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Stress Corrosion of Wrought Magnesium Base Alloys By Hugh L. Logan and Harold Hessing

The rates of corrosion of many metals and alloys are increased by the presence of tensile stresses in the materials. In order to determine the effect of tensile stress on the behavior of magnesium alloys exposed to corrosive media, stress-corrosion tests were made on several magnesium base alloys at two weather exposure sites, and in the laboratory by continuous immersion in a NaCl+ K_2 CrO₄ solution and by intermittent immersion in a dilute NaCl solution.

The periods of time to failure of specimens exposed under tensile stress in the atmosphere at the National Bureau of Standards were less than those for the same materials exposed in a marine atmosphere at Hampton Roads, Va.

The M1 clad AZ31X-h sheet alloy proved more resistant to stress corrosion than bare AZ31X-h, AZ51X, or AZ61X sheet alloys and AZ80X extruded alloys.

Predictions of relative susceptibility of materials to stress corrosion from intermittent immersion laboratory tests in a 0.01-percent NaCl solution were in good agreement with the results obtained from weather exposure tests at this Bureau.

I. Introduction

Early in World War II a number of welded airplane booms, constructed of AZ61X magnesium alloy sheet and extrusions were subjected to static loading until buckling occurred and were then stored in the open exposed to the weather. Subsequent examination revealed stress corrosion in the buckled areas.

Inasmuch as the Bureau of Aeronautics, Navy Department, was interested in the possibilities of the use of magnesium alloys in aircraft structures, it gave financial support to the National Bureau of Standards to study the resistance of various commercial wrought alloys to stress corrosion. The investigation was carried out with the specific purpose of determining the relative susceptibility of various alloys to stress corrosion, and whether a correlation could be established between the results of various laboratory tests and those of weather exposure tests.

II. Materials and Methods of Test

1. Materials

Seven wrought magnesium alloys were studied in this investigation. Four of these materials, the M1, AZ31, AZ61, and AZ80 alloys had been used in this country in structural applications prior to the war. An AZ31 alloy clad with thin layers of the M1 alloy; an alloy designated as AZ51 and a zinc, zirconium alloy, designated by Dow as ZK60, all developed by the Dow Chemical Co. during the war, were also included in the study. The materials were supplied by the American Magnesium Corp. and the Dow Chemical Co. The ASTM and manufacturers' designations, and the nominal chemical compositions are given in table 1.

Materials received from the Dow Chemical Co. generally had an oiled finish. After the sheet or extruded material had been machined into specimens the thickness was reduced approximately 0.002 in. by immersion in an aqueous solution containing 30 ml of concentrated HNO₃, and 10 ml of concentrated H₂SO₄ per liter.

Materials received from the American Magnesium Corp. had a chrome-pickled finish. They were machined into tensile specimens, and the chrome-pickle film was then removed by immersion in an aqueous solution containing 22.5 g of Na₂CO₃, and 15 g of NaOH per liter.

 TABLE 1. ASTM and manufacturers' designations, and nominal compositions a of the magnesium base alloys investigated

I	Che tion	Chemical composi- ion Mg=remainder				
ASTM	American Magne- sium Corp.	Dow Chemical Co.	Al	Zn	Mn	Zr
			%	%	%	%
M1	A M38	M			1.5	
AZ31X	AM-C52S	FS-1	3.0	1.0	0.3	
AZ51X		JS-1	5.0	1.0		
AZ61X	AM-C57S	J-1	6.5	0.7	. 2	
AZ80X	AM-C58S	0-1	8.5	. 5	. 2	
		^b ZK-60		5.7	.07	0.6
MlClad AZ31X		MClad FS-1				
AZ31X		Core	3.0	1.00	0.3	
M1		Cladding			1.5	

*Nominal compositions given are those prevailing at the time the materials were received for test. Analyses furnished by the manufacturers did not differ significantly from the above values.

 $^{\rm b}\operatorname{Composition}$ determined by chemical and spectrochemical means at this Bureau

After the treatment to remove the chromepickle film or the outer layers of material, all of the specimens except the clad AZ31X-h material, whose thin cladding would have been seriously affected, were ground parallel to the long axes of the specimens, prior to testing, using number 400 Aloxite belts on a rubber backed wheel. The chemical and grinding treatments were used to obtain uniform surface conditions on specimens before exposure.

