1	50.6	1,100	1,296	1	.00	17.9	33	33	34	1,400	680	360	53	
2BM8	20	80	39.2	10.0	50.8	1,100	1,296	1	.03	18.3	32	32	33	2,000	740	500	80	
2BM87	13	87	42.8	6.6	50.9	1,100	1,296	1	.00	16.1	39	39	37	1,400	660	390	75	
2BM9	5	95	46.8	2.5	50.9	1,100	1,286	1	.00	18.1	113	110	110	130	85	47	
BT2	0	100	49.0	51.0	1,100	1,290	1	.02	16.4	204	200	197	100	70	35	
	5MgO:2TiO ₂	2BaO:3TiO ₂																
M3T3	100	0	55.8	44.2	1,245	1,415	1	.00	15.8	14	14	14	2,200	5,000	10,000	3,400		
2B3M1	90	10	5.6	50.2	44.2	1,245	1,350	1	.00	12.6	18	16	16	>1,500	10,000	10,000	270	
2B3M2	77	23	12.9	43.0	44.1	1,245	1,350	1	.00	16.9	18	18	16	>1,700	10,000	10,000	340	
2B3M3	66	34	19.0	36.3	44.1	1,245	1,200	1	.00	14.9	26	26	26	800	680	800	60	
2B3M3	50	50	28.1	27.9	44.0	1,100	1,300	1	.00	18.9	47	47	46	270	235	130	
2B3M6	35	65	36.5	19.5	44.0	1,100	1,300	1	.00	16.3	117	116	116	770	630	245	
2B3M7	25	75	42.1	14.0	43.9	1,100	1,300	1	.02	16.9	159	155	154	780	430	180	
2B3M9	10	90	50.5	5.5	43.9	1,100	1,265	1	.00	16.3	207	204	201	480	260	180	
2B3M96	5	95	53.3	2.8	43.9	1,100	1,300	1	.02	14.7	227	225	215	1,500	195	110	
B2T3	0	100	56.1	43.9	1,250	1,300	1	.01	19.8	910	900	890	70	50	25	
	4MgO:TiO ₂	BaO:TiO ₂																
M4T	100.0	0.0	66.9	33.1	1,245	1,425	1	.00	17.2	12	12	12	2,500	9,000	7,000	3,100		
BM1	90.0	10.0	8.6	60.2	33.2	1,245	1,350	1	.00	15.7	15	15	15	>1,200	10,000	5,000	370	
BM2	81.0	19.0	12.5	54.2	33.3	1,245	1,350	1	.01	15.7	21	21	21	2,100	2,000	2,000	90	
BM3	70.4	29.6	19.4	47.1	33.5	1,245	1,350	1	.01	11.2	37	37	37	800	800	300	20	
BM4	66.9	33.1	29.0	37.4	33.6	1,245	1,350	1	.00	12.8	120	120	120	140	100	55	
BM5	49.7	50.3	39.0	27.0	34.0	1,100	1,350	1	.01	11.6	250	250	250	230	400	150	
BM7	28.5	71.5	48.9	17.0	34.1	1,100	1,385	1	.00	15.6	550	550	550	800	800	300	
BM8	14.3	85.7	65.0	10.0	34.0	1,100	1,385	1	.00	15.4	850	800	800	700	900	360	
BM9	10.0	90.0	69.1	6.7	34.2	1,100	1,385	1	.00	15.9	1,050	1,050	1,050	600	800	300	
BM96	5.0	95.0	82.4	3.3	34.3	1,100	1,385	1	.00	16.6	1,500	1,500	1,500	510	660	770	
BT	0.0	100.0	65.7	34.3	1,245	1,385	2	.05	10.6	1,400	1,400	1,400	100	130	70	
											±200	±200	±200					
	MgO:TiO ₂	BaO:TiO ₂																
MB1	90.0	10.0	6.6	30.0	63.4	1,100	1,200	1	.01	15.5	18	18	18	3,000	600	1,700	800	
MB19	80.7	19.3	12.7	27.0	60.3	1,100	1,200	1	.00	16.9	19	19	19	3,000	7,000	10,000	1,500	
MB30	70.0	30.0	19.5	23.0	57.0	1,100	1,290	1	.00	17.0	21	21	21	3,000	8,000	10,000	1,200	
MB37	62.6	37.4	24.0	21.0	54.5	1,100	1,275	1	.03	13.3	20	20	20	1,900	3,000	5,000	250	
MB48	51.8	48.2	32.0	17.0	50.5	1,100	1,290	1	.01	15.1	41	41	40	750	750	400	
MB59	40.3	59.7	39.0	13.5	47.5	1,100	1,290	1	.01	15.7	94	94	96	500	340	130	
MB70	30.4	69.6	47.0	9.5	43.5	1,100	1,300	1	.01	15.9	200	200	200	600	250	110	
MB79	20.9	79.1	51.8	7.0	41.2	1,100	1,315	1	.01	15.3	350	350	350	900	300	130	
MB85	14.3	85.7	56.0	4.8	39.2	1,100	1,315	1	.01	17.2	540	540	540	2,000	300	180	
MB91	8.6	91.4	60.0	2.8	37.2	1,100	1,300	1	.00	17.3	800	800	800	450	600	300	
MB96	4.9	95.1	62.5	1.5	36.0	1,100	1,300	1	.00	16.2	1,060	1,060	1,060	400	600	300	

