## Standard X-ray Diffraction

## Powder Patterns

UNITED STATES DEPARTMENT OF COMMERCE

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Howard E. Swanson and Eleanor Tatge



# National Bureau of Standards Circular 539 

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## Contents

Page

1. Introduction ..... 1
2. Patterns ..... 3
3. 4. Nagnesium ..... 10
1. 2. Aluminum ..... 11
1. 3. Nickel ..... 13
1. 4. Copper ..... 15
2.5. Zinc ..... 16
2.6. Germanium ..... 18
2.7. Mol ybdenum ..... 20
2.8. Palladium- ..... 21
2.9. Silver ..... 23
2.10. Tin, beta ..... 24
1. 11. Tellurium ..... 26
1. 12. Tungsten ..... 28
2.13. Tantalum ..... 29
2.14. Platinum ..... 31
1. 15. Gold ..... 33
1. 16. Lead ..... 34
2.17. Beryllium oxide, BeO (bromellite) ..... 36
1. 18. Magnesium axide, Mgo (periclase) ..... 37
1. 19. Low or $\alpha$-cristobalite, $\mathrm{SiO}_{2}$ ..... 39
2.20. High or $F$-cristobalite, $\mathrm{SiO}_{2}$ ..... 42
2.21. Calcium oxide, Ca () ..... 43
1. 22. Titanium dioxide (rutile), $\mathrm{TiO}_{2} \ldots \ldots$. ..... 44
1. 23. Titaniurn dioxide (anatase), $\mathrm{TiO}_{2} \ldots \ldots$ ..... 46
2.24. Nickelous oxide, NiO (bunsenite) ..... 47
2.25. Cupric oxide, CuO (tenorile) ..... 49
2.26. Germanium dioxide, $\mathrm{Ge} \mathrm{O}_{2}$ ..... 51
1. 27. Arsenic trioxide, $\mathrm{As}_{2} \mathrm{O}_{3}$ (arsenolite) _ ..... 51
1. 28. Selenium dioxide, $\mathrm{SeO}_{2}$ (selenolite) ..... 53
1. $2^{9}$. Stannic oxide, $\mathrm{SnO}_{2}$ (cassiterite) ..... 54
2.30. Ceric oxide, $\mathrm{CeO}_{2}-$ ..... 56
2. Patterns-Continued Page
3. 31. Thorium oxide, $\mathrm{ThO}_{2}$ (thorianite) ..... 57
2.32. Calcium hydroxide, $\mathrm{Ca}(\mathrm{OH})_{2}$ (port- landite) ..... 58
2.33. Ammonium chloride, $\mathrm{NH}_{4} \mathrm{Cl}$ (sal- ammoniac) ..... 59
1. 34. Lithium fluoride, LiF ..... 61
2.35. Lithium chloride, LiCl ..... 62
2.36. Sodium fluoride, NaF (villiaumite) ..... 63
2.37. Potassium fluoride, KF ..... 64
2.38. Potassium chloride, KCl (sylvite) ..... 65
1. 39. Potassium bromide, KPr ..... 66
1. 40. Potassium iodide, KI ..... 68
2.41. Calcium fluoride, $\mathrm{CaF}_{2}$ (fluorite) ..... 69
2.42. Barium fluoride, $\mathrm{FaF}_{2^{-}}$ ..... 70
1. 4.. Mercurous chloride, $\mathrm{Hg}_{2} \mathrm{Cl}_{2}$ (calome1) ..... 72
2.44. Nercuric chloride, $\mathrm{HeCl}_{2}-$ ..... 73
2. 45. Mercuric iodide, HgI ..... 74.
1. 46. Lead fluochloride, FlFC1 (matlock- ite) ..... 76
2.47. Potassium cyanide, KCN ..... 77
1. 48. Sodium cyanide (cubic), NaCN ..... 78
2.49. Sodium cyanide (orthorhombic), NaCN ..... 79
2.50. Strontium nitrate, $\mathrm{Sr}\left(\mathrm{NO}_{3}\right)_{2}$ ..... 80
2.51. Barium nitrate, $\mathrm{Ba}\left(\mathrm{NO}_{3}\right)_{2}$ (nitrobarite) ..... 81
2.52. Zinc horate, $\mathrm{ZnF}_{2} \mathrm{O}_{4}$ ..... 83
2.53. Magnesium silicate, $\mathrm{Nig}_{2} \mathrm{SiO}_{4}$ (forster- ite) ..... 93
2.54. Nagnesium tungstate, $\mathrm{MgHO}_{4}$ ..... 84
1. Keferences ..... 87
2. Cumulative index to volumes I and

II ..... 95

# STANDARD X-RAY DIFFRACTION POWDER PATTERNS Vol. I-Data for 54 Inorganic Substances 

Lioward E. Swanson and Eleanor Tatge

In continuation of the National Bureau of Standards project for improving the file of X-ray diffraction patterns published by the American Society for Testing Materials, sets of patterns in the file, each representing a different chemical, have been reviewed with the object of supplanting them with single standard patterns. Reports are made on substances for each of which a pattern prepared at the Rureau is offered to replace a set now in the file. Four additional reports are included, one for high cristobalite, for which no pattern was prepared at the NBS, and three, which are not represented in the ASTM file, on selenium dioxide, zinc borate, and magnesium tungstate.

The substances reported upon are $\mathrm{Mg}, \mathrm{Al}, \mathrm{Ni}, \mathrm{Cu}, \mathrm{Zn}, \mathrm{Ge}, \mathrm{Mo}, \mathrm{Pd}, \mathrm{Ag}, \mathrm{Sn}, \mathrm{Te}, \mathrm{W}$, $\mathrm{Ta}, \mathrm{Pt}, \mathrm{Au}, \mathrm{Pb}, \mathrm{BeO}, \mathrm{MgO}, \mathrm{SiO}_{2}$ (low cristobalite), $\mathrm{SiO}_{2}$ (high cristobalite), $\mathrm{CaO}, \mathrm{TiO} 2$ (rutile), $\mathrm{TiO}_{2}$ (anatase), $\mathrm{NiO}, \mathrm{CuO}, \mathrm{GeO}_{2}, \mathrm{As}_{2} \mathrm{O}_{3}, \mathrm{SeO}_{2}, \mathrm{SnO}_{2}, \mathrm{CeO}_{2}, \mathrm{ThO}_{2}, \mathrm{Ca}\left(\mathrm{OH}\right.$ ), $\mathrm{NH}_{4} \mathrm{Cl}$, $\mathrm{LiF}, \mathrm{LiCl}, \mathrm{NaF}, \mathrm{KF}, \mathrm{KCl}, \mathrm{KBr}, \mathrm{KI}, \mathrm{CaF}_{2}, \mathrm{HaF}_{2}, \mathrm{Hg}_{2} \mathrm{Cl}_{2}, \mathrm{HgCl}_{2}, \mathrm{HgI}, \mathrm{PbFCl}, \mathrm{KCN}, \mathrm{NaCN}$ (cubic), NaCN (orthorhombic), $\mathrm{Sr}\left(\mathrm{NO}_{3}\right)_{2}, \mathrm{Ba}\left(\mathrm{NO}_{3}\right)_{2}, \mathrm{ZnF}_{2} \mathrm{O}_{4}, \mathrm{Mg}_{2} \mathrm{SiO}_{4}$, and MgWO 4 .

The ASTM patterns are tabulated for comparison with additional patterns from the literature and one prepared at the NBS. Miller indices are derived from the calculation of spacings by desk calculator or the electronic computer SEAC. Interplanar spacings in angstroms (except where otherwise noted) and relative intensities from 1 to 100 are tabulated. For the NBS pattern the three strongest lines are given, as well as the lattice constants and the computed density. The index of refraction of the sample is noted if it could be determined. Crystal-structure data from the literature are noted.

## 1. Introduction

Three hundred or more substances are represented in the X -ray Diffraction Pattern card file [l] of the American Society for Testing llaterials by more than one pattern each, many of the patterns differing materially from each other. Upon the recommendation of the Joint Committee on Chemical Analysis by X-ray Diffraction Methods, ${ }^{2}$ a critical examination of repeated patterns in the card file is being undertaken as part of a program for the general improvement of the file. Patterns made recently for 53 substances at the National Bureau of Standards are presented, compared with those in the file and in the literature, and recommended for adoption as standard fatterns, 50 of them to replace 170 patterns now in the file, and three of them ( $\mathrm{Sen}_{2}, \mathrm{ZnB}_{2} \mathrm{O}_{4}$, and $\mathrm{MgWn}_{4}$ ) offered as additions

[^0]to the file. Also, the patterns for $\beta$ cristobalite ( $\mathrm{SiO}_{2}$ ), for which no NBS pattern was prepared, are discussed, and one in the ASTM file is recommended as a standard. The eight patterns given in an earlier paper [218], in which the technique used in the NBS laboratory was outlined, are included in this paper with some slight revisions. A complete list of the patterns reported is given on page 3.

Priefly, for preparing the NBS pattern, a Geiger-counter spectrometer with a 160 degree arc was used, which permits recording the patterns into the back reflection region. Copper Ka x-radiation with a wavelength of 1.5405 A was considered most satisfactory for general use. Separate charts were made for interplanar spacing and for intensity measurements so that the flat sample surface desirable for the former did not preclude the disorientation of particles necessary for the latter. Actual peak height from background was used for intensity measurements. Samples
used were sufficiently fine-grained, usually less than 25 microns, to give reproducible results. The spacings for all NBS standard patterns were corrected by an internal standard of tungsten, except that for the tungsten pattern, silver was used. The unit cell used for tungsten calibration was 3.1648 A and for silver calibration, 4.0861 A , both at $25^{\circ} \mathrm{C}$ [119]. Lines occasionally hidden by tungsten lines were obtained from the intensity diagrams. All spacing errors inherent in sample mountings, sample density variations, spectrometer alinement, and recorder lag were easily compensated by the use of the internal standard. The samples used were of high chemical purity, and chemical or spectrographic analyses are given (rarely both). Phase purity was checked microscopically where possible. The temperature was allowed to vary not more than $\pm 1^{\circ} \mathrm{C}$ from that recorded in the respective tables.

The diffraction lines were indexed and the lattice constants determined. For the cubic substances the indexing was done by comparing theoretical spacings calculated on a desk calculator. The electronic computer SEAC (National Bureau of Standards Eastern Automatic Computer), used under the direction of Dr. Fred Ordway, proved a time-saving device for computing the spacings for many of the substances.

So far as possible the interplanar spacing data reproduced in the tables were reduced to angstrom units as internationally agreed upon in 1946 [264]. In some cases the data are known to be, or assumed to be, in $k X$ units, which are less than angstroms by a factor of 1.00202 [264], and thus easily converted. In others, Bragg angle data are computed to obtain spacings directly in angstroms; or the wavelength of the radiation used in preparing the pattern is compared with the actual wavelength in angstroms to obtain a conversion factor. The dates heading
the column are those of the first publication of the basic data.

With regard to other units, the values of the coefficient of linear thermal expansion given are those which could be readily located in the literature and which cover a satisfactory temperature range. The density is given in grams per cubic centimeter. Indices of refraction for the NBS materials were obtained in white light with oils standardized by sodium light; indices quoted from the literature are accompanied by subscript $D$ for the sodium $D$ line, or $L i$ for lithium.

Lattice constants from the literature are presentedin order of date in unnumbered tables in the text for comparison with those obtained from the NBS data. They are tabulated in angstrom units and are given at $25^{\circ} \mathrm{C}$, the most commonly stated temperature, if a coefficient of expansion is available from the literature for use in recalculating. This results in an occasional difference between the NBS lattice constant given in the text (at $25^{\circ} \mathrm{C}$ ) and that appearing with the tabulated pattern (at the experimental temperature, usually $26^{\circ} \mathrm{C}$, noted at the head of the column).