In general, the alloys supplied in sheet form were tested in each of two tempers, annealed (-a)and hardened by rolling (-h), and also in two thicknesses 0.064 and 0.125 in. The extruded material (except the ZK-60 alloy) was supplied in two rectangular cross sections. The heattreatable extruded alloy AZ80X was supplied both in the as-extruded and in the extruded, heattreated and aged (HTA) conditions. The tensile properties of the sheet materials as received are given in tables 2, 3, and 4. The sheet material was rolled by the usual manufacturing process, except that two lots of material (see table 4) were rolled by a special process developed by M. A. Hunter of Rensselaer Polytechnic Institute. The cross-sectional dimensions and tensile properties of the extruded materials are given in table 5.

TABLE 2. Tensile properties of the AZ61X and M1 magnesium base sheet alloys and threshold stresses (as determined in the $NaCl+K_2CrO_4$ solution) of the AZ61X alloy

Alloy and condition a	Supplier b	Nominal thickness	Specimen orienta- tation °	Tensile properties of uncorroded material ^d			Threshold stress	
				Yield strength	Ultimate tensile strength	Elongation in 2 in.	Average	Percentage of yield strength
		in.		$lb/in.^2$	$lb/in.^2$	Percent	$lb/in.^2$	Percent
AZ61X-a	Dow	0.064	L	24, 300	42, 700	20	21, 500	88
Do	do	.064	Т	27, 800	39, 800	7	23, 500	85
Do	Am. Mg. Corp	.064	L	23, 100	41, 300	19	22, 500	97
Do	do	,064	Т	27, 200	41,000	17	24,000	89
AZ61X-a	Dow	.125	L	23,600	37, 800	7	22, 500	95
Do	Am. Mg. Corp	.125	L	23, 200	39, 400	9	21,000	90
Do	do	.125	T	26, 500	40,000	7	23,000	87
						1.		
AZ61X-h	Dow.	.064	L	35,600	46, 200	6	28,000	79
Do	do	. 064	Т	37, 800	50, 400	14	31,000	82
Do	Am. Mg. Corp	.064	L	39, 300	47, 100	5	28, 500	72
Do	do	.064	Т	41, 200	51, 600	8	28,000	69
1 g (1 g () ()))))))))))))			*	67	· · · ·			
AZ61X-h	Dow	. 125	L	33, 300	43, 800	10	29,000	87
D0	do	. 125	Т	35, 700	47, 500	7	31, 500	88
Do	Am. Mg. Corp	. 125	L	37, 500	45, 500	5	29,000	77
Do	do	. 125	Т	36, 400	48, 200	7	31,000	85
Ml-h	Dow	.064	L	29,300	34, 800	3		
\mathbf{D}_{0}	do	.064	Т	22, 400	34, 300	13		
Do	Am. Mg. Corp.	.064	L	30. 700	35.700	4		

a "a" indicates annealed material; "h" material hardened by rolling.

^b Dow indicates Dow Chemical Co.; Am. Mg. Corp. indicates American Magnesium Corp.

° L and T designate, respectively, longitudinal and transverse specimens.

^d Each value is the average of 5 or more determinations.

TABLE 3. Tensile properties and threshold stresses (as determined in the $NaCl+K_2CrO_4$ solution) of AZ31X magnesium base sheet alloy