* Heat treated previously (see table 2).

The data for a given composition are considered to be the most representative among those obtained from measurements of 4 to 10 specimens.

Measurements at 3,000 mc/s and 25° C, made with the coaxial wave-guide instrument, gave values of K and Q that may be in error by a few percent for specimens of high dielectric constant. In testing such specimens with this instrument, the accuracy of measurements is dependent largely upon the exact determination of the average diameter of the central hole through the specimen. The relation of composition of the specimens to approximate maturing temperature is shown in figure 2. No attempts were made to determine the range in temperature for the production of mature specimens made from the individual preparations. When the compositions were in the region of $2\text{MgO}:\text{3TiO}_2$, the specimens were difficult to mature. Despite systematic variations in the duration and final temperature of the heat treatments, these specimens had 0.2 to 0.8 percent of absorption. Reheating the specimens, however, to the same or higher temperatures was effective in reducing the absorption, as illustrated by the data in table 2. These compositions were near that of a eutectic in the system MgO-TiO_2 [5].

The effects of systematically varying the composition of the specimens upon the values of K and Q may be obtained from the data in table 1. For example, when the content of

TABLE 2. Effect of consecutive heat treatments at the same or higher temperatures upon the absorption of some specimens

Specimen designation	Heat treatment		Absorption
	Temperature	Time	
	° C	hr	Percent
M2T3	1,275	3	0.64
	1,285	3	.53
	1,290	3	.00
MT	1,290	3	.22
	1,300	3	.00
	1,350	1	1.02
M2T	1,370	1	0.87
	1,380	1	.00
2BM1	1,310	1	.15
	1,310	1	.00
2BM2	1,310	1	.40
	1,310	1	.01

TiO_2 is maintained at a given percentage within the range 60 to 90 and MgO is substituted for BaO , the values of Q are affected more than those of K . Substituting a few percent (3 to 8) of MgO for BaO causes a rapid decrease in the values of Q from several hundred to 8 or 22 at a frequency of 1 mc/s and 25° C. Further substitution of MgO for BaO results in a gradual increase in the values of Q , which become high (350 to 4,000) when the substitution is complete.

A similar substitution of MgO for BaO causes slight increase in the relatively low values of K (34 to 74), followed by a decrease to the values of K (16 to 47) characteristic of specimens with compositions in this portion of the system MgO-TiO_2 (60 to 90 percent of TiO_2).

For specimens having compositions in the remainder of the system investigated, 33 to 60 percent of TiO_2 , the values of both K and Q are affected greatly by the substitution of MgO for BaO . In this region of compositions, there is a continuous decrease in the values of K from several hundred, characteristic of the barium-titanate specimens, to the low values of 12 to 14 typical of the specimens of the magnesium titanates. The variation in Q -values is typified by a gradual rise and fall followed by a rapid rise to the high values (5,000 to 10,000) characteristic of the magnesium titanate specimens with a content of TiO_2 less than 60 percent.

Some of the specimens having a relatively high content of BaO exhibited changes in the values of K and Q with time after the final heat treatment.

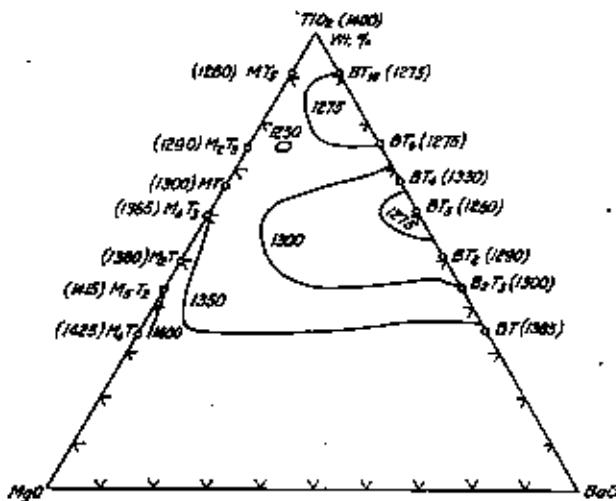


FIGURE 2. Approximate maturing temperature (° C) after calcining treatment.