The cumulative maximum error of interplanar spacing and lattice constant measurements on NBS patterns varies not more than $\pm 5$ in the last significant figure recorded. In most cases the last significant figure for the density depends rather on the precision to which atomic weights are known than on that of the lattice constant, which is usuually greater.

The original sources are noted of all patterns, tabulated lattice constants, and, where practicable, structure data. For some of the simpler structures it was not always easy to ascertain who was first responsible for the determination of the structure of the compound in question and a reference could not be given with certainty.

## 2. Patterns

The duplicate patterns considered here are listed in table l. In the table the file card numbers are given for both the old (194041) and new (1950) files of the ASTM, followed by the index numbers (interplanar spacings for the three strongest lines).

Tables 2 to 55 list Miller indices, interplanar spacings, and relative-intensity measurements for the substances considered. Also, in the case of cubic materials, the lattice constants calculated from each spacing are given and averaged at the bottom of the table. The text preceding each table furnishes the following information: Origin of the patterns (such as ASTM cards, or literature); source of the NBS material, its chemistry, and its treatment preliminary to preparing the pattern; basis for converting the spacings of each pattern to angstrom units; the three strongest lines of the NBS patternthe lines used for indexing the ASTM cards; crystal structure data, such as the type of lattice, space group, and the number of molecules in the unit cell; the lattice constant determined from the NBS pattern, compared with constants obtained from the literature; the density, calculated from the NBS lattice constant; and the index of refraction, if it could be determined on the NBS material, or was easily available for other pure material from the literature.

A complete list of the patterns given is magnesium (hexagonal); aluminum (cubic); nickel (cubic); copper (cubic); zinc (hexagonal); germanium (cubic); molybdenum (cubic); palladium (cubic); silver (cubic); tin-white or $F$ (tetragonal); tellurium (hexagonal); tungsten (cubic); tantalum (cubic); platinum (cubic); gold (cubic); lead (cubic); BeO-bromellite (hexagonal); MgO -periclase (cubic); $\mathrm{SiO}_{2}$-low or a cristobalite (tetragonal); $\mathrm{SiO}_{2}$-high or E cristobalite (cubic); $\mathrm{CaO}-\mathrm{lime}$ (cubic); $\mathrm{TiO}_{2}$-rutile (tetragonal); $\mathrm{TiO}_{2}$-anatase (tetragonal); NiO -bunsenite (cubic); CuO -tenorite (monoclinic); $\mathrm{GeO}_{2}$ (hexagonal); $\mathrm{As}_{2} \mathrm{O}_{3}$ arsenolite (cubic); $\mathrm{SeO}_{2}$-selenolite (tetrag-
onal); $\mathrm{SnO}_{2}$-cassiterite (tetragonal); $\mathrm{CeO}_{2}$ (culic); $\mathrm{ThO}_{2}$ - thorianite (cubic); $\mathrm{Ca}\left(\mathrm{CHi}_{2}\right.$ portlandite (hexagonal); $\mathrm{NH}_{4} \mathrm{Cl}$-salammoniac (cubic); LiF (cubic); LiCl (cubic); NaF-villiaumite (cubic); KF (cubic); KCl-sylvite (cubic); KBr (cubic); KI (cubic); $\mathrm{CaF}_{2}$-fluorite (cubic); $\mathrm{BaF}_{2}$ (cubic); $\mathrm{Hg}_{2} \mathrm{Cl}_{2}$-calomel (tetragonal); $\mathrm{HgCl}_{2}$ (orthorhombic) $\mathrm{HgI}_{2}$ (tetragonal) ; $\mathrm{PbFCl}-$ matlockite (tetragonal); KCN (cubic); NaCN (cubic); NaCN (orthorhombic); $\mathrm{Sr}\left(\mathrm{NO}_{3}\right)_{2}$ (cubic); $\mathrm{Ba}\left(\mathrm{NO}_{3}\right)_{2}$-nitrobarite (cubic); $\mathrm{ZnB}_{2} \mathrm{O}_{4}$ (cubic); $\mathrm{ME}_{2} \mathrm{SiO}_{4}$-forsterite (orthorhombic); and $\mathrm{MgWO}_{4}$ (monoclinic).

Table 1. ASTM card to be superseded

| Card number |  | Index lines |  | Source |
| :---: | :---: | :---: | :---: | :---: |
| old | 1950 | Old | 1950 |  |
| 1. Magnesium |  |  |  |  |
| 2942 | 3124 | 2.44 | 2.44 | [100] |
|  | 1-1151 | 2.75 | 2.75 |  |
|  | 1-1148 | 1.61 | 1.61 |  |
| 2924 | 3080 | 2.45 | 2.45 | [85] |
|  | 1-1135 | 2.77 | 2.77 |  |
|  | 1-1141 | 2.60 | 2.60 |  |
| 2. Aluminum |  |  |  |  |
| 3060 | 3224 | 2.33 | 2.33 | [102] |
|  | 1-1180 | 2.025 | 2.03 |  |
|  | 1-1179 | 1.21 | 1.21 |  |
| 3049 | 3242 | 2.34 | 2.34 | [59] |
|  | 1-1186 | 1.221 | 1.22 |  |
|  | 1-1176 | 2.02 | 2.02 |  |
| ------- | 3223 | ----- | 2.32 | [170] |
|  | 3-0938 | -.-.- | 2.03 |  |
|  | 3-0932 | -- | 1.43 |  |
| 3061 | 3225 | 2.33 | 2.33 | [85] |
|  | 1-1181 | 2.02 | 2.02 |  |
|  | 1-1180 | 1.430 | 1.43 |  |
| II-2503 | 3243 | 2.33 | 2.33 | Crystallographic Labo ratory, Cambridge. |
|  | 2-1117 | 1.22 | 1.22 |  |
|  | 2-1109 | 2.02 | 2.02 |  |
| 3. Nickel |  |  |  |  |
| 3462 | 3645 | 1.95 | 1.95 | [103] |
|  | 1-1272 | 1.13 | 1.13 |  |
|  | 1-1272 | 0.74 | 0.74 |  |
| 3362 | 3595 | 2.038 | 2.04 | [104] |
|  | 1-1270 | 1.067 | 1.07 |  |
|  | 1-1258 | 1.766 | 1.77 |  |

Table 1. ASTM cards to be superseded-Con.


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| 20. Silicon dioxide ( $\beta$-cristobalite) |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: |
| 956 | 0964 | 4.14 | 4.14 | [254] |
|  | 1-0430 | 2.53 | 2.53 |  |
|  | 1-0424 | 1.639 | 1.64 |  |
| ---- | 0965 | ----- | 4.14 | [8] |
|  | 3-0259 | ----- | 2.52 |  |
|  | 3-0257 | ----- | 1.64 |  |
| II-588 | 0963 | 4.14 | 4.14 | [ 255,8 ] |
|  | 2-0276 | 2.53 | 2.53 |  |
|  | 2-0278 | 1.64 | 1.64 |  |
| 21. Calcium oxide |  |  |  |  |
| II-244I | 3140 | 2.40 | 2.40 | United Steel Compa- |
|  | 2-1079 | 1.70 | 1.70 | nies, England, and |
|  | 2-1088 | 1.45 | 1.45 | [89; 40]. |

Table 1. ASTM cards to be superseded-Con.

| Card number |  | Index lines |  | Source |
| :---: | :---: | :---: | :---: | :---: |
| O1d | 1950 | Old | 1950 |  |
| 21. Calcium oxide-Con. |  |  |  |  |
| 2989 | $\begin{array}{r} 3183 \\ 1-1172 \\ 1-1160 \\ 3784 \\ 3-1127 \\ 3-1123 \end{array}$ |  | $\begin{aligned} & 2.39 \\ & 1.69 \\ & 2.76 \\ & 1.69 \\ & 2.39 \\ & 0.98 \end{aligned}$ | $[85]$ $[48]$ |
| 22. Titanium dioxide (rutile) |  |  |  |  |
| 3653 | $\begin{array}{r} 3774 \\ 1-1292 \\ 1-1292 \\ 1774 \\ 2-0526 \\ 2-0494 \\ \\ 3773 \\ 3-1124 \\ 3-1122 \end{array}$ | $\begin{aligned} & 1.69 \\ & 3.24 \\ & 2.49 \\ & 3.24 \\ & 1.68 \\ & 1.36 \\ & \\ & 1.69 \\ & 3.25 \\ & 1.36 \end{aligned}$ | $\begin{aligned} & 1.69 \\ & 3.24 \\ & 2.49 \\ & 3.24 \\ & 1.68 \\ & 1.36 \\ & \\ & 1.69 \\ & 3.25 \\ & 1.36 \end{aligned}$ | [85] <br> British Museum, Crystallographic Lab., Cambridge, and [19, 125,246 ]. <br> United Steel Companies, England. |
| 23. Titanium dioxide (anatase) |  |  |  |  |
| II-911 | $\begin{array}{r} 4111 \\ 3-1332 \\ 3-1332 \\ 1390 \\ 2-0411 \\ 2-0406 \\ 1324 \\ 1-0572 \\ 1-0562 \\ \\ 1323 \\ 2-0391 \\ 2-0387 \end{array}$ | $--\cdots-$ .--- 3.47 1.88 1.69 3.52 1.88 1.70 3.51 1.89 1.70 | ----- $\cdots----$ 3.47 1.88 1.69 3.52 1.88 1.70 3.51 1.89 1.70 | [233] <br> British Museum, Crystallographic Lab., Cambridge, and [246]. [85] <br> United Steel Companies, England. |
| 24. Nickelous oxide |  |  |  |  |
| II-2809 | $\begin{array}{r} 3516 \\ 2-1238 \\ 2-1216 \\ 4066 \\ 3-1287 \\ 3-1287 \\ 3471 \\ 1-1238 \\ 1-1238 \end{array}$ | $\begin{aligned} & 2.09 \\ & 1.48 \\ & 2.41 \\ & ------- \\ & ----- \\ & 2.08 \\ & 2.40 \\ & 1.474 \end{aligned}$ | 2.09 <br> 1. 48 <br> 2.41 $\qquad$ $\qquad$ $\qquad$ <br> 2.08 <br> 2.40 <br> 1. 47 | United Steel Companies, England, and [139; 93; 134] [46] <br> [85] |
| 25. Cupric oxide |  |  |  |  |
| - | $\begin{array}{r} 3014 \\ 3-0867 \\ 3-0867 \end{array}$ | ---------- | 2.52 2.31 1.04 | [165] |

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Table 1. ASTM cards to be superseded-Con.

| Card number |  | Index lines |  | Source |
| :---: | :---: | :---: | :---: | :---: |
| Old | 1950 | Old | 1950 |  |
| 42. Barium fluoride |  |  |  |  |
| II-2633 | 3305 | 2.20 | 2.20 | [221] |
|  | 2-1147 | 1.87 | 1.87 |  |
|  | 2-1157 | 1.43 | 1.43 |  |
| 1346 | 1311 | 3.58 | 3.58 | [85] |
|  | 1-0563 | 2.19 | 2.19 |  |
|  | 1-0533 | 1.86 | 1.86 |  |
| 43. Mercurous chloride |  |  |  |  |
| 1869 | 1792 | 3.155 | 3.16 | [90] |
|  | 1-0757 | 4.143 | 4.14 |  |
|  | 1-0768 | 1.962 | 1.96 |  |
|  | 1842 | ----- | 3.17 | [112] |
|  | 3-0522 | -- | 1.96 |  |
|  | 3-0516 | -- | 1.04 |  |
| II-1222 | 1921 | 3.13 | 3.13 | [196] |
|  | 2-0574 | 1.94 | 1.94 |  |
|  | 2-0560 | 4.05 | 4.05 |  |
| 945 | 0959 | 4.16 | 4.16 | [85] |
|  | 1-0426 | 3.17 | 3.17 |  |
|  | 1-0420 | 1.97 | 1.97 |  |
| 44. Mercuric chloride |  |  |  |  |
| 812 | 0852 | 4.35 | 4.35 | [85] |
|  | 1-0377 | 3.00 | 3.00 |  |
|  | 1-0365 | 2.70 | 2.70 |  |
| II-524 | 0826 | 4.34 | 4.34 | [26] |
|  | 2-0249 | 4.08 | 4.08 |  |
|  | 2-0255 | 3.36 | 3.36 |  |
| 45. Mercuric iodide |  |  |  |  |
| 3204 | 3336 | 2. 183 | 2.18 | [91] |
|  | 1-1216 | 3.563 | 3.56 |  |
|  | 1-1217 | 6.192 | 6.19 |  |
|  | 4060 | ----- |  | [99] |
| 1362 | 3-1281 | ----- | ----- |  |
|  | 3-1281 | ----- | ----- |  |
|  | 1312 | 3.56 | 3.56 | [85] |
|  | 1-0564 | 2.18 | 2.18 |  |
|  | 1-0542 | 4.11 | 4.11 |  |
| 46. Lead fluochloride |  |  |  |  |
| II-856 | 1310 | 3.54 | 3.54 | British Museum. |
|  | 2-0388 | 2.25 | 2.25 |  |
|  | 2-0377 | 1.77 | 1.77 |  |
|  | 3928 | ----- | 1.22 | [164] |
|  | 3-1184 | ----- | 1.29 |  |
|  | 3-1182 |  | 1.79 |  |

Table 1. ASTM cards to be superseded-Con.