	Supplier	Nominal thickness	Specimen orienta- tion ^b	Tensile properties of uncorroded material °			Threshold stress	
Alloy and condition a				Yield strength	Ultimate tensile strength	Elongation in 2 in.	Average	Percentage of yield strength
4.721X o	Dow	in.	т	$\frac{lb}{in.^2}$	$\frac{lb/in.^2}{27,200}$	Percent	$\frac{lb}{in.^2}$	Percent
AZ31A-a	do	0.004		22, 500	37,200	20	20,000	02
Do	Am Mg Corp	.004	T.	22, 300	36,100	19	18,500	93
Do	do	064	T	19,300	36,200	22	18,000	93
D0		.001	· ·	13, 500	50, 200	22	10,000	
AZ31X-a	Dow	. 125	\mathbf{L}	21, 500	36, 600	22	19,500	90
Do	do	. 125	T	22, 900	37,200	20	20,000	88
Do	Am. Mg. Corp.	. 125	\mathbf{L}	20, 200	35,900	21	19,000	94
Do	do	. 125	Т	19, 500	35,800	22	18,000	92
AZ31X-h	Dow	. 064	\mathbf{L}	33, 400	42,800	10	30,000	89
Do	do	. 064	т	32, 500	42,700	11	29, 500	90
Do	Am. Mg. Corp	. 064	\mathbf{L}	35, 500	41,800	7	29, 500	84
Do	do	. 064	т	30, 600	43,300	16	28, 500	94
AZ31X-h	Dow	. 125	\mathbf{L}	31, 800	41, 400	12	29,000	92
Do	do	. 125	т	32, 400	43, 400	17	29, 500	91
Do	Am. Mg. Corp	. 125	\mathbf{L}	36, 200	42,600	8	29, 500	81
Do	do	. 125	Т	33, 000	45,000	15	28, 500	86

a "a" indicates annealed material; "h" material hardened by rolling.

^b L and T designate, respectively, longitudinal and transverse specimens.

• Each value is the average of 5 or more determinations.

TABLE 4. Tensile properties and threshold stresses (as determined in the NaCl+K₂CrO₄ solution) of the AZ51X-h and the AZ31X-h magnesium base sheet alloys 0.064 in. thick, rolled by two processes and supplied by the Dow Chemical Co.

	Specimen	Tensile p	roperties of un material ^b	Threshold stress		
Alloy and condition	$\operatorname{orienta-}_{d}$	Yield strength	Ultimate tensile strength	Elongation in 2 in.	Average value	Percentage of yield strength
ROLLE	D BY TH	E STANDAR	D DOW PR	OCESS		
1 - 2 - 3		$lb/in.^2$	$lb/in.^2$	Percent	$lb/in.^2$	Percent
AZ51X-h	\mathbf{L}	34,800	47,000	11	34,000	98
A Z51 X-h	т	40,600	50, 600	15	37,000	91
AZ31X-h	\mathbf{L}	32,900	43, 100	10	29,000	88
AZ31X-h	Т	34,800	44, 700	12	29,000	83
ROL	LED BY 7	THE HUNTE	R PROCES	3		
AZ51X-h	L	34,300	46,000	11	33,000	96
AZ51X-h	T	39, 200	49, 300	12	36, 500	93
AZ31X-h	\mathbf{L}	32, 700	42, 600	11	29,500	90
AZ31X-h	Т	34, 200	44,000	18	31, 500	92

^a L and T designate, respectively, longitudinal and transverse specimens

^b Each value is the average of 5 or more determinations.

TABLE 5. Tensile properties and threshold stresses (as determined in the $NaCl+K_2CrO_4$ solution) of the 5.7-percent zinc, 0.6-percent zirconium (Dow ZK-60), the AZ61X, and the AZ80X magnesium base alloy extrusions

		Nominal dimensions		Tensile properties of uncorroded material ^a			Threshold stress	
Alloy and condition	Supplier	Thickness	Width	Yield strength	Ultimate tensile strength	Elongation in 2 in.	Average	Percentage of yield strength
TESTED AS EXTRUDED								-
ZK-60	Dow	in. 1⁄16	$\frac{in.}{1}$	$lb/in.^{2}$ 39, 500	$\frac{lb/in.^2}{50,000}$	Percent 13	$\frac{lb}{in.^2}$ 33,000	Percent 84
AZ61X	Dow.	1/16	3/4	25,800	43,900	12	23, 500	91
Do	Am. Mg. Corp	1/16	1	29, 500	42,900	16	27, 500	87
Do	Dow	1/8	11/4	26, 900	44,800	17	23, 500	87
Do	Am. Mg. Corp	1/8	$1\frac{1}{4}$	31,800	43, 600	16	25,500	81
AZ80X	Dow	1/16	3/4	28, 900	47, 200	12	26, 500	92
Do	Am. Mg. Corp	1/16	1	29, 900	46, 800	17	25,000	84
Do	Dow	1/8	$1\frac{1}{4}$	31,000	48,400	12	26,500	85
Do	Am. Mg. Corp	1⁄8	$1\frac{1}{4}$	34, 500	48, 500	19	27,000	78
	EXT	RUDED, HI	EAT TREA	TED, AND	AGED			
4°Z80X-HT 4	Dow	1/1/4	34	34 200	49,000	7	32 500	95
Do	Am Mg Corp	716	1	29,000	49,200	13	27, 500	95
Do	Dow	1.6	114	37,200	49,600	4	32,000	86
Do	Am. Mg. Corp	1/8	11/4	35, 100	50, 600	10	31, 500	90
6								

^a Each value is the average of 5 or more determinations.