B = BaO; M = MgO; T = TiO₂.

The data on K and Q in table 1, however, were obtained when such specimens had aged for 6 months or more, and no further changes in these properties were observed.

The stability of the dielectric constant and power losses with respect to time was determined by remeasuring the values of K and Q at 1 mc/s and 25° C for specimens stored 6 months at room temperature. Specimens containing more than 30 percent of BaO and less than 50 percent of TiO₂ had lower values of K and higher values of Q than when freshly prepared, as shown by the data in table 3. The changes in K and Q for specimen BM6 are illustrated in figure 3. These changes are reversible, because a partial restoration of the original values of K and Q occurred when the specimens were reheated to 600° or 700° C. Even a moderate heat treatment to approximately 100° C will increase the value of K of a specimen that previously had come to equilibrium at 25° C. For example, specimen MB91 exhibited a decrease in K from 897 to 808 in 6 months but when reheated to 85° C for 15 minutes and maintained at 25° C for 6 hours, the value of K was 873.

TABLE 3. Changes in K and Q of some specimens, at 25° C and 1 mc/s, after 6 months

Specimen designation	K			Q		
	After 1 day	After 6 months	Change	After 1 day	After 6 months	Change
	Percent			Percent		
MB48	34	34	0	480	480	0
MB50	79	76	-4	113	191	62
MB70	267	239	-14	156	316	103
MB76	343	323	-6	295	390	100
MB85	536	499	-7	235	455	94
MB91	897	808	-10	450	630	40
MB95	1,133	1,084	-9	450	670	49
BT	1,650	1,530	-7	74	107	45
BM3	28	37	+3	330	495	50
BM4	125	130	+4	79	100	27
BM6	266	226	-11	408	730	80
BM7	706	598	-23	220	470	114
BMB	1,090	848	-23	480	900	57
BMP	1,270	1,050	-17	440	790	80
BM95	1,770	1,550	-12	350	565	61
2B5M3	26	26	0	1,480	1,800	10
2B5M5	46	48	0	235	310	39
2B5M6	129	118	-10	250	330	152
2B5M7	156	167	+15	250	460	84
2B5M8	231	204	-12	148	253	73
2B5M9	260	228	-13	183	195	20

* After 2 years.

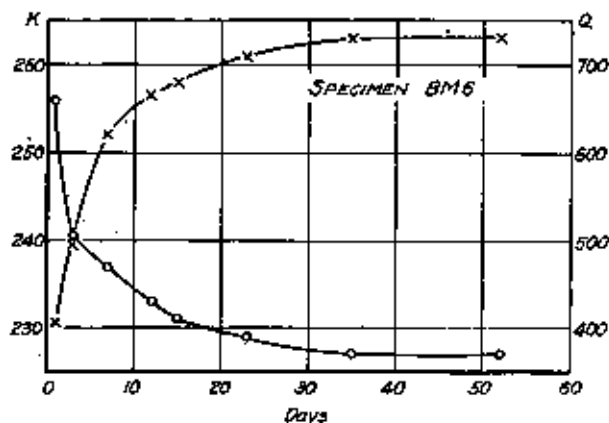


FIGURE 3. Changes in values of K and Q with time after the maturing heat treatment.

O= K ; X= Q .

In determining the stability of barium-magnesium titanates, some specimens were included from another investigation [1] on dielectrics with compositions in the system BaO-SrO-TiO₂, table 4. These data show that significant changes, mostly increases, in Q -values occurred with most of the specimens. The significant changes in K were decreases that occurred when the specimens contained more than about 30 percent of BaO and less than approximately 55 percent of TiO₂. In this group of specimens, the changes in K and Q were a maximum for specimens with about 40 percent of BaO.

Although decreases in Q -values may be attributed to moisture adsorption, it is very improbable that moisture also caused the values of Q to increase. Possibly slow changes in the crystalline structure, such as inversion or variation in the amount of solid solution, are associated with the instability of the specimens.

Although the dielectric constant and power losses of some specimens are affected by the thermal history, the respective values of K and Q become constant for given specimens when maintained at a constant temperature for several months. Thus, the thermal history of some titanates is one out of a number of factors that contribute to variations in the dielectric constant and power losses.