### 2.1. Magnesium (Hexagonal)

In addition to the two patterns recorded in the ASTM file (see table l) four were found in the literature: 1920, Bohlin [18]; 1923, Owen and Preston [176]; 1929, Grime and Morris-Jones [84]; and 1933, Finch and Quarrell. [69]. These are compared in table 2 with a pattern prepared at the NBS.

The magnesium sample used for the NBS pattern was obtained from the Dow Chemical Co. Spectrographic analysis at the NBS indicates the presence of calcium <0.01 percent, and traces of $\mathrm{Al}, \mathrm{Cu}, \mathrm{Fe}$, and Si .

In table 2 the data of Hull and Bohlin were derived directly in angstrom units from the published Bragg angle data. The electron diffraction measurements of Finch and Quarrell

Table 2. Magnesium (hexagonal)

are presented as published. Measurements by the remaining workers were converted to angstroms from $k X$ units. The two patterns published by full gave the same interplanar spacings; only the one reproduced on the ASTM card is given in the table. The Bohlin pattern shows two lines for approximately the spacing requi red for 101 , one of which is extraneous to the structure. Bohlin also found lines at 001 and 003 not found in any other pattern, and at lll, occurring only in the electron diffraction pattern of Finch and Quarrell. It is surprising that neither Hull, nor Owen and Preston reported the strong 002 line.

Two patterns were published by Hull [100]; they are much alike, and only one appears in the ASTM file and in table 2. The three lines 104,300 , and 205 recorded in table 2 as $<1$, were miscalculated in converting them for the ASTM card as 3 rather than as 0.3 . The intensities of Hanawalt, Rinn, and Frevel and of the NBS compare closely except that the 002 line is almost twice as strong for the pattern of the latter. The NBS sample, in minute spheres formed upon atomizing the material, was particularly satisfactory for this determination as particle orientation was not possible.

The structure of magnesium, which is hexagonal close-packed, was worked out by Hull [103] in 1917. The space group is $D_{6 h}^{4}$ ( $\mathrm{C} 6 / \mathrm{mmc}$ ); there are two atoms in the unit cell.

Some recent unit cell determinations, after the addition of corrections for temperature and conversion to angstrom units, are tabulated:

Unit cell at $25^{\circ} \mathrm{C}$, angstroms

|  |  | a | c |
| :---: | :---: | :---: | :---: |
| 1932 | Stenzel and Weertz [212] | 3.2091 | 5.2104 |
| 1935 | Jette and Foote [119] | 3.2095 | 5.2107 |
| 1935 | Owen, Pickup, and Roberts [175].- | 3.2091 | 5.2115 |
| 1938 | Ievinš, Straumanis, and Karlsons [114] | 3.20927 | 5.21033 |
| 1939 | Raynor and Hume-Rothery [191]. | 3.20948 | 5.2113 |
| 1940 | Foote and Jette [71, 72$]$ | 3.2095 | 5.2107 |
| 1942 | Raynor [190] | 3.20949 | 5.21096 |
| 1953 | Swanson and Tatge.-..---.-.---.--- | 3. 2094 | 5.2103 |

The 1939 determinations of Raynor and HumeRothery [191] of the coefficients of expansion were used in correcting for temperature: $27.9 \times 10^{-6}$ parallel to $c$, and $27 \times 10^{-6}$ parallel to $a$. The density based on the NBS unit-cell determination is 1.737 .

### 2.2. Aluminum (Cubic)

Five patterns of aluminum recorded on ASTM cards (see table l) are compared in table 3 with a pattern prepared at the NBS and one by Scherrer [202] obtained from the literature. The material used for the NBS sample was a melting-point Standard Sample of aluminum prepared in the chemistry laboratories of the Bureau. The chemical analysis (in percent) is $\mathrm{Si}, 0.011 ; \mathrm{Cu}, .006 ; \mathrm{Fe}, .007$; $\mathrm{Ti}, .0001 ; \mathrm{Zr}, .003 ; \mathrm{Ga}, .004 ; \mathrm{Mo}, .00002 ; \mathrm{S}$, .0001; Al, $99.9+$ (by difference).

The intensity measurements on the ASTM card accompanying the spacings ascribed to the Crystallographic Laboratory correspond to those of the 1925 Davey pattern except for one line, and were probably supplied from that source; the 400 , a very weak line, has the intensity giverı as 80 , surely a misprint for the 40 of the Davey pattern. This set of intensity measurements is the only one showing the 311 line stronger than the 200 . All patterns show the 111 line as the strongest. The order shown by the NBS pattern is 111,200 , and 311 as first, second, and third strongest lines.

Aluminum has a face-centered cubic lattice [102], four atoms to the unit cell, and the space group $0_{h}^{5}$ ( $\mathrm{Fm}_{\mathrm{m}} 3 \mathrm{~m}$ ). Unit cell values from the literature are compared below with the NBS determination.

Unit cell, angstroms at $25^{\circ} \mathrm{C}$

|  |  | $a$ |
| :---: | :---: | :---: |
| 1933 | Owen and Yates [178] | 4.0495 |
| 1935 | Jette and Foote [119] | 4.0496 |
| 1936 | Jevins and Straumanis [121] | 4.04961 |
| 1936 | Straumanis and Ievinš [216] | 4.0489 |
| 1940 | Foote and Jette [71] | 4.0496 |
| 1941 | Lu and Chang [ 141 | 4.0498 |
| 1941 | Van Bergen [230] | 4.04955 |
| 1948 | Axon and Hume-Rothery | 4.0495 |
| 1953 | Swanson and Tatge....... | 4.0494 |

These were accompanied by temperature data, and by means of a coefficient of expansion of $23.84 \times 10^{-6}$ [4, 67, 127, 230] were converted to angstroms at $25^{\circ} \mathrm{C}$. Using the

NBS lattice constant, the density was calculated as 2.697 at $25^{\circ} \mathrm{C}$.

For table 3, the spacings of three of the five ASTM patterns were converted to angstroms from the $k X$ units in which they were given. The Olshausen interplanar spacings were calculated for the table directly from the measurements given of the Bragg angle. Hull used 0.712 as the wavelength for molyb-
denum radiation; his spacings were converted to angstroms to correspond with a wavelength of 0.709 A . The pattern by Scherrer, a slight improvement on that of Hull made the year before, was calculated for table 3 directly in angstroms from Bragg angle data. Agreement among the patterns on spacings is excellent, as demonstrated by the uni form unit-cell values shown at the bottom of the table.

Table 3. Aluminnum (cubic)


### 2.3. Nickel (Cubic)

An unusually large number of patterns has been published for nickel. Table 4 is based on 15 patterns, 6 of which are in the ASTM card index file of $X$-ray diffraction patterns (see table 1), nine additional patterns found in the literature, and the proposed standard pattern made at the NBS. The literature sources are 1917, Hull [103]; 1920, Bohlin [18]; 1922, Wever [247] ; 1925, Levi and Tacchini [139]; 1925, Clark, Asbury, and Wick [49]; 1926, Holgersson [95]; 1928, Roux and Cournot [195]; 1929, Greenwood [83]; 1939, Boochs [20].

With regard to the ASTM cards, the earliest pattern recorded, the 1917 pattern of Hull [103], was retracted in 1921 [106], and is replaced in table 4 by a second pattern from the same 1917 publication, not in the ASTM file. Hull's 1921 pattern was published twice in successive articles in the same journal $[104,106]$. The ASTM card ascribes it to the second of these. Jung's published pattern [123] comprises four lines, of which the fourth is omitted from the ASTM card, probably due to its lack of precision.

The sample of nickel used for the NBS pattern was prepared by the Johnson, Mat they \& Co., Ltd., laboratories of London, England; it is numbered 3236 . Their spectrographic analysis (in percent) showed as impurities Mg , <0.01; Si, <0.01; Ca, <0.01.

Table 4 compares the interplanar spacings, intensity measurements and unit cell dimensions of the 15 patterns. Nine of the investigators published $A$ or $\sin \theta$ values rather than interplanar spacings. The spacings were calculated for the table directly in angstrom units. The spacings of the Davey, the Hanawalt, Rinn, and Frevel, and the Boochs patterns were converted to angstrom from $k X$ units. The two Hull patterns, one made with tungsten radiation with a wavelength given as 0.212 , the other with molybdenum (wavelength 0.712 ), and the spacings of the Roux and Cournot pattern giving the wavelength of the molybdenum radi -
ation used as 0.712 , were converted to angstrom units on the basis of the wavelengths used. Wever tabulated the $e$ values for several samples of nickel; these are closely parallel and one, of a sample of high purity, was selected for table 4. Only one of two similar patterns published by Davey was chosen for the table. Mazza and Nasini likewise published several patterns, of which one was selected for the ASTM card and is reproduced here. Comparison of the lattice constants for the lines of each pattern shows that none of the interplanar spacings of the published patterns is accurate to more than two decimal places.

Five of the patterns record intensity measurements numerically. The older ones show the effects of uncorrected absorption and focusing errors. Those of Hanawalt, Rinn, and Frevel and of the NBS, although utilizing different radiations in their preparation, agree in designating the three strongest or index lines as the 111,200 , and 320 .

The common form of nickel discussed here has a face-centered cubic lattice [103], four atoms to the unit cell, and the space group $0_{h}^{5}$ ( Fm 3 m ). About 40 lattice constants were found in the literature, many of these of high precision. Six, accompanied by the necessary data for conversion to angstroms at $25^{\circ} \mathrm{C}$, are tabulated below with the NBS determination. For the conversion the coefficient of expansion of $13.4 \times 10^{-6}$ was used, an average of two recent values [122, 179].

Unit cell in angstroms, $25^{\circ} \mathrm{C}$

| 1931 | Phragmén [185]..------------------------ | 3.5255 |
| :---: | :---: | :---: |
| 1932 | Oven and Yball [173] | 3. 5254 |
| 1934 | Jesse [118] | 3.525 |
| 1935 | Jette and Foote $[71,119,120]$ | 3.5239 |
| 1936 | Owen and Yates [179] | 3.5247 |
| 1941 | $L_{u}$ and Chang [141] | 3.5247 |
| 1941 | Fricke [76 | 3.5239 |
| 1953 | Swanson and Tatge.---------------------- | 3.5238 |

The density, based on the NBS lattice constant, is 8.907 at $25^{\circ} \mathrm{C}$.

Table 4. Nickel (cuozc)


[^1]Table 4. Nackel (cuoic)-Con.

${ }^{\text {a }}$ Average of last three lines.