2. Preparation of Specimens

Tests were made with $\frac{1}{2}$ -in. reduced-section standard ASTM tension specimens for sheet metals, except that the grip ends were 1 or $1\frac{1}{4}$ in. wide, when possible, instead of the standard $\frac{3}{4}$ in.,¹ to minimize the failure of specimens as the result of stress corrosion around the bolt holes. Specimens of the sheet material machined with their long axes in and at right angles to the direction of rolling of the sheet, are designated hereafter as longitudinal (L) and transverse (T) specimens, respectively.

3. Methods of Test

Exposure tests were made on stressed specimens on the roof of the Northwest Building at this Bureau and in a marine atmosphere at Hampton Roads, Va. The stress-corrosion racks at Hampton Roads have been described in an earlier paper.² Those at this Bureau are shown in figure 1, and the method of supporting the specimens, using flexure plates, is shown in figure 2. The ends of the specimens were gripped between two pieces of 61S-T6 aluminum alloy bar by means of %-in. bolts of 17S-T4 aluminum alloy.

A stress-corrosion crack that developed in the reduced section of one of the specimens exposed in the weather is shown in figure 3. Loose³ has suggested that a specimen should be considered to have failed when a stress-corrosion crack first appears. It was not possible with the personnel available to make the necessary inspections, at



FIGURE 1. Stress-corrosion racks on the roof of the Northwest Building at the National Bureau of Standards.

 $^{^1}$ One set of extrusions supplied by the Dow Chemical Co. had a rectangular cross section 34 by $\frac{1}{26}$ in.

² H. L. Logan and H. Hessing, J. Research NBS 41, 69 (1948) RP1905.

Box at extreme left houses electric solenoid counters to determine exposure period of specimens to failure.

 $^{^{3}\,\}mathrm{W.}$ S. Loose, private communication, Dow Chemical Co., Midland, Mich.



FIGURE 2. Method of supporting stress-corrosion specimens in rack shown in figure 1, using flexure plates above and below stressed specimens.

Fixtures were 618–T6 aluminum alloy; bolts, 178–T4 aluminum alloy. Knife switches were connected in series with solenoid counters shown in figure 1. Unstressed specimens were mounted adjacent to stressed specimens.

either weather exposure site, to determine when stress-corrosion cracks first appeared. Accordingly, the period to failure for the various specimens was determined as the interval between insertion and actual failure as recorded with electric solenoid counters that have been described in an earlier paper (see footnote 2).

At the time this investigation was begun other workers in the field were advocating the testing of sheet magnesium alloys in the laboratory by continuous immersion, under stress, in a sodium chloride-potassium chromate solution. "Threshold" stresses, defined as the maximum stresses that materials can withstand without failure for a fixed period of time when immersed in the corroding medium, were determined as a measure of the resistance of the alloys to stress corrosion.

The specimens tested in corroding solutions at this Bureau were immersed in glass cells and stressed by means of lever systems. The apparatus used in an earlier investigation (see footnote 2) was modified for these tests by the insertion, in the supports, of flexure plates (made of stainless steel sheet 1 inch wide by 0.020 inch thick), which are modifications of those used by Loose and Barbian (see footnote 3). The corrosive medium, a solution containing 3.5-percent NaCl and 2.0percent K_2CrO_4 was suggested by R. H. Brown and G. F. Sager.⁴

Preliminary work in this laboratory and results obtained by other workers indicated that different values for the threshold stress were obtained depending on whether the specimens were loaded in tension before or after the solution was added. Work hardening of the specimens following loading and prior to adding the solution gave threshold stresses considerably higher than those obtained if the solution was added prior to the application of the major part of the load. Therefore all succeeding tests were made by loading the specimens after the solution was added to the cells.