Changes in the temperature of the specimens affect the dielectric constant. The data in table 5, for specimens having compositions in the system BaO-MgO-TiO₂, were obtained by

TABLE 4. Changes in K and Q of some specimens, with compositions in the system BaO-SrO-TiO₂, after storage for more than 6 months

[Measured at 1 mc/s and 25° C]

Specimen designation	Composition weight			Storage period Months	K			Q		
	BaO	SrO	TiO ₂		After 1 day	After storage	Change	After 1 day	After storage	Change
B273.....		48.3	53.7	19	196	198	0	2,000	2,800	+80
B822.....	13.2	33.1	53.7	20	151	151	0	660	700	+4
B823.....	18.1	28.2	53.7	20	160	160	0	690	590	-10
B826.....	23.7	22.6	53.7	19	147	146	-0.6	1,450	1,750	+21
B826.....	28.0	18.3	53.7	17	133	132	-0.7	560	580	+4
B8271.....	32.8	12.5	53.7	23	153	150	-2	650	1,080	+36
B8279.....	36.7	9.6	53.7	23	203	191	-6	475	700	+23
B828.....	40.5	5.8	53.7	23	255	234	-8	172	266	+94
B829.....	43.5	2.8	53.7	23	214	205	-4	102	135	+33
B T 2.2.....	46.3		53.7	23	95	93	-2	75	81	+6
BT.....		55.4	43.6	7	258	256	-0.8	9,000	9,000	0
B281.....	6.2	50.2	43.6	20	255	257	+0.8	1,300	287	-1023
B282.....	13.2	43.2	43.6	20	283	286	+0.8	860	770	-90
B283.....	18.1	38.3	43.6	19	300	300	0	1,900	1,320	-580
B284.....	23.7	32.7	43.6	19	374	373	-0.5	1,030	900	-130
B285.....	28.1	28.2	43.6	18	451	447	-1	750	1,100	+350
B286.....	33.7	22.6	43.6	18	600	622	+6	630	700	+70
B2872.....	40.5	15.7	43.6	18	650	345	-305	167	315	+148
B288.....	46.6	9.6	43.9	18	1,070	985	-85	100	145	+45
B289.....	50.5	5.6	43.9	25	695	662	-33	66	90	+24
B2896.....	53.3	2.8	43.9	26	820	797	-23	52	60	+8
B273.....	56.1		43.9	29	900	845	-55	38	43	+5
B89.....	6.2	50.8	43.0	7	300	295	-5	4,700	4,800	+100
B890.....	32.8	28.2	39.0	7	872	863	-9	7,000	7,200	+200

* Changes are within experimental error in determinations.

measuring the dielectric constant at 1 mc/s and at 10-deg. intervals from -60° to +85° C. Because the temperature at each interval was maintained constant for 15 minutes only before measurements were made, equilibrium values of K were not found for specimens with high content of BaO. For stable specimens, the average values of temperature coefficient of K , last column of table 5, are considered to be not better than ± 10 ppm. or 5 percent, whichever is greater. Approximately half of these values are within the range +120 to +500 ppm. Where no values are given, computations of the coefficient of K were not made, because large irregularities appear in the curves for values of K plotted against temperature. In order to illustrate the variation of K resulting from changes in temperature and composition, figures 4, 5, and 6 were constructed for the temperatures -60°, 0°, and 60° C, respectively. These diagrams contain isodielectric-constant lines derived from the data in table 5.

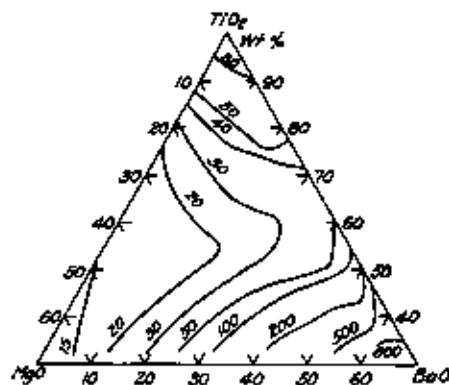


FIGURE 4. Constant K with varying composition at 1 mc/s at -60° C.