### 2.4. Copper (Cubic)

The six patterns for copper in the ASTM file (see table l) are supplemented in table 5 by two from the literature, by Sidhu [207] and by Terrey and Wright [219]. The NBS pattern was made with a sample of copper from the metallurgical laboratory of the Bureau. It had been heated in a hydrogen atmosphere at $300^{\circ} \mathrm{C}$. Spectrographic examination at the NBS showed the following impurities from 0.001 to 0.01 percent: $\mathrm{Ag}, \mathrm{Al}, \mathrm{Bi}, \mathrm{Fe}, \mathrm{Si}$, and Zn .

The spacings for the Jung patterns were calculated for table 5 directly in angstroms from the published Bragg angle data. For the other patterns the spacings were converted from kX to angstrom units, except that of Allis-Chalmers (presumably a personal communication), which was left as it appears on the ASTM card, the unit employed not being known. The second of the two Jung patterns appearing on one ASTM card was published with additional $\mathrm{Cu}_{2} \mathrm{O}$ lines which are omitted in the table. For the Jung and the Waldo patterns two columns of intensity measurements are given, the first as originally published, the second as converted to numerical values on
the ASTM card. For most of the patterns the three strongest lines are 111,200 , and 220 .

The lattice was first determined by Bragg [31] in 1914 as face-centered cubic. The space group is $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ) [107], and there are four atoms in the unit cell. Nine unit cell determinations are compared below with that of the NBS. All were converted to angstroms at $25^{\circ} \mathrm{C}$; the coefficient of expansion $16.99 \times 10^{-6}$ [67] was used.

Unit cell in angstroms at $25^{\circ} \mathrm{C}$

| 1933 |  | 3.6155 |
| :---: | :---: | :---: |
| 1933 | Obinata and Wasserman [168] | 3.6155 |
| 1936 | Hume-Rothery, Lewin, and Reynolds [109] | 3.6148 |
| 1939 | Owen and Roberts [177] | 3.6151 |
| 1940 | Foote and Jette [71] | 3.6151 |
| 1941 |  | 3.6149 |
| 1941 | Fricke [76] | 3.615 |
| 1942 | Hume-Rothery and Andrews [108]..------- | 3.6151 |
| 1945 |  | 3.616 |
| 1953 | Swanson and Tatge | 3.6150 |

From the NBS unit-cell determination the density was calculated as 8.932 at $25^{\circ} \mathrm{C}$.

Table 5. Copper (cubic)

${ }^{\text {a }}$ As first published. ${ }^{\mathrm{b}}$ On ASTM card. ${ }^{\mathrm{c}}$ Unit not known. ${ }^{\text {d Average of four lines only. }{ }^{\text {e }} \text { Average of three }}$ lines only.

### 2.5. Zinc (Hexagonal)

Two patterns for zinc recorded in the ASTM file (see table 1) are compared in table 6 with a pattern made at the NBS and with 12 found in the literature. The literature sources are 1925, Peirce, Anderson, and van Dyck [184]; 1926, Freeman, Sillers, and Brandt [73]; 1928, Roux and Cournot [195]; 1929, Osawa and Ogawa [171]; 1929, McLennan and Monkman [148]; 1933,

Finch and Quarrell [69] ; 1935, Kotin and Losada [132]; 1936, Brindley [37]; 1937, Wollan and Harvey [251]; 1937, Miller [154]; 1938, Wroński [252]; 1943, Köhler [128].

The sample of zinc used to obtain the NBS pattern presented here was supplied by the New Jersey Zinc Co., and was numbered 11837. Spectrographic analysis at the Bureau showed a trace of lead and faint traces of copper,
magnesium, and silicon. A lump of zinc sublimed in an evacuated tube yielded a fine powder.

The interplanar spacings of all patterns listed in table 6 are in angstroms, some of them changed from $k X$ units, some computed directly in angstroms from Bragg angle data, and others converted from angstrom units based on old wavelength values. Only experimental data are listed; published spacings computed by some investigators in the course of work on intensity measurements do not appear in the table. Freeman, Sillers, and Brandt in 1926 published a pattern showing the presence of every possible line. The Köhler pattern
of 1943 misses the 006 line, and shows an extraneous line between 114 and 210.

There is general agreement that 101 is the strongest line. The patterns since 1936 give 002 and 100 as second and third strongest, respectively, except for that of Wroński, 1938, which places these in reverse order.

The structure of zinc [106] is based on a hexagonal lattice, space group $\mathrm{L}_{6 \mathrm{~h}}^{4}$ ( $\mathrm{C} 6 / \mathrm{mmc}$ ). There are two atoms to the unit cell. Values, presumed all in $k X$ urits, found in the literature were converted to angstror units for the following table and corrected for terrperature by means of the coefficients of expansion

Table 6. Zinc (hexagonal)

| hkl | 1921 <br> Hull <br> Mo, 0.7093 A |  | 1925 <br> Peirce, Anderson and van Dyck Mo, 0.7093 A |  | 1926 <br> Freeman, Sillers and Brandt Mo, 0.7093 A |  | 1928Roux and <br> Cournot$\mathrm{Cu}, \quad 1.5405 \mathrm{~A}$ |  | 1929 <br> Osawa and Ogawa <br> $\mathrm{Fe}, 1.9360 \mathrm{~A}$ | $1929$ <br> McLennan and Monkman $\mathrm{Cu}, 1.5405 \mathrm{~A}$ | 1933 <br> Finch and <br> Quarre 11 <br> Electron diffraction |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I | ${ }^{\text {d }}$ | I | d | I | d | I | a | a | a |
|  | A |  | A |  | A |  | 4 |  | A | A | A |
| 002 | 2.462 | 30 | 2. 479 | 25 | 2.461 | 30 | 2.489 | vs | 2. 477 |  | 2.56 |
| 100 | 2.284 | 10 | 2.306 | 13 | 2.293 | 20 | 2.356 | w | ------------- |  | 2.32 |
| 101 | 2.069 | 100 | 2.094 | 100 | 2.078 | 100 | 2.130 | vs | 2.099 | 2.078 | 2.12 |
| 102 | 1.678 | 20 | 1.690 | 25 | 1.677 | 15 | 1.742 | 1 | 1.687 | 1.684 | 1.74 |
| 103 | 1.334 | 100 | 1.340 | 44 | 1.335 | 30 | 1.402 | s | 1.343 | 1.339 | 1.39 |
| 110 | 1.327 | 100 |  |  | 1.324 | 30 | 1.304 | w | ------------ | 1.329 | 1.35 |
| 004 | 1.230 | 5 | 1.236 | 2 | 1.231 | 3 | 1.232 | w | ----------- |  | 1.30 |
| 112 | 1.167 | 70 | 1.173 | 25 | 1.166 | 35 | 1.201 | w | 1.174 | 1.169 | 1.19 |
| 200 | 1.148 | 5 | 1.150 | - | 1.146 | 3 | ------- | -.. | ------------- |  | 1.17 |
| 201 | 1.117 | 40 | 1.124 | 19 | 1. 117 | 35 | ------ | -- | 1.123 | 1.119 | 1.13 |
| 104 | 1.084 | 5 | 1.090 | 3 | 1.085 | 1 | ------ | --- | 1.090 | 1.085 |  |
| 202 | 1.040 | 10 | 1.046 | 2 | 1.039 | 1 | ------ |  | 1.046 | 1.043 | 1.06 |
| 203 | 0.943 | 20 | 0.944 | ------ | 0.940 | 3 | ------ | --- | ------------ | 0.943 | 0.96 |
| 105 | ------- |  |  |  | . 905 | 5 |  |  |  |  | . 93 |
| 114 | 0.906 | 20 | 0.908 | 6 | . 901 | 5 |  |  |  |  |  |
| 210 | ------- |  |  |  | 0.866 |  |  |  |  | 0.870 | 0.87 |
| 211 | 0.856 | 30 | 0.859 | 6 | . 853 | 8 | ------ |  |  | . 856 | . 86 |
| 204 | ------- | ----- | . 847 | 2 | . 839 | 1 | ------ | --- | ------------ | ------------- | ---------- |
| 006 |  |  | \{------- | ------ | . 821 | 1 | ------ | ---- | ------------ | 0.825 | ---------- |
| 212 | 0.824 | 10 |  |  | . 817 | 1 | ------ |  | ----------- | . 821 | 0.84 |
| 106 |  |  |  | ------ | . 773 | ------- | ------ | --- | ------------ | ----------- | . 80 |
| 213 |  |  | $\{0.772$ | 3 | . 766 | 3 | ------ |  | ------------ | ------------ | . 79 |
| 300 | 0.770 | 20 | ------ |  | . 764 | 3 |  |  |  | ----------- |  |
| 205 | . 753 | 10 | --------- | ------ | . 748 | ----- | ------ |  | ------------ | ------------ | . 78 |
| 302 | . 734 | 20 |  |  | . 729 | 1 |  |  |  |  | . 74 |
| 214 | . 714 | 5 | --------- | ------ | - |  | ------ |  | ----------- | ----------- | . 72 |
| 116 | . 700 | 5 | --------- | --..-- | ------ |  | ------ |  | ------------ | ---------- |  |

Table 6. Zinc (hexagonal)-Con.

| hkl | 1935 <br> Kotin and Losada $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | 1936 <br> Brindley <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ | 1937 <br> Wollan and Harvey $\mathrm{Cu}, 1.5405 \mathrm{~A}$ | 1937 <br> Miller <br> Mo, 0.7093 A | 1938 <br> Wroński <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ | 1938 <br> Hanawalt, Rinn and Frevel Mo, 0.7093 A |  | 1943 <br> Köhler <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | 1953 <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$, $26^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I | I | I | I | I | d | $I$ | $a$ | I | $a$ | I |
|  | A |  |  |  |  |  | A |  | A |  | A |  |
| 002 | 2.475 | 100 | 35 | 33 | 32 | 41 | 2.46 | 25 | 2.48 | 81 | 2.473 | 53 |
| 100 | 2.310 | 100 | 28 | 30 | 26 | 44 | 2.30 | 20 | 2.31 | 68 | 2.308 | 40 |
| 101 | 2.092 | 80 | 100 | 100 | 100 | 100 | 2.08 | 100 | 2.09 | 100 | 2.091 | 100 |
| 102 | 1.687 | 60 | 15 | 15 | 16 | 26 | 1.68 | 14 | 1.69 | 50 | 1.687 | 28 |
| 103 | \} 1.337 | 36 | 28 | 15 | 34 | 56 | 1.333 | 18 | $\{1.34$ | 53 | 1.342 | 25 |
| 110 |  |  |  |  |  |  |  |  | 1.33 | 36 | 1.332 | 21 |
| 004 | 1.237 | 31 |  |  | 2 |  | -- | ----- | 1.23 | 8 | 1.237 | 2 |
| 112 | 1.176 | 23 | 13 | 14 | 16 | 32 | 1.171 | 12 | 1.171 | 33 | 1.1729 | 23 |
| 200 | 1.155 | 25 |  |  |  |  | - | - | 1. 153 | 6 | 1.1538 | 5 |
| 201 | 1.124 | 21 | 9 | 11 | 11 | 26 | 1.122 | 8 | 1. 123 | 22 | 1.1236 | 17 |
| 104 | 1.090 | 25 |  |  |  |  |  | ----- | 1.089 | 8 | 1.0901 | 3 |
| 202 | 1.046 | 21 |  |  |  |  | 1.042 | 2 | 1.045 | 6 | 1.0456 | 5 |
| 203 | 0.9458 | 11 | 4 | 4 | 4 | 12 | 0.943 | 2 | 0.945 | 17 | 0.9454 | 8 |
| 105 | ) 9086 |  |  |  |  |  | \{. 907 | 2 | . 909 | 17 | . 9093 | 6 |
| 114 | \}. 9086 | 7 | 8 | 7 | 5 | 15 | \{----- | ----- | . 906 | 25 | . 9064 | 11 |
|  |  |  |  |  |  |  |  | ---- | . 905 | 14 | ------ |  |
| 210 | 0.8726 | 10 |  |  |  |  | ------- | ----- | . 872 | 8 | 0.8722 | 5 |
| 211 | . 8593 | 11 | 10 | 11 | 6 | 26 | ------- | ----- | . 859 | 31 | . 8589 | 9 |
| 204 | . 8438 | 5 | --------- | ----------- | -- | ------------ | ------- | ----- | . 857 | 19 | . 8437 | 2 |
| 006 | ------ | ----- |  |  |  |  |  |  | ----- | ---- | . 8245 | 1 |
| 212 |  | ----- |  | 4 | 1 |  |  |  | \{ 0.822 | 14 | . 8225 | 9 |
| 106 | ------ | ----- | --------- | --------- | ----------- | ----------- | ------- | ----- |  |  |  |  |
| 213 | ------- | ---- |  | ---------- | ----------- | ----------- | ------- | ----- | ---- |  |  | ---- |
| 300 |  |  |  |  |  |  |  |  | --- |  |  |  |
| 205 |  |  |  |  |  |  |  |  |  |  |  |  |
| 302 | ------- |  | ----------- | ----------- | ----------- | ------------ | ------- |  |  | --- |  |  |
| 214 |  |  |  | ------------ | ------------ | ------------ | ------- | ----- | ------ | ----- | ------- | --- |
| 116 |  |  |  |  |  |  |  |  |  |  |  |  |

$60.8 \times 10^{-6}$ parallel to the $c$-axis and $14.3 \times 10^{-6}$ perpendicular to it [175].