In order to supplement and possibly replace the continuous immersion test in the NaCl+



FIGURE 3. Stress-corrosion crack in specimen exposed to weather at the National Bureau of Standards.

⁴ R. H. Brown, private communication (Aluminium Research Laboratories, New Kensington, Pa.)

Stress Corrosion of Magnesium Alloys



FIGURE 4. Specimens mounted for intermittent immersion tests in the laboratory.

Flexure plates were used in the supports above and below specimens, corrodent was raised from flasks (see lower right) to cells by means of compressed air.

 K_2CrO_4 solution, intermittent immersion tests were made in various corrosive media at a constant temperature of 95° F (35° C). Specimens were mounted in cells as shown in figure 4, and the corroding solution was raised into the cells by means of compressed air, wetting the specimens four times per hour. They were immersed for approximately 1 min and were permitted to dry for 14 min in each cycle. Intermittent immersion tests in a 3.5-percent sodium chloride solution produced such severe pitting of the specimens that stress-corrosion effects were obscured. Intermittent immersion in a 0.01-percent NaCl solution or in a 0.01 normal NaHCO₃ solution produced stress corrosion in magnesium alloys without producing general or pitting corrosion. Exposure periods to failure for a given alloy were approximately the same in either of these dilute solutions. However, the 0.01-percent NaCl solution was used for the balance of the intermittent immersion tests.

III. Results and discussion of Stress-Corrosion Tests

1. Weather Exposure Tests

The results of weather exposure tests on the various magnesium alloys are shown graphically in figures 5 to 9, inclusive, and are given for the AZ31X-h material in table 6. The plotted values in figure 5 generally represent the averages of three or more specimens exposed at a given stress. The results for individual M1-Clad specimens (exposed in the marine atmosphere), however, are shown as separate points in the figure. All the specimens exposed at a given stress, and duplicate specimens stressed differently were not necessarily exposed concurrently.

The data indicate that the M1-clad AZ31X-h material was more resistant to stress corrosion at high stresses than any of the other materials studied. Specimens of this material may be expected to have a life in a marine atmosphere of at least 500 days at a stress of 30,000 lb/in.² The bare (unclad) material, exposed in a marine

TABLE 6. Results of stress-corrosion tests on the AZ31X-h magnesium base alloy, 0.064 in. thick, exposed in the weather at the NBS, Washington, D. C. and at Hampton Roads, Va.

	Specim V	ens expo Vashingt	osed at NBS, on, D. C.	Specin Han	ecimens exposed at Tampton Roads, Va.			
A verage stress	Num-	Perio	d to failure	Num-	Period to failure			
	speci- mens	Aver- age	Range	speci- mens	Aver- age	Range		
<i>lb/in</i> ² 32,000		Days	Days	7	Days 13. 5	<i>Days</i> 8.3 to 20		
31,000 30.000				8. 15	14.3 22.1	10.8 to 20 0.5 to 73		
29,000	4	14	10 to 23	13	21.2	11.5 to 68		
28,000	σ	17, 3	1 to 47	24	31.3	12.5 to 81		
27,000	23	13.2	5 to 35	15	36.5	16.0 to 55		
26,000 25,000	4	26.5	17 to 45	9	55.0 41.0	33.0 to 115 20.5 to 69		
23,000 24,000	a 12	26.8	13 to 53	12	76.1	30.0 to 130		
24,000	b 3	27.3	20 to 45		~~			
23.000	8	26.8	16 to 45	/				
22,000				7	127	5.6 to 150		
20,000	a 44	95. 2 159	11 to 280		914	76 to 222		
20,000	b 6	132	144 to 280			70 10 320		
90,000	4.9	20	C1 to 79					
20,000	e 3	69 37	$\frac{64}{37}$ to $\frac{72}{5}$					
18,000	a 37	97.4	32 to 255					
18,000	e 3	89	74 to 104					
16,000	e 6	131	26 to 248					

^a Total number specimens tested under designated stress.

^b Tested simultaneously with 0.125-in. thick material.

° Tested simultaneously.

 $^{\rm d}$ Tested simultaneously with ${\rm AZ51\,X}$ and ${\rm AZ61\,X}$ material.

e Tested simultaneously with annealed material.



FIGURE 5. Stress-exposure period to failure curves for magnesium alloys exposed to the weather.