The most frequently observed value for the dielectric constant of MgTiO₃ was 17 (table 1), although values from 15 to 18 were found. Other investigators have reported values of 17 [6] and 14 [7]. Along the join MgTiO₃-BaTiO₃, the values of K change gradually from 17 for Mg-

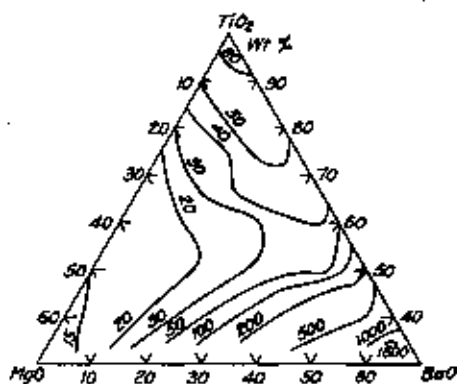


FIGURE 5. Constant K with varying composition at 1 mc/s at 0° C.

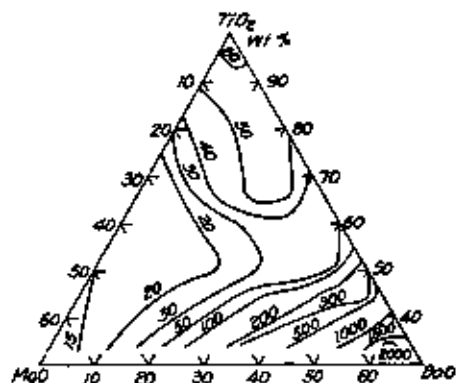


FIGURE 6. Constant K with varying composition at 1 mc/s at 60° C.

TiO_2 to about 1,400 for BaTiO_3 . The specimens with compositions in this join exhibit no peak values of K within the temperature range -60° to $+85^\circ$ C in contrast to the peaks observed previously for specimens with compositions in the join $\text{SrTiO}_3\text{-BaTiO}_3$ [1].

The effects of variation in frequency and composition on the ranges in values of Q , measured at 25° C, are illustrated in figures 7, 8, and 9 for frequencies of 50, 1,000, and 20,000 kc/s, respectively. The upper central portions of these diagrams show that, as the frequency is raised, there is a considerable enlargement in the area for compositions with very low ranges in values of Q . Changes in frequency had the least effect upon the ranges in values of Q for specimens with relatively high content of MgO and low content of BaO and of TiO_2 , as shown in the lower left regions of figures 7, 8, and 9.

With increasing frequency, 50 kc/s to 3,000

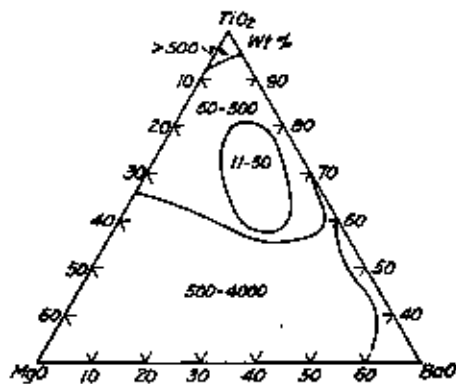


FIGURE 7. Ranges in Q -values with varying composition at 25° C at 50 kc/s.

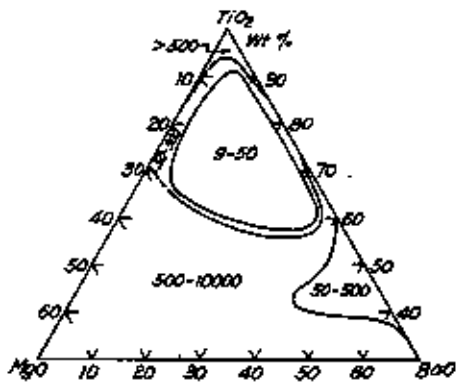


FIGURE 8. Ranges in Q -values with varying composition at 25° C at 1,000 kc/s.

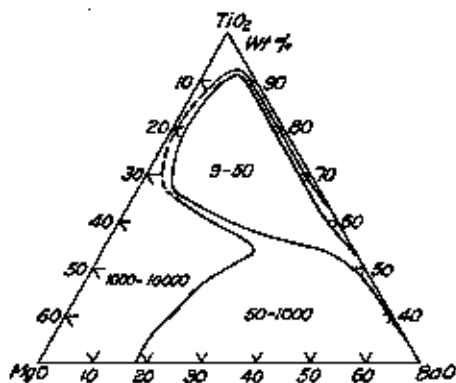


FIGURE 9. Ranges in Q -values with varying composition at 25° C at 20,000 kc/s.

me/s, the low values of K (16 to 78) tend to decrease when the content of TiO_2 ranges from 60 to 90 percent. For specimens having less than 60 percent of TiO_2 , the dielectric constant tends to remain constant at all frequencies used.