Unit cell at $25^{\circ} \mathrm{C}$ in angstrom units

|  |  | $a$ | c |
| :---: | :---: | :---: | :---: |
| 1929 | McLennan and Monkman [148]. | 2.662 | 4.960 |
| 1932 | Stenzel and Weertz [212] | 2.6643 | 4.9472 |
| 1932 | Boas [17] --------------------------- | 2.6640 | 4.9468 |
| 1933 | Hansen and Stenzel [86] | 2.6646 | 4.9466 |
| 1933 | Owen and Iball [174] | 2.6646 | 4.947 |
| 1935 | Jette and Foote [119].----------- | 2.6649 | 4.9468 |
| 1935 | Owen, Pickup, and Roberts [175].-- | 2.6648 | 4.9474 |
| 1953 | Swanson and Tatge | 2.665 | 4.947 |

The density, based on the NBS unit cell, is 7.134 at $25^{\circ} \mathrm{C}$.

### 2.6. Germanium (Cubic)

The two patterns for germanium in the ASTM file (see table l) were not published elsewhere; information regarding the first is limited to the author's name and date, and the second is a combined pattern from two sources. The two patterns are compared in table 7 with one prepared at the NBS and three from the literature, by Kolkmeijer [130], Nitka [166], and König [129].

The germanium used for the NBS pattern was obtained from Johnson, Matthey \& Co., Ltd., numbered 4065. Their spectrographic examination showed faint traces of silver, copper, sodium, and iron present as impurities.

The Kolkmeijer and Nitka patterns were published as Bragg angle data and the values of the spacings were calculated for table 7 directly in angstroms. The spacings of the other patterns were converted from $k X$ units to angstroms. In Nitka's pattern the reflection from the 400 plane was omitted; it is shown by the other patterns to be very weak. In the patterns of Nitka, König, and Fuller no reflections were recorded with indices higher than 51l. König noted and indexed as 222 a reflection in his electron diffraction pattern which is not consistent with the assumed diamond structure of germanium as found in X-ray patterns.

The intensity measurements of the patterns of Kolkmeijer and Nitka are estimated values represented by letters. No intensity values accompany the electron diffraction pattern by König. Schatzlein's pattern shows the customary high values associated with the absorption and focusing errors of much film work. A comparison between the Fuller pattern and the NBS pattern, after a rough conversion of the Fuller-Hanawalt intensity meas-
urements from molybdenum to copper radiation by means of the ASTM conversion chart ([1] page 108 of index covering original set of cards, or card No. vii of the introduction to the 1950 file), indicates good agreement of values. The NBS pattern is in close agreement with those of other investigators using copper radiation.

Germanium, cubic, has the structure of diamond and the space group $\mathrm{O}_{\mathrm{h}}^{7}(\mathrm{Fd} 3 \mathrm{~m})$ [105] wi th eight atoms in the unit cell. The lattice constant determined at the NBS is compared in the following table with determinations found in the literature, after their conversion to angstrom units at $25^{\circ} \mathrm{C}$. The coefficient of expansion of $5.92 \times 10^{-6}$ was used for the temperature conversions.

Unit cell, angstroms at $25^{\circ} \mathrm{C}$

| 1937 | Nitka [166].---------------------- | 5.659 |
| :---: | :---: | :---: |
| 1952 | Straumanis and Aka [215a].------ | 5.657640 |
| 1953 | Swanson and Tatge---------------- | 5.6576 |

The density, based on the NBS lattice constant, is 5.325 at $25^{\circ} \mathrm{C}$.

Table 7. Germanium (cubic)


### 2.7. Molybdenum (Cubic)

Molybdenum is represented in table 8 by three patterns from the ASTM file (see table 1); no additional patterns were found in the literature. A pattern was made at the NBS from a sample prepared by fused salt electrolysis by Seymour Senderoff of the Bureau. Spectrographic analysis showed very weak lines of $\mathrm{Al}, \mathrm{Fe}, \mathrm{Mg}$, and Si , and traces of $\mathrm{Ca}, \mathrm{Cu}$, Mn , and Pb . The unit-cell size remained unchanged after heating the finely divided powder in a vacuum furnace at $1,430^{\circ} \mathrm{C}$ for 1 hour.

The spacings of the three ASTM card patterns were converted to angstrom units for table 8. For the Davey and the Hanawalt, Rinn, and Frevel patterns the conversion was from kX units to angstroms; for the Hull pattern the radiation wavelength cited as 0.712 unit for molybdenum was used as the basis for the conversion. Only the first of four series
of interplanar spacings published by Cavey is given on an ASTM card, and only this is represented in table 8. Two patterns closely resembling these were published a year later by Davey in a German article [60]. Copper radiation used for the recording of the NBS pattern permitted the deternination of only seven lines.

The first and second strongest lines are generally agreed upon as the 110 and 211 , respectively. Hull and Davey list the 321 and 310 as third strongest, but their intensity values show the effect of sample absorption when compared with those of Hanawalt, Rinn, and Frevel and of the NBS, which agree upon the 200 as third strongest.

The molybdenum lattice is body centered cubic [106]. Molybdenum has the space group $\mathrm{O}_{\mathrm{h}}^{9}(\operatorname{Im} 3 \mathrm{~m})$, and two atoms in the unit cell. Correcting temperatures to $25^{\circ} \mathrm{C}$ with the

Table 8. Molybdenum

coefficient of expansion Michel [152] gives as $5 \times 10^{-6}$, and converting to angstrom units, the following lattice constants compare thus with the NBS determinations:

Unit cell at $25^{\circ} \mathrm{C}$ in angstrom units

| 1935 | Jette and Foote [119 | 3.1474 |
| :---: | :---: | :---: |
| 1941 | Lu and Chang [141] | 3.1467 |
| 1953 | Swanson and Tatge | 3.1472 |

The density, using the NBS unit-cell value, is 10.220 at $25^{\circ} \mathrm{C}$.

### 2.8. Palladium (Cubic)

Four patterns recorded on ASTM cards (see table l) and two patterns by Barth and Lund [7] and Jaeger and Zanstra [116] are represented in table 9. The sample of sponge palladium used for the NBS pattern was obtained from Johnson, Matthey \& Co., Ltd. Spectrographic analysis (in percent) at the Bureau showed $\mathrm{Ag}, 0.1$ to $0.01 ; \mathrm{Ca}, 0.01$ to 0.001 ; Cu , 0.01 to $0.001 ; \mathrm{Mg}, 0.01$ to 0.001 ; $\mathrm{Pb},<0.0001$; Pt, 0.01 to $0.001 ; \mathrm{Si}, 0.1$ to 0.01 . The sample was heated at $700^{\circ} \mathrm{C}$ for 15 minutes in vacuum and rechecked for a change in unitcell size. No appreciable change took place.

There is good agreement among various workers on the interplanar spacings of palladium. The spacings of the Hull pattern were converted to angstrom units on the basis of 0.712 as the wavelength used by Hull for molybdenum radiation. Those of the other patterns were converted from $k X$ units. The two Lavey patterns are essentially the same. The two most recent patterns, the Hanawalt, Rinn, and Frevel, and that of the NBS agree closely.

In contrast to the agreement of spacings among various workers, the intensity values
are not in complete accord. The 311 is recorded on all but two patterns as either first or second strongest. The two most recent patterns, that of Hanawalt, Rinn, and Frevel, and that of the NBS, show the lll, 200 , and 220 as first, second, and third strongest lines, putting the 311 in fourth place.

On the fiull card of the 1950 file the lattice constant is given as 3.950 and the density as 11.40. Although these data are referred to Wyckoff and to "C.C.," respectively, they are in fact from hull's own published work. The two Davey caro's of the new file both have lattice constants and densities ascribed to Kyckoff and to "C.C.," respectively; on the 1925 card these data are from Davey's own work, while on the 1926 card the lattice constant is from Kyckoff as represented, and the source of the density was not determined.

Palladium crystallizes in the cubic system [178] and has a space group $\mathrm{O}_{\mathrm{h}}^{5}(\mathrm{Fm} 3 \mathrm{~m})$. Two unit cell determinations made at specified temperatures were found in the literature. Converted to angstrom from $k X$ units and corrected to $25^{\circ} \mathrm{C}$ temperature, these are compared below with the NBS determination. For the correction, a coefficient of expansion of $11.8 \times 10^{-6}$ published by Owen and Yates [178] was used.

Unit cell, angstroms at $25^{\circ} \mathrm{C}$

| 1931 | Stenzel and Weerts [213] .. | 3.889 |
| :--- | :--- | :--- |
| 1933 | Owen and Yates [178] $\ldots . .-$ | 3.8905 |
| 1953 | Swanson and Tatge_......... | 3.8898 |

The density calculated from the NBS lattice constant is 12.04 at $25^{\circ} \mathrm{C}$.