 \Box , M1 clad AZ31X-h sheet exposed at NBS; \blacksquare , M1 clad AZ31X-h sheet exposed at Hampton Roads; \triangle , ZK60 extrusions exposed at NBS; \bigcirc , AZ31X-h sheet exposed at NBS; \bullet , AZ31X-h sheet exposed at Hampton Roads; X, M1-h sheet exposed at NBS; \rightarrow , specimen had not failed.

atmosphere, fractured under stresses of 20,000 lb/in.² in an average period of 214 days. The clad specimen thickness was 0.080 in. compared to 0.064 in. for the bare alloy. This difference in thickness of the two materials would not be expected to produce more than a small difference in the expected life of specimens of the same material. The results, therefore, showed that the clad material was definitely more resistant to stress corrosion than the bare material.

Data shown in figure 5 and given in table 6 indicate that the exposure period to failure was generally shorter for specimens exposed under a given stress at the National Bureau of Standards site than for the same material exposed under the same stress in a marine atmosphere at Hampton Roads, Va. As a further check, 18 specimens were machined from the same sheet of the AZ-31X-h alloy and were exposed to the weather on the same date and at the same stress, 20,000 lb/in.², one group at this Bureau, the other at Hampton Roads. The average period to failure for specimens exposed at the National Bureau of Standards was 152 days and at Hampton Roads, 214 days: ranges were 61 to 263 days and 76 to 323 days. respectively. No analysis of the atmosphere in the vicinity of the racks at this Bureau was made, but the smoke stack on an adjacent building. shown in figure 1, in all probability increased the corrosiveness of the atmosphere in the neighborhood of the stress-corrosion racks.

The results shown in figure 5 also indicate that the M1 sheet and the ZK-60 extruded materials

Stress Corrosion of Magnesium Alloys

(exposed at the National Bureau of Standards site) were relatively more resistant to stresscorrosion at moderate stresses than other bare materials studied. The M1 alloy specimens stressed at 16,000 lb/in.² at the National Bureau of Standards site had not failed in 857 days. Specimens of the AZ31X-h alloy placed under test on the same date at the same stress failed after an average exposure period of 131 days (range 26 to 248 days). Two ZK-60 specimens exposed under stresses of 20,000 lb/in.² failed after 555 and 596 days exposure. Three specimens



FIGURE 6. Resistance to stress corrosion of hardened (h) and annealed (a) AZ31X sheet materials exposed to the weather at the National Bureau of Standards site at three different stresses.

1, Strees, 16,000 lb/in.²; 2, stress, 18,000 lb/in.²; 3, stress, 20,000 lb/in.²; +, average value; \longleftrightarrow , range.



FIGURE 7. Effect of specimen thickness on the resistance of the AZ31X-h material to stress corrosion.

1, Stress 20,000 lb/in.²; 2, stress, 24,000 lb/in.²; +, average value; \longleftrightarrow , range. A, 0.064-in. material; B, 0.125-in. material.

exposed at the same stress had not failed in 600 days.

It was noted that, although an occasional specimen failed during a period of dry weather, most failures occurred during the drying period after the specimens had been wet by rain. This fact is believed to account, at least in part, for the wide variations in exposure periods required for failure, for specimens subjected to the same stress, seen in table 6 and shown graphically in figures 6 to 9 inclusive. Because of the increased number of failures occuring after rain had fallen, direct comparisons of weather exposure results, shown in figures 6 to 9, inclusive, were made on specimens inserted in the racks on the same dates under the same stresses.

Results obtained on the AZ31X alloy in the annealed and rolled conditions (see fig. 6) indicated that for stresses of 18,000 and 20,000 lb/in.² the hardened material was more resistant to stress-corrosion than the annealed material. Data obtained at 16,000 lb/in.² are not so conclusive because of the small difference between the average values and the large scatter in the observed values.

The effect of specimen thickness on the resistance of the AZ31X-h material to stress corrosion is shown in figure 7. The average period to failure for the 0.125-in. thick material was greater than that for the 0.064-in. thick material. However, the scatter was such that it is not possible to say that any particular 0.064-in. specimen would fail



FIGURE 8. Effect of aluminum content on the stress corrosion behavior of three magnesium alloys.