TABLE 5.—Dielectric constant, at 1 mc/s, from -60° to 85° C, and average temperature coefficient of dielectric constant

Specimen designation	Values of K at ($^{\circ}$ C)—															Average temperature coefficient of K		
	-60	-50	-40	-30	-20	-10	0	+10	20	30	40	50	60	70	80		85	
MT ₁	48.8	48.4	48.0	47.7	47.4	47.1	46.9	46.7	46.5	46.3	46.1	46.0	45.8	45.7	45.5	45.5	45.5	ppm -200
18BMa.....	78.6	78.9	77.1	77.3	77.4	77.5	77.7	77.7	77.8	77.8	77.8	77.8	77.8	77.8	77.8	77.9	77.9	+120
MT ₂	15.35	15.28	16.41	15.48	15.47	15.48	15.53	15.54	15.59	15.63	15.66	15.68	15.68	15.70	15.73	15.75	15.77	180
6BM2.....	21.45	21.7	21.95	22.2	22.45	22.65	22.9	23.15	23.35	23.6	23.7	23.8	23.9	24.0	24.1	24.15	24.15	810
6BM5.....	34.8	37.0	38.2	39.3	40.4	41.3	42.3	43.2	44.0	44.9	45.7	46.3	47.0	47.7	48.4	48.7	48.7	2,050
6BM8.....	49.5	51.3	53.5	55.3	57.6	59.8	62.0	63.5	65.0	67.0	69.4	70.2	72.4	74.7	77.0	78.3	78.3	3,100
6BM9.....	47.9	48.3	48.8	49.4	49.9	50.4	50.9	51.4	51.7	52.1	52.3	52.6	52.9	53.1	53.3	53.3	53.3	740
MT.....	16.90	16.93	16.96	16.98	16.99	16.99	16.99	16.99	16.99	16.99	16.99	16.99	16.99	16.99	16.99	16.99	16.99	160
4BM2.....	22.8	23.3	23.9	24.5	25.0	25.5	25.9	26.3	26.6	26.9	27.2	27.5	27.9	28.2	28.5	28.7	28.7	1,660
4BM4.....	28.6	29.6	30.8	31.9	33.0	34.1	35.3	36.3	37.4	38.6	39.5	40.3	41.0	41.7	42.3	42.6	42.6	2,650
4BM6.....	35.9	37.2	38.6	40.2	41.8	43.3	45.0	46.7	48.5	50.1	51.4	52.8	54.0	55.2	56.0	56.4	56.4	3,000
4BM7.....	38.3	39.8	41.5	43.0	44.7	46.4	48.1	49.8	51.5	53.2	55.0	56.6	58.0	59.3	60.5	61.3	61.3	3,170
4BM9.....	38.7	39.4	40.0	40.6	41.3	41.9	42.6	43.2	43.7	44.3	44.8	45.2	45.6	46.0	46.3	46.5	46.5	1,260
MB1.....	20.5	20.8	21.0	21.2	21.3	21.5	21.7	21.9	22.1	22.4	22.6	22.9	23.2	23.5	23.7	23.8	23.8	1,090
MB19.....	18.5	18.5	18.6	18.6	18.6	18.7	18.7	18.8	18.8	18.9	19.0	19.0	19.1	19.1	19.2	19.2	19.2	260
MB30.....	20.3	20.4	20.4	20.5	20.5	20.5	20.5	20.6	20.6	20.6	20.7	20.7	20.8	20.9	21.0	21.0	21.0	270
MB37.....	20.0	20.0	20.1	20.1	20.1	20.1	20.2	20.2	20.2	20.3	20.3	20.3	20.4	20.4	20.4	20.4	20.4	140
MB48.....	32.1	32.5	32.9	33.2	33.5	33.9	34.1	34.3	34.3	34.3	34.3	34.5	34.8	35.0	35.2	35.5	35.7
MB59.....	66	67	69.5	72	74	75.5	79	80.5	81	82.5	84.5	87	91	94.5	100	104	
MB70.....	182	193	205	217	230	243	256	263	263	266	263	275	290	310	337	363	
MB79.....	265	275	295	320	340	360	390	400	410	435	440	475	500	550	640	705	
MB85.....	350	370	400	430	470	510	560	600	610	630	670	730	820	900	1,100	1,300	
MB91.....	530	570	630	680	760	830	910	990	990	990	930	1,020	1,150	1,300	1,600	1,900	
MB95.....	680	700	800	950	1,100	1,250	1,300	1,250	1,200	1,200	1,200	1,370	1,700	2,200	3,200	3,900	
MT ₁	16.8	16.8	16.3	16.4	16.4	16.4	16.4	16.5	16.5	16.5	16.5	16.5	16.6	16.6	16.7	16.7	16.7	170
3BM2.....	17.0	17.1	17.2	17.2	17.3	17.3	17.4	17.5	17.5	17.6	17.6	17.7	17.7	17.7	17.7	17.6	17.6	320
3BM5.....	22.9	23.2	23.4	23.7	24.0	24.5	24.8	25.1	25.5	25.8	26.2	26.6	27.0	27.3	27.6	28.0	28.0	1,400
3BM6.....	26.0	26.5	27.0	27.5	28.1	28.6	29.1	29.7	30.3	31.0	31.7	32.3	33.0	33.6	34.1	34.4	34.4	1,940
3BM7.....	30.3	31.7	33.1	34.0	34.7	36.6	38.6	37.5	38.3	39.1	40.2	41.4	42.5	43.7	44.6	45.2	45.2	2,700
3BM8.....	36.0	36.5	37.7	39.8	41.5	42.9	44.0	44.0	45.9	46.9	47.9	49.8	49.8	50.8	51.0	51.4	51.4	2,400
3BM9.....	34.9	35.4	35.8	36.2	36.6	36.8	37.0	37.1	37.2	37.3	37.5	37.7	37.8	38.0	38.1	38.2	38.2	480
MT.....	13.75	13.8	13.8	13.8	13.85	13.9	13.9	13.9	13.95	14.0	14.0	14.0	14.1	14.1	14.1	14.1	14.1	190
2BM1.....	15.9	15.9	16.0	16.0	16.1	16.1	16.2	16.2	16.2	16.2	16.2	16.4	16.4	16.4	16.6	16.6	16.6	300
2BM2.....	16.6	16.7	16.7	16.8	16.8	16.9	16.9	17.0	17.0	17.0	17.1	17.1	17.2	17.2	17.3	17.3	17.3	300
2BM4.....	23.0	23.1	23.2	23.3	23.4	23.5	23.6	23.7	23.7	23.8	23.9	24.0	24.0	24.1	24.2	24.2	24.2	380
2BM5.....	27.5	27.6	27.8	28.0	28.1	28.3	28.4	28.5	28.6	28.7	28.7	28.8	28.9	29.0	29.2	29.2	29.2	440
2BM6.....	30.1	30.2	30.4	30.5	30.7	30.8	31.0	31.1	31.2	31.3	31.4	31.5	31.7	31.9	32.0	32.1	32.1	440
2BM7.....	31.6	32.0	32.3	32.5	32.8	33.0	33.1	33.2	33.3	33.5	33.6	33.8	34.0	34.2	34.4	34.5	34.5	620
2BM8.....	31.1	31.3	31.5	31.7	32.0	32.1	32.2	32.3	32.4	32.5	32.7	32.8	33.0	33.2	33.3	33.4	33.4	490
2BM87.....	34.6	34.9	35.1	35.4	35.6	35.7	35.8	35.9	36.1	36.3	36.5	36.7	36.9	37.1	37.2	37.2	37.2	560
2BM9.....	87	90	93	96	99	102	105	107.5	110	111.5	113.5	115.5	117.5	120	122	123	123	2,300