Table 9. Palladium (cubic)


| hkl |  | $\begin{aligned} & 1931 \\ & \text { r and } \mathrm{Zar} \\ & 1.9360 \end{aligned}$ |  | Hanawa | 1938 <br> Rinn, $0.709$ | Frevel |  | $1953$ <br> nand $5405 \mathrm{~A}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I | a | d | I | $a$ | ${ }^{\text {d }}$ | I | a |
| 111 | $\begin{gathered} A \\ 2.221 \end{gathered}$ | 60 | A 3.847 3.85 | $\begin{gathered} A \\ 2.23 \end{gathered}$ | 100 | A 3.86 3.88 | A 2.246 | 100 | A 3.891 3.89 |
| 200 | 1.926 | 50 | 3.852 | 1.94 | 50 | 3.88 | 1.945 | 42 | 3.889 |
| 220 | 1.367 | 90 | 3.866 | 1.374 | 27 | 3.886 | 1.376 | 25 | 3.891 |
| 311 | 1.166 | 100 | 3.867 | 1.172 | 27 | 3.887 | 1.1730 | 24 | 3.8904 |
| 222 | 1.117 | 40 | 3.876 | 1.122 | 5 | 3.887 | 1.1232 | 8 | 3.8909 |
| 400 | ----- | ------- | ----- | 0.972 | 1 | 3.888 | 0.9723 | 3 | 3.8890 |
| 331 |  |  | ----- | . 893 | 5 | 3.892 | . 8924 | 13 | 3.8896 |
| 420 | ----- | ----.-- |  | . 871 | 5 | 3.895 | . 8697 | 11 | 3. 8893 |
| 422 | ----- | ------- |  | . 795 | 2 | 3.895 | --------- | ------- | --------- |
| 511 | ----- | ---...- | ---- | . 750 | 2 | 3.897 | -------- | ----.-- | -------- |
| 440 |  |  |  |  |  | ---- | --. | --- |  |
| 531 |  | ---- |  |  | ----- | ----- | ------ | ----- |  |
| 600 | ----- | ------- | ---- | --- | ------- | ------- | ------- | ------ | ------- |
| 620 | ----- |  |  |  |  |  |  |  |  |
| Average unit cell for last five lines. |  |  |  |  |  |  |  |  | 3.8898 |
|  |  |  | 3.862 | ----- | ------ | 3.893 | -------- | ----- |  |

### 2.9. Silver (Cubic)

There are six cards for silver in the ASTM file of diffraction patterns (see table 1). One of these (number 2-1098), for a "bismuth rich" silver, is not listed in either table 1 or 10 . Three lines in this pattern are not silver lines and are probably due to
a compound of silver and bismuth. When this card was duplicated for the 1950 reprinting, a unit cell measurement of pure silver was included that does not depend on any of the interplanar spacings on the card and misleadingly indicates that the pattern is for pure silver. Another card (3-1316), which is

Table 10. Silver (cubic)

listed in table l, gives no pattern, but only a unit cell measurement-a measurement actually appearing in only one [231] of the two papers to which it is ascribed. Of the remaining four patterns, the two by Jung [124] were published in the same paper. All four are compared in table 10 with a more recently published pattern by Harcourt [88], and one prepared at the NBS.

The NBS pattern was made from a sample furnished by Johnson, Matthey \& Co., Ltd., London, with a purity of more than 99.999 percent. Their spectrographic analysis indicated faint traces of calcium, iron, and copper.

The interplanar spacings of the Jung patterns were calculated directly in angstroms from the Bragg angle data given; the spacings of the other patterns were converted from $k X$ units. All patterns show 111 as the strongest line, but there is considerable difference as to the second and third strongest-Hanawalt, Rinn, and Frevel agree wi th the NBS on 200 and 311, respectively.

The atoms in silver are arranged in a face-centered lattice [231]. Silver has the space group $0_{h}^{5}$ ( $F m 3 m$ ), and four atoms in the unit cell. Published unit cell values are compared in the following table with that derived from the NBS pattern. Conversion to $25^{\circ} \mathrm{C}$ was made by means of a coefficient of expansion of $19.59 \times 10^{-6}$ [67], and all were corrected from $k X$ to angstrom units.

Unit cell in angstroms at $25^{\circ} \mathrm{C}$

| 1930 | Sachs and Weerts [ 200 | 4.0863 |
| :---: | :---: | :---: |
| 1932 | Owen and Iball [173] | 4.0862 |
| 1933 | Owen and Yates [178] | 4.0860 |
| 1933 | Saini [201] | 4.0862 |
| 1935 | Jette and Foote [119] | 4.0861 |
| 1936 | Hume-Rothery, Lewin, and Reynolds [109] | 4.0862 |
| 1939 | Owen and Roberts [177 | 4.0860 |
| 1940 | Foote and Jette [71] | 4.0861 |
| 1953 | Swans on and Tatge | 4.0862 |

The density of silver based on the NBS unit cell is 10.500 at $25^{\circ} \mathrm{C}$.

### 2.10. Tin (White or $\beta$ ) (Tetragonal)

The two patterns in the ASTM X-ray diffraction pattern file for tin are both for the tetragonal modification, referred to as white or $\beta$-tin (see table l). Three patterns for tin not included in the ASTM file were found in the literature; these are by Bijl and Kolkmeijer [14], Van Arkel [228], and Willot and Evans [248].

The sample of tin used for the NBS pattern was furnished by Johnson, Matthey \& Co., L.td., London, with the notation that the metal had been specially purified by Capper, Pass, \& Sons, Limited, who furnished the following analysis (in percent): lead, 0.0012; antimony, 0.001 ; iron, 0.00027 ; copper, 0.0002 ; arsenic, 0.0002 ; bismuth, 0.00012 ; sulfur, 0.00003 ; tin, 99.997 (by difference). Spectrographic analysis by Johnson, Matthey \& Co., Ltd., showed the following impurities: lead, faint; bismuth, faint; iron, very faint; sodium, faint; cadmium, very faint; calcium, very faint; magnesium, very faint; aluminum, barely visible; copper, barely visible; indium, barely visible in one spectrum only. The sample was annealed for 12 hours at $160^{\circ} \mathrm{C}$ before it was mounted in the spectrometer.

Interplanar spacings and intensity measurements of the six patterns are compared in table 1l. The interplanar spacings of Bijl and Kolkmeijer and of Van Arkel were calculated directly in angstrom units from their published Bragg angle data; for the remaining patterns they were converted from $k X$ units to angstroms. Intensity values are given numerically by only three of the patterns. These are in agreement with those of the NBS in designating the 200,101 , and 211 as the first, second, and third strongest lines, respectively.

White or $\beta$-tin belongs to the tetragonal system; Mark and Polanyi [142] in 1923 assigned it to space group $\mathrm{D}_{4}^{19}$ (I4/amd), a bodycentered lattice with two atoms in the unit cell. Recent unit cell measurements have

Table 11. Tin (white or $\beta$ )

been made by several workers, whose results are compared in the following table after conversion to angstroms at $26^{\circ} \mathrm{C}$. In making the temperature corrections coefficients of expansion obtained from Kosolapov and Trapeznikov [131] of $46.4 \times 10^{-6}$ perpendicular to the $c$ axis and of $22.4 \times 10^{-6}$ parallel with it were used.

## Unit cell in angstroms at $26^{\circ} \mathrm{C}$

|  |  | a | c |
| :---: | :---: | :---: | :---: |
| 1932 | Stenzel and Weertz [212]. | 5.8326 | 3. 1821 |
| 1935 | Jette and Foote [119]. | 5.83126 | 3.1814 |
| 1936 | Kosolapor and. Trapeznikov [131]. | 5.8311 | 3.1810 |
| 1938 | Ieviňs, Straumanis, and Karlsons [114] | 5.83146 | 3.18129 |
| 1953 | Swanson and Tatge. | 5.831 | 3.182 |

The density of tin based on the NBS lattice constant is 7.286 at $26^{\circ} \mathrm{C}$.

### 2.11. Tellurium (Hexagonal)

Of the eight patterns in table 12, six are recorcled on ASTM cards (see table l), one, by Bose and Ray [21], was found in the literature, and one was prepared at the NBS. The sample used for the NBS pattern was prepared in the laboratories of Johnson, Mathey and Co., Ltd., London, and was numbered 3824. Their spectrographic analysis showed $\mathrm{Si}, \mathrm{Fe}$, Mg , and Al present as faint traces. After being finely ground the sample was annealed in a vacuum furnace at approximately $400^{\circ} \mathrm{C}$ for 15 minutes.

For purposes of comparison, the spacings of the patterns in table 12 were converted to angstrom units except for those by Olshausen and by Bose and Ray, whose Bragg angle data enabled the derivation of interplanar spacings directly in angstrom units, and those of

Bradley and Slattery for which a correction factor could not be determined. The patterns by Harcourt, by Hanawalt, Rinn, and Frevel, and by the Institute of Physics at Cardiff, Wales, were presumed to be in kX units, and were converted accordingly. The Olshausen pattern includes a spacing of 3.58 A , incompatible with the tellurium structure and parameters, and another such spacing, of 5.8 A , is included in the Hanawalt, Rinn, and Frevel pattern.

With the exception of the Slattery pattern of 1924, all patterns accompanied by numerical relative intensity values are in agreement with the NBS pattern as to the three strongest lines: 101,102 , and 110 , in decreasing order.

The tellurium lattice is hexagonal closepacked. The space group was determined in 1924 [25] as enantiomorphic $\mathrm{D}_{3}^{4}$ or $\mathrm{D}_{3}^{6}\left(\mathrm{C}_{1} 2\right.$ or $\mathrm{C}_{2} 2$ ). There are three atoms in the unit cell. The only precision determination of the lattice constants found in the literature is by Straumanis [215], whose measurements, corrected for temperature and converted from kX to angstrom units are compared with the NBS values in the table below. For the temperature correction the coefficient of expansion given in the same paper was used; $27.51 \times 10^{-6}$ perpendicular to the $c$ axis, and $-1.70 \times 10^{-6}$ parallel to it.

Unit cell in angstroms at $25^{\circ} \mathrm{C}$

|  |  | $a$ | c |
| :---: | :---: | :---: | :---: |
| 1940 | Straumanis [215]. | 4.45653 | 5.92682 |
| 1953 | Swanson and Tatge. | 4.4570 | 5.9290 |

The density calculated from the NBS lattice constants is 6.2311 at $25^{\circ} \mathrm{C}$.

Table 12. Tellurium (hexagonal)