AZ61X contained 6.5% Al; AZ51X, 5.0% Al; AZ31X, 3.0% Al. 1, Intermittent immersion in 0.01% NaCl solution; A, AZ61X-h; B, AZ51X-h; C, AZ31X-h. 2, Weather exposure NBS; A, AZ61X-h; B, AZ51X-h; C, AZ31X-h. Stress=20,000 lb/in.²; +, average value; $\leftarrow \rightarrow$, range.

Journal of Research



FIGURE 9. Resistance to stress corrosion of two extruded magnesium alloys under stress of 20,000 lb/in².

1, AZ80X, 2, AZ61X; weather exposure. 3, AZ80X, intermittent immersion. —+, average value; \longleftrightarrow , range. H=heat treated and aged; E=as extruded.

in a shorter period of time than a given 0.125-in. thick specimen.

The results discussed above for the M1-h and AZ31X-h material and the results of tests on the AZ31X-h, AZ51X-h, and AZ61X-h alloys, shown at the right in figure 8 indicate that at moderate stresses the resistance of the wrought magnesium alloys to stress-corrosion decreased with increased aluminum content up to about 6.5 percent of aluminum.

Although it was badly discolored, the ZK-60 extruded alloy, as indicated above, was the most resistant to stress corrosion of any of the extruded materials studied. The average periods to failure and the ranges for the AZ80X alloy exposed as extruded and after heat treatment and aging, and for the AZ61X extruded alloy (4 to 6 specimens of each alloy) all exposed in the weather at this Bureau under stress of 20,000 lb/in.² are shown

Stress Corresion of Magnesium Alloys

graphically in figure 9. It can be seen that the average period to failure of the heat-treated and aged AZ80X material was greater than the maximum value for the material as extruded. Because of the scatter obtained with both materials, it was difficult to say whether the AZ61X was more susceptible to stress-corrosion than the AZ80X-HTA alloy.

Ultimate tensile strengths of the bare AZ31X-h, the ZK-60, and the AZ80X alloys, determined after specimens had been exposed unstressed in the weather at this Bureau for periods of 750 to 800 days, were at least 94 percent of those of the unexposed material. Changes in ductility (measured in percentage elongation in 2 in.) varied with the different alloys; values for the exposed ZK-60 and AZ31X-h materials were 85 percent or more of those of the unexposed materials. No elongation data were obtained on the AZ80X material.

The decrease in the ultimate tensile strength of M1-clad AZ31X-h material exposed unstressed for periods up to 960 days in a marine atmosphere was small; the tensile strength was 95 percent or more of that of the original material. Elongations, however, were reduced to 25 to 50 percent of those of the unexposed material.

The maximum exposure periods for the AZ61X-h and AZ51X-h materials, exposed unstressed at this Bureau, were 50 days and 22 days, respectively. Ultimate tensile strengths were 94 percent or more of those of the unexposed materials; elongations of the AZ61X-h and AZ51X-h materials were 70 percent or more and 85 percent or more, respectively, of those of the unexposed materials.

The data indicate that: (1) the M1-clad AZ31X-h alloy is relatively immune to stress corrosion at stresses of about 30,000 lb/in.² (2) The ZK-60 and the M1-h allovs are relatively immune to stress corrosion at stresses below 20,000 and 16,000 lb/in.², respectively. The tensile properties of the M1 alloy were lower than those of other alloys studied. Hence it was exposed only at a stress of 16,000 lb/in.² on longitudinal specimens. The AZ51X, AZ61X, and AZ80X alloys were all susceptible to stress corrosion in the weather when subjected to stresses of 20,000 lb/in.² These alloys were not exposed in the weather at lower stresses. (3) The AZ31X-h alloy was susceptible. to stress-corrosion in the weather at stresses of 16,000 or more lb/in.²

2. Continuous Immersion Tests in the $N\alpha Cl + K_2 CrO_4$ Solution

The curve in figure 10 shows the relationship between the stress and the exposure period to failure for specimens of annealed AZ31X material. The stresses that the materials would withstand for 24 hr or longer in the NaCl+K₂CrO₄ solution, as determined from curves of this type, are defined as the threshold stresses. It was suggested that such a test could be used as an accelerated test to evaluate the resistance of magnesium alloys to stress corrosion, if a relationship could be established between the threshold values and the ultimate tensile strengths or yield strengths of the material or the long-time stress-corrosion data.