BM7T	14.4	14.4	14.4	14.4	14.4	14.5	14.55	14.6	14.6	14.6	14.6	14.7	14.7	14.7	14.7	14.7	14.7	14.7	14.7
2B ₁ M ₁	16.6	15.9	16.9	16.0	16.0	16.0	16.1	16.1	16.1	16.1	16.1	16.2	16.2	16.2	16.2	16.2	16.2	16.2	16.2
2B ₂ M ₁	17.3	17.4	17.4	17.4	17.5	17.5	17.5	17.6	17.6	17.6	17.6	17.7	17.7	17.7	17.7	17.7	17.7	17.7	17.7
2B ₃ M ₁	21.3	24.4	24.8	24.9	26.0	26.1	26.3	26.4	26.6	26.8	26.8	27.1	27.1	27.1	27.1	27.1	27.1	27.1	27.1
2B ₄ M ₁	38.9	39.8	41.0	42.0	43.2	44.2	45.6	45.9	46.8	47.3	48.2	48.4	48.4	48.9	49.7	50.9	51.7	52.7	53.7
2B ₅ M ₁	89	95	103	106	112	118	125	126	135	138	143	148	149	167	178	196	200	207	210
2B ₆ M ₁	132	141	151	158	167	175	182	188	199	205	215	219	227	243	257	283	295	300	305
2B ₇ M ₁	172	181	190	197	206	208	210	211	217	225	232	240	246	259	280	298	306	308	310
2B ₈ M ₁	192	190	200	206	213	224	231	233	240	249	260	270	284	294	300	300	310	310	310
BM7	12.35	12.4	12.4	12.4	12.4	12.4	12.4	12.45	12.5	12.5	12.5	12.5	12.5	12.55	12.6	12.6	12.6	12.6	12.6
BM1	15.4	15.4	15.4	15.45	15.45	15.45	15.5	15.5	15.5	15.5	15.5	15.5	15.5	15.55	15.6	15.6	15.6	15.6	15.6
BM2	20.4	20.5	20.7	20.9	21.1	21.1	21.2	21.3	21.5	21.7	22.0	22.3	22.3	22.4	22.5	22.5	22.5	22.5	22.5
BM3	32.7	33.4	34.0	34.5	35.0	35.5	36.1	36.4	37.1	38.1	39.3	40.5	41.8	42.9	44.1	44.9	44.9	44.9	44.9
BM4	79	82	86	92	98	105	114	115	120	126	133	144	144	158	179	186	186	186	186
BM5	160	170	185	196	210	226	240	245	257	270	287	305	305	330	370	402	402	402	402
BM6	370	400	440	480	520	560	600	630	650	690	760	830	830	900	1,010	1,100	1,100	1,100	1,100
BM7	680	690	690	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700
BM8	680	690	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700
BM9	680	690	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700	700
BM9S	680	1,040	1,150	1,200	1,260	1,330	1,400	1,440	1,490	1,550	1,600	1,660	1,700	1,750	1,800	1,850	1,900	1,950	2,000