| $h k l$ | $1924$ <br> Bradley |  | $1925$ <br> Slattery |  | $\begin{gathered} 1925 \\ \text { Olshausen } \end{gathered}$ |  | $1927$ <br> Harcourt |  | 1938 <br> Hanawalt, Rinn, and Frevel Mo, 0.7093 A |  | $\begin{gathered} 1941 \\ \begin{array}{c} \text { Bose and } \\ \text { Ray } \end{array} \\ \mathrm{Cu}, 1.5405 \mathrm{~A} \end{gathered}$ |  | Institute Physics, Wales $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | $\begin{gathered} 1953 \\ \text { Swanson and } \\ \text { Tatge } \\ \mathrm{Cu}, 1.5405 \mathrm{~A}, \\ 26^{\circ} \mathrm{C} \end{gathered}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | d | I | d | I | d | I | d | I | d | I | d | $I$ | d | I |
|  | ( ${ }^{\text {) }}$ |  | ( ${ }^{\text {a }}$ |  | A |  | A |  | $\begin{array}{r} A \\ 5.8 \end{array}$ | 19 | A |  | A |  | A |  |
| 100 | 3.845 | 84 | 3.83 | 20 | 3.81 | w | 3.86 | 50 | 3.87 | 14 | 3.67 | w | 3.80 | 40 | 3.86 | 20 |
|  |  | -- | -.---- | --- | 3.58 | m | ------ | --- |  | ---- | ---- | --- | - | ---- | ------ | --- |
| 101 | 3.220 | 5 | 3.22 | 100 | 3.224 | s | 3.23 | 100 | 3.25 | 100 | 3.05 | vs | 3. 20 | 100 | 3.230 | 100 |
| 102 | 2.344 | 11 | 2.34 | 50 | 2.350 | m | 2. 33 | 80 | 2.34 | 48 | 2.86 | vvw | 2.33 | 80 | 2.351 | 37 |
| 110 | 2.219 | 16 | 2.22 | 40 | 2.215 | m | 2. 22 | 70 | 2.22 | 32 | 2.12 | s | 2.21 | 80 | 2.228 | 31 |
| 111 | 2.078 | 63 | 2.08 | 20 | 2.019 | vw | 2.07 | 50 | 2.08 | 14 | -- | --- | 2.06 | 60 | 2.087 | 11 |
| 003 | 1.968 | 63 | 1.969 | 20 | 1.965 | w | 1.97 | 50 | 1.96 | 14 | 1.99 | w | 1.96 | 60 | 1.980 | 8 |
| 200 | ----- | --- |  | -- |  | ---- | ---- | -- |  |  |  |  |  | ---- | 1.930 | 4 |
| 201 | 1.830 | 27 | 1.834 | 40 | 1.836 | m | 1.82 | 60 | 1.83 | 28 | 1. 87 | vvw | 1.82 | 80 | 1.835 | 20 |
| 112 | 1.765 | 79 | 1.777 | 20 | ----- |  | 1.77 | 30 | 1.77 | 10 | 1.78 | vvw | 1.77 | 20 | 1.781 | 7 |
| 103 | -.-..-- | -- | --.---- | --- | -- | ---- | ------- | -- | ------- | ---- | ------ | --- | ------ | ---- | 1.758 | 2 |
| 202 | 1.614 | 52 | 1.614 | 25 | 1. 619 | m | 1.61 | 60 | 1.61 | 20 | 1.55 | vvw | 1.61 | 60 | 1.616 | 12 |
| 113 | ) 1.464 | 32 | 1.469 | 30 | \{1.471 | ${ }^{\text {m }}$ | 1.47 |  | ) 1.473 | 28 |  |  | $\{1.47$ | 60 | 1. 479 | 13 |
| 210 | $)^{1.464}$ | 32 | 1.469 | 30 | 1.448 | w | 1. 448 | 30 | f 1.473 | 28 |  |  | \1.45 | 40 | 1.459 | 8 |
| 211 | 1.410 | 50 | 1.412 | 20 | 1.409 | ${ }^{w}$ | 1.413 | 50 | 1.421 | 13 | 1.41 | vvw | 1.41 | 60 | 1.417 | 8 |
| 104 | $\} 1.375$ | 50 | 1. 377 | 20 | 1.377 | ${ }^{*}$ | 1. 378 | 50 | 1.383 | 16 |  | ---- | 1.38 | 60 | 1.383 | 7 |
| 212 | 1.308 | 68 | 1.307 | 10 | 1.305 | m | 1.303 | 30 | 1.312 | 8 |  |  | 1. 30 | 40 | 1. 309 | 6 |
| 300 | 1.287 | 68 | ------- | --- |  |  | ------- | --- | ------- | ----- |  |  | --- | --- | 1.287 | 1 |
| 301 | 1.255 | 74 | 1. 267 | 3 | ------ | ---- | 1.254 | 20 | 1. 260 | 5 | ------ |  | 1.25 | 20 | 1.257 | 4 |
| 114 | --- | --- |  |  |  | ---- | 1.232 | 20 |  |  |  |  |  |  | 1. 234 | 1 |
| 302 |  |  |  |  |  | -- |  | --- |  |  |  |  |  | --- | 1.1802 | 3 |
| 204 213 | $\} 1.172$ | 21 | 1.171 | 30 | 1. 171 | m | 1. 172 | 70 | 1. 177 | 14 |  |  | 1.17 | 60 | 1. 1740 | 8 |
| 105 | 1.131 | 84 | 1. 130 | 8 | 1.129 | ${ }^{\text {v w }}$ | 1. 127 | 20 | 1. 121 | 5 |  |  |  | --- | 1. 1334 | 3 |
| 221 | 1. 092 | 95 |  | --- | 1.096 | vw | ------- | --- | ------- |  |  |  | 1.09 | 10 | 1.0951 | 2 |
| 303 | ------ | ---- | ------- | --- | ------ | ---- | ------- | --- |  |  |  | ----- | ------ | ---- | 1.0784 | 1 |
| 310 | 1.075 | 100 | ------- | --- | ------ | ---- | ------- | --- | -------- | -- |  |  |  | ---- | 1.0705 | 1 |
| 311 | ------ | ---- |  | --- | 1.049 | w | 1.050 | 20 | 1.047 | 5 |  |  | 1.05 | 20 | 1. 0535 | 3 |
| 115 | --- | - |  | --- | ----- | -- | ------ | -- | -------- |  |  |  |  | ---- | 1.0432 | 2 |
| 222 | 1.038 | 42 | ------- | --- | 1.038 | w | 1.039 | 10 | -------- | ----- | --.--- | ----- | 1.04 | 20 | 1.0399 | 3 |
| 214 | ----- | ---- | ------- | --- |  | -- | ------ | -- |  |  |  |  |  | --- | 1.0104 | 2 |
| 205 |  |  |  | --- | 1.009 | w | 1.007 | 10 | 1.007 | 5 |  |  | 1.00 | 20 | 1.0071 | 3 |
| 312 |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
| 006 |  | ---- |  | --- | ------ | --- | ------ | --- | -------- | ----- |  |  | ------ | ---- | 0.9889 | 1 |
| 304 |  |  |  | --- | ------ | ---- |  | --- | -------- | ----- |  |  |  | --- | . 9714 | 2 |
| 223 |  | ---- |  | --- | 0.966 | vd |  | --- | 0.970 | 2 |  |  |  | -- | . 9650 | 1 |
| 313 |  |  |  | --- | . 944 | w |  | --- |  |  |  |  |  |  | . 9413 | 1 |
| 215 |  |  |  | -.. | . 921 | w | - | -- |  |  |  |  |  |  | . 9201 | 2 |
| 116 |  | ---- |  | --- | . 905 | w | 0.900 | 10 |  | ----- | ------ |  | ---- | -- | . 9032 | 2 |
| 224 |  |  |  |  |  |  |  |  |  |  |  |  |  |  | (. 8909 | 1 |
| 320 | \}----- |  |  | --- | . 889 | w |  | --- |  |  |  |  |  |  | . 8858 |  |
| 206 |  |  |  |  |  |  |  |  |  |  |  |  |  |  | . 8760 | 1 |
| 321 |  |  |  |  |  |  |  | --- |  | ----- |  | ----- |  | ---- | . 8719 | , |
| 305 |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
| $314$ | \}----- | ---- |  | --- | 0.870 | $\pi$ | 0.866 | 10 | 0.868 | 2 |  |  | 0.868 | 10 | . 8675 | 4 |
| 322 |  |  |  |  | . 851 |  |  |  |  |  |  |  |  |  | . 8485 | 1 |
| 411 |  |  |  |  | . 838 |  |  |  |  |  |  |  |  |  | . 8339 | 2 |
| 107 |  |  |  | --- |  |  |  | --- |  |  |  |  |  |  | . 8270 | 1 |
| 216 |  |  |  |  |  |  |  | --- |  |  |  |  |  |  | . 8180 | 2 |
| 225 |  |  |  | --- |  |  |  | --- |  |  |  |  |  |  | . 8119 | 1 |
| 412 |  |  |  | --- |  |  |  | --- |  | ----- |  | ----- | ----- | -- | . 8102 | 2 |
| 404 |  |  |  |  |  |  |  |  |  |  |  |  |  |  | . 8082 | 3 |
| 315 |  |  |  |  |  |  |  |  |  |  |  |  |  |  | . 7945 | 2 |
| 117 |  |  |  |  |  |  |  | -- |  |  |  |  |  |  | . 7917 | 1 |

[^2]
### 2.12. Tungsten (Cubic)

Two tungsten patterns from the ASTM file (see table l) and five additional patterns from the literature, of which three comprise interplanar spacings only, are compared in table 13 with a pattern prepared at the NBS.

The patterns from the literature are by Becker [12], Debye [62], Neuburger [160], Sidhu [207], and Zeidenfeld [261]. The NBS pattern was made from a sample prepared and contributed by the Westinghouse Electric Corporation, who provided the following chemical analysis (in

Table 13. Tungsten (cubic)

${ }^{\text {a }}$ Unit not known. b Average for last two lines. ${ }^{c}$ Average for last three lines. ${ }^{d}$ Average for two lines preceding last line.
percent) by A. Pettel, Jr.: $\mathrm{SiO}_{2}, 0.04$; K, $0.05 ; \mathrm{Mo}, 0.01 ; \mathrm{Al}_{2} \mathrm{O}_{3}, 0.01 ; \mathrm{Fe}, 0.01$. This was verified by spectrographic analysis at the Bureau.

For table 13, data given in Bragg angles were used directly to derive interplanar spacings in angstroms for the patterns of Debye, Becker, Neuburger, and Sidhu. Davey's pattern was left in its original form, since the radiation wavelength was not given. Zeidenfeld gave a radiation wavelength of too few significant figures to show whether his data are in $k X$ units or angstroms. The spacings of Hanawalt, Rinn, and Frevel were converted from $k X$ units to angstroms.

Only three sets of intensity measurements are given with numerical values. Those of Hanawalt, Rinn, and Frevel, and of Swanson and Tatge show the same three strongest or index lines: 110, 211, and 321. Sidhu's estimated intensities agree with them.

The tungsten lattice is body-centered cubic with two atoms in the unit cell. Tungsten has the space group $\mathrm{O}_{\mathrm{h}}^{9}$ ( $\operatorname{Im} 3 \mathrm{~m}$ ) [160]. The lattice parameters derived by several investigators are compared in the table following. They were converted to angstrom units at $25^{\circ} \mathrm{C}$. The coefficient of expansion of $4.3 \times 10^{-6}$ of Michel [152] was used.

Unit cell at $25^{\circ} \mathrm{C}$ in angstroms

| 1932 | Owen and lball [173] | 3. 1657 |
| :---: | :---: | :---: |
| 1934 | Neuburger [160] | 3.1654 |
| 1935 | Jette and Foote [119] | 3.1648 |
| 1936 | Cohen [51] | 3.16473 |
| 1936 | Straumanis and Ieviňs [216] | 3. 1651 |
| 1941 | Lu and Chang [141] | 3. 1650 |
| 1953 | Swanson and Tatge.------ | 3. 1648 |

The density determined from the NBS lattice constant is 19.265 at $25^{\circ} \mathrm{C}$.

A less common form of tungsten, likewise stable at room temperature but with a simple cubic lattice, is represented by a third tungsten card in the ASTM file (old file number II-2579, new file number $2-1138$, index lines $2.25,2.06,1.34)$. This form, of different structure, is not to be confused with the form discussed here.

### 2.13. Tantalum (Cubic)

The three patterns given in the ASTM file (see table l) are supplemented by three additional patterns found in the literature, by Becker and Ebert [13], McLennon and Monkman [148], and Horn and Ziegler [96]. One of the ASTM patterns (Quill [189]) is recorded as made with molybdenum radiation al though copper radiation was actually employed. These patterns are compared with an NBS pattern in table 14.

The sample of tantalum used for the NBS pattern was procured from Johnson, Matthey \& Co., Ltd, London. The material contained dissolved gases which caused broadening of diffraction peaks, and TaH, which contributed extra lines. After annealing at $1,500^{\circ} \mathrm{C}$ in vacuum for 30 minutes in a tantalum boat the sample gave very sharp lines including only traces of the hydride. The spectrographic analysis furnished with the sample indicated faint traces of $\mathrm{Nb}, \mathrm{Al}, \mathrm{Si}, \mathrm{Fe}$, and Mn .

The interplanar spacings of table 14 are all given in angstrom units. The Becker, and the Ebert and Quill patterns were originally recorded as a series of Bragg angles, from which the interplanar spacings in angstroms were derived directly for the table. Hull's pattern was calculated by him with the use of a wavelength of 0.712 , on the basis of which his spacings were converted to angstrom units. The McLennon and Monkman, and the lianawalt, Rinn, and Frevel interplanar spacings were converted from kX units to angstroms. The Horn and Ziegler data are published presumably in angstroms.

The Horn and Ziegler measurements as well as those of Hull and Quill suffer from focusing and absorption effects. The Hanawalt, Rinn, and Frevel data agree wi th those of the NBS in designating the three strongest lines as the 110,211 , and 200 , in decreasing strength.

The tantalum lattice is body-centered cubic; the space group is $\mathrm{O}_{\mathrm{h}}^{9}$ (Im3m) [104]. There are two atoms in the unit cell. Two measurements of the lattice constant are compared in the table below with that of the NBS.

Table 14. Tantalum (cubic)



[^3]The data were converted to angstroms at $25^{\circ} \mathrm{C}$; the coefficient of expansion of $6.6 \times 10^{-6}$ [94] was used.

Unit cell in angstroms at $25^{\circ} \mathrm{C}$

| 1932 | Owen and Iball [173] | 3.3183 |
| :---: | :---: | :---: |
| 1936 | Neuburger [161] | 3. 3027 |
| 1953 | Swanson and Tatge | 3. 3058 |

The density as calculated from the NBS lattice constant is 16.626 at $25^{\circ} \mathrm{C}$.