Threshold stresses for the various sheet and extruded alloys, given in tables 2 to 5, inclusive, show that with the exception of 0.064-in. thick AZ61X-h sheet material supplied by the American Magnesium Corp. the threshold stresses were greater than 75 percent of the yield strength of the material. The threshold stresses (in terms of yield strengths) of the annealed AZ61X alloy were generally higher than those of the hardened (rolled) material. The threshold stresses of other materials studied, in percentages of the yield strength, were generally approximately the same whether the material was in the annealed or rolled conditions.

The threshold stresses obtained in the NaCl+ K_2CrO_4 solution do not indicate the minimum



FIGURE 10. Stress-exposure behavior of AZ31X-a specimens, immersed continuously in a NaCl+K₂CrO₄ solution.

○, Specimen failed; ●, specimen did not fail.

stresses at which the material will fail by stresscorrosion cracking in the weather. Furthermore, relative susceptibilities of some of the alloys, as determined from threshold stresses given in tables 2 to 5 do not agree with weather exposure data shown in figures 8 and 9.

3. Intermittent Immersion Tests in the 0.01-Percent NaCl Solution

The results of intermittent immersion stresscorrosion tests on the AZ61X, AZ51X, and AZ31X alloys (all in the rolled condition) are shown graphically in figure 8 (left). For comparison, results of weather exposure tests on these alloys are shown at the right in figure 8. The resistance to stress corrosion of the three materials was in the same order in the intermittent immersion tests in the 0.01-percent NaCl solution as in the weather exposure tests.

The results of intermittent immersion tests in the 0.01-percent NaCl solution for the AZ80X alloy after heat treatment and aging and for the alloy as extruded are shown graphically in figure 9. The results are in qualitative agreement with those obtained in weather exposure tests for this alloy in the two conditions.

There were insufficient data from intermittent immersion tests on the other materials studied to determine whether these results show close correlation with weather exposure data.

4. Discussion of Laboratory Accelerated Tests

It was pointed out in section III, 2, that continuous immersion tests in the NaCl+K₂CrO₄ solution have two drawbacks. First, threshold stresses (stresses below which stress-corrosion fractures did not occur in the NaCl+K₂CrO₄ solution) were much higher than those obtained for the same alloy in the weather. Second and more important, the correlation was poor between predictions of relative susceptibilities of various magnesium alloys, made from the results of this test and the results of weather exposure data.

There was general agreement between the results of intermittent immersion and weather exposure tests on the AZ31X-h, AZ51X-h, and AZ61X-h alloys as was indicated in section III, 3. These results, and the limited results obtained on other materials, indicate that intermittent immersion tests in a dilute chloride solution are a better

laboratory means of predicting the relative susceptibilities of magnesium alloys to stress corrosion in the atmosphere than is continuous immersion in a sodium chloride-potassium chromate solution.

IV. Summary

[•] 1. Stress-corrosion tests were made on M1, bare and clad AZ31X, AZ51X, AZ61X, AZ80X, and ZK-60 wrought magnesium base alloys. Stressexposure period to failure results were obtained both in weather exposure and in two types of laboratory tests.

2. The decrease in tensile properties of unstressed specimens exposed at the National Bureau of Standards site was small.

3. The M1-clad AZ31X-h alloy was resistant to stress corrosion in a marine atmosphere at stresses up to 30,000 lb/in.²

4. The bare AZ31X alloy was susceptible to stress corrosion in the weather at stresses of 16,000 lb/in.² or more.

5. The sheet M1 and the extruded Dow ZK-60 alloys were resistant to stress corrosion in weather exposure tests at stresses up to 16,000 and 20,000 lb/in.², respectively.

6. The AZ51X, AZ61X, and AZ80X alloys were all susceptible to stress corrosion in the weather when subjected to stresses of 20,000 lb/in.² These alloys were not tested at lower stresses.

7. The susceptibility of magnesium base alloys to stress corrosion increased with the aluminum content up to approximately 6.5-percent aluminum.

8. Results of intermittent immersion laboratory tests, using a 0.01-percent NaCl solution as a corrodent, were in better agreement with those obtained in weather exposure than were those obtained by continuous immersion in a 3.5-percent NaCl+2.0-percent K_2CrO_4 solution.

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