Increasing frequency causes more irregular variations in Q -values than in those of K . For example, specimens with compositions in the system $MgO-TiO_2$ tend to exhibit higher values of Q at frequencies of 1 and 20 mc/s than at those of 50 kc/s and 3,000 mc/s. This trend is shown also by the barium titanates with a content of TiO_2 from 67 to 90 percent. The opposite tendency, lower values of Q at intermediate frequencies than at the extremes of frequency, is exhibited by specimens having compositions on some of the joins (60 to 67 percent of TiO_2) between magnesium titanates and barium titanates (specimens 4BM2, 4BM4, 3BM5, and 3BM6 in table 1).

The percentage of linear thermal expansion was fairly high, except the lower value of M_2T_3 , despite a wide variation in composition (table 6). These dielectrics would be cracked by local heating to high temperatures. Consequently, preheating at a slow rate would be necessary in order to solder connections to the metal-coated dielectrics.

TABLE 6. Linear thermal expansion

Specimen designation	Temperature range from 25° C to —						
	100° C	200° C	300° C	400° C	500° C	600° C	700° C
	Per-cent	Per-cent	Per-cent	Per-cent	Per-cent	Per-cent	Per-cent
MT ₁	0.06	0.13	0.21	0.31	0.40	0.50	0.60
4BM5.....	.09	.15	.25	.36	.46	.57	.67
MT.....	.06	.15	.25	.36	.46	.58	.70
MB37.....	.06	.15	.26	.36	.48	.57	.71
M ₁ T.....	.06	.15	.25	.34	.44	.55	.67
M ₁ T ₂04	.10	.19	.26	.32	.39	.46

IV. Summary

Dielectrics having compositions indicated by points in the system $BaTiO_3-4MgO:TiO_2-TiO_2$ can be prepared from mixtures of titanium dioxide with barium and magnesium carbonates.

Mature specimens, less than 0.1 percent of absorption, result from dry-pressing these calcined mixtures and heating the disks thus formed to various temperatures within the range 1,250° to 1,425° C.

The dielectric constant, K , of matured specimens varies from 12 (high content of MgO) to several hundred (high content of BaO). Most of the specimens have positive temperature coefficients of K . The Q -values range from 8 to 10,000 (high content of either MgO or TiO_2). The values of K and Q are affected by the thermal history of specimens that have a content of BaO greater than 30 percent and a content of TiO_2 less than 50 percent. The dielectric constant decreases and Q -values increase for several weeks after these specimens receive the final heat treatment. Although reheating causes a reversal of those changes in K and Q , a decrease of K and an increase of Q again occur with time. After remaining at a constant temperature for a few months, these specimens have constant values of K and Q .

Relatively high values of linear thermal expansion were obtained with specimens that varied widely in composition.

V. References

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