### 2.14. Platinum (Cubic)

Three patterns for platinum are given in the X-ray diffraction pattern files of the ASTM (see table 1). Four additional patterns were obtained from the literature; these were made by Barth and Lunde [7], Jaeger and Zanstra [116], Rusterholz [199], and by Uspenski and Konobejewski [227]. The sample used to obtain a pattern at the NBS was prepared by R. Gilchrist of the Chemistry Division of the Bureau. The NBS Spectroscopic Laboratory estimated the purity at $>99.99$ percent.

All the interplanar spacings of the eight patterns of table 15 are given in angstrom units except those of Davey, for which a conversion constant could not be determined. The spacings of Hull were converted to angstroms on the basis of the wavelength $\lambda=0.712$ given for molybdenum $K a$ radiation; the remainder were calculated directly in angstroms from the Bragg angle data given.

It may be observed from table 15 that several of the patterns omit the weak 400 line. The three earliest patterns, by Hull, by Uspenski and Konobejewski, and by Davey, made
with molybdenum radiation, include 511, 531, and 600 lines beyond the range of patterns made with copper radiation. The pattern of Hanawalt, Rinn, and Frevel agrees with that of the NBS upon 111 and 200 as the two strongest lines, but shows 220 and 311 as equal in strength whereas the NBS pattern shows 311 as plainly stronger. This is evidently due to the difference in the radiation used, for upon recalculation of the Hanawalt, Rinn, and Frevel intensity values derived with molybdenum radiation to a copper radiation base ([1] page 108 of index covering original set of cards, or card number vii of introduction to 1950 file), the 311 is plainly the stronger in this pattern also. The earlier intensity measurements vary widely, suffering from the defects common to uncorrected film values.

The platinum lattice is face-centered cubic [106], $\mathrm{O}_{\mathrm{h}}^{5}(\mathrm{Fm} 3 \mathrm{~m})$, with four atoms in the unit cell. Of the many unit cell determinations found in the literature, two are accompanied by the temperature at which they were measured. These values were converted to $25^{\circ} \mathrm{C}$ by means of the coefficient of expansion $8.3 \times 10^{-6}$, an average of two published values [66, 178]. After conversion from $k X$ to angstrom units, comparison with the NBS data gives:

Unit cell at $25^{\circ} \mathrm{C}$, angstroms

| 1933 | Owen and Yates [178] ---------------------------------------------- | 3.9240 |
| :--- | :--- | :--- |
| 1937 | Moeller [155] | 3.92261 |

The density based on the NBS lattice constant is 21.472 at $25^{\circ} \mathrm{C}$.

Table 15. Platinum (cubic)

${ }^{a}$ Unit not know. ${ }^{b}$ Average for two lines only.

### 2.15. Gold (Cubic)

Three patterns for gold are included in the X-ray diffraction pattern file of the ASTM (see table l). These are compared in table 16 wi th a pattern prepared at the NBS. The sample used for the NBS pattern was purified by R. Gilchrist of the Chemistry Division of the Bureau. Spectrographic analysis showed faint traces of silicon and calcium (about 0.001 percent each), and possibly a faint trace of silver; thus the sample is about 99.997 percent gold.

In table 16 all interplanar spacings are given in angstrom units. The spacings of the Davey, the Hanawalt, Rinn, and Frevel, and the Harcourt patterns were converted from $k X$ units. Jung presented his data in Bragg angle values from which interplanar spacings were computed directly in angstroms for the table. The intensity measurements of Hanawalt, Rinn, and Frevel, of Harcourt, and of Swanson and Tatge agree as to the three strongest or index lines: 111, 200, and 311.

The gold lattice is face-centered cubic [232]; the space group is $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ). There are four atons per unit cell. Four measurements of the unit cell edge at specified temperatures were found in the literature. These: values were corrected to $25^{\circ} \mathrm{C}$ by the use of the coefficient of expansion of $15.2 \times 10^{-6}$ [66], and converted from $k X$ to angstrom units. They compare with the NBS determination, after all are corrected for units and temperature, as follows:

Unit cell at $25^{\circ} \mathrm{C}$, angstrows

| 1930 | Sachs and Weerts [200] | 4.0785 |
| :---: | :---: | :---: |
| 1933 | Owen and Yates [178] | 4.0786 |
| 1935 | Jette and Foote [119] | 4.0786 |
| 1938 | Esser, Eilander, and Bungardt [66]...- | 4.078 |
| 1953 | Swanson and Tatge--------------- | 4.0786 |

The density determined from the NBS lattice constant is 19.302 at $25^{\circ} \mathrm{C}$.

Table 16. Gold (cubic)

|  |  | 1925 |  |  | 1926 |  |  | 1926 |  |  | 1938 |  |  | 1942 |  |  | 1953 |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $h k l$ |  | Davey |  |  | Dave |  |  | Jung |  | Hana an | $1 \mathrm{t}$ Fre | Rinn, el |  | cou |  | Swanso | and | Tatge |
|  | Mo, | 0.709 | A | Mo, | 0.70 | 3 A | Cu, | 1.54 | 5 A | Mo, | 0.70 | 3 A | Cu , | 1. 54 | 5 A | $\mathrm{Cu}, 1.5$ | 405 | $26^{\circ} \mathrm{C}$ |
|  | d | $I$ | $a$ | d | I | $a$ | d | $I$ | $a$ | d | $I$ | a | ${ }^{\text {d }}$ | I | $a$ | d | I | $a$ |
|  | A |  | A | A |  | A | A |  | A | A |  | A | A |  | A | A |  | A |
| 111 | 2.35 | 100 | 4.07 | 2.35 | 100 | 4.08 | 2.349 | $s$ | 4.069 | 2.35 | 100 | 4.07 | 2.36 | 100 | 4.09 | 2.355 | 100 | 4.079 |
| 200 | 2.03 | 75 | 4.06 | 2.03 | 75 | 4.07 | 2.038 | ms | 4.076 | 2.03 | 53 | 4.06 | 2.04 | 67 | 4.08 | 2.039 | 52 | 4.078 |
| 220 | 1.439 | 75 | 4.070 | 1.440 | 75 | 4.072 | 1.436 | $s$ | 4.062 | 1.442 | 33 | 4.078 | 1.44 | 44 | 4.07 | 1.442 | 32 | 4.078 |
| 311 | 1.227 | 88 | 4.071 | 1.228 | 88 | 4.074 | 1.229 | s | 4.076 | 1.229 | 40 | 4.077 | 1.23 | 56 | 4.08 | 1.230 | 36 | 4.079 |
| 222 | 1.175 | 62 | 4.071 | 1.175 | 62 | 4.071 | 1.179 | w | 4.084 | 1. 175 | 9 | 4.071 | 1.177 | 12 | 4.078 | 1.1774 | 12 | 4.0786 |
| 400 | 1.018 | 38 | 4.072 | 1.018 | 38 | 4.072 |  | -- |  | 1.021 | 3 | 4.084 | 1.019 | 3 | 4.076 | 1.0196 | 6 | 4.0784 |
| 331 | 0.935 | 75 | 4.075 | 0.935 | 75 | 4.075 | 0.935 | s | 4.073 | 0.937 | 9 | 4.084 | 0.935 | 22 | 4.075 | 0.9358 | 23 | 4.0790 |
| 420 | . 911 | 75 | 4.073 | . 911 | 50 | 4.073 | . 913 | s | 4.083 | . 912 | 7 | 4.078 | . 912 | 22 | 4.078 | . 9120 | 22 | 4.0786 |
| 422 | . 832 | 50 | 4.074 | . 832 | 50 | 4.074 |  | -- |  | . 834 | 4 | 4.084 | . 832 | 33 | 4.074 | . 8325 | 23 | 4.0784 |
| 511 |  |  |  | . 784 | --- | 4.077 |  | -- |  | . 786 | 4 | 4.082 | . 786 | 33 | 4.082 |  |  |  |
| Average unit cell for last five lines $\qquad$ |  |  | 4.073 | ----- | --- | 4.074 | ----- | -- | 4.076 | ----- |  |  |  |  |  |  |  |  |
|  |  |  | --- |  |  |  |  |  |  |  | 4.082 | ----- | --- | 4.077 | ------ | --- | 4. 0786 |

### 2.16. Lead (Cubic)

Lead is represented by six patterns in the ASTM X-ray diffraction pattern file (see table l). An additional pattern to those of the ASTM cards, by Solomon and Jones [16], 1931, was found in the literature. They are compared in table 17 with a pattern prepared at the NBS.

The sample of lead used for the NBS diffraction pattern was obtained from the American Smelting and Refining Company. Spectrographic analysis at the NBS showed faint traces of bismuth and magnesium; the purity of the sample is believed greater than 99.999 percent. It was annealed for one hour at $180^{\circ} \mathrm{C}$ in petrolatum.

The interplanar spacings of the Levi and the Solomon and Jones patterns were calculated for table 17 directly in angstrom units from the published Bragg angle data. The remaining published patterns were converted from $k X$ units to angstroms. The interplanar spacings for the 1925 pattern of Davey were selected for the ASTM card from two sets of values published in adjacent columns. These two sets were averaged to obtain the pattern published by Davey in German in 1926. Two cards in the ASTM file have patterns credited to Harcourt; the interplanar spacings of these are identical and are given only once in table 17.

The intensity measurements of the Levi pattern were published as visual estimates, which were given numerical designations for the ASTM cards. The intensity measurements of the two Davey patterns are identical, as published; the strongest line has a value of 6, the others proportionately lower. These were converted to a base 100 for the strongest
line in transferring the data to the ASTM cards. However, the converted figures were given to two places for the 1925 pattern, and rounded off to one place for the 1926 pattern, with the result shown in table 17. The intensity values for the Solomon and Jones pattern, which is not included in the ASTM file, are given in the table as they were published. The intensity measurements of both Harcourt patterns are given; column $I_{1}$ refers to the set published in 1942, $I_{2}$ to the set found only in the ASTM file.

Lead has a face-centered cubic lattice [232]. It belongs to the space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ), and has four atoms to the unit cell. The length of the unit cell edge has been determined, with great accuracy, by many investigators. The following lattice constants, of fairly recent date, are converted to angstroms for comparison at a standard temperature. As published they are supposedly all in $k X$ units. The temperature correction was made by means of Owen and Yates' [178] value of $29.1 \times 10^{-6}$ for the coefficient of expansion at $20^{\circ} \mathrm{C}$.

Unit cell in angstroms at $25^{\circ} \mathrm{C}$

| 1932 | Owen and Iball [173] | 4.9505 |
| :---: | :---: | :---: |
| 1933 | Owen and Yates [178] | 4.9506 |
| 1933 | Obinata and Schmid [167 | 4.9496 |
| 1934 | Ölander [169] | 4.9492 |
| 1941 | Stokes and Wilson [214 | 4.9503 |
| 1941 | Fricke [76] | 4.950 |
| 1941 | Lu and Chang [141 | 4.9500 |
| 1946 | Klug [126] | 4.9508 |
| 1953 | Swanson and Tatge. | 4.9505 |

The density of lead based on the NBS determination of the unit cell is 11.341 at $25^{\circ} \mathrm{C}$.

Table 17. Lead (cubic)


| $h k l$ | Hanawal |  | d Frevel <br> A |  |  | 5 A |  |  | $1953$ <br> 5405 |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | $a$ | d | $I^{\text {a }}$ | $I^{\text {b }}$ | $a$ | d | I | $a$ |
| 111 | A 2.86 | 100 | $\begin{gathered} A \\ 4.95 \end{gathered}$ | $\begin{gathered} A \\ 2.85 \end{gathered}$ | 90 | 100 | $\begin{gathered} A \\ 4.94 \end{gathered}$ | $\begin{gathered} A \\ 2.855 \end{gathered}$ | 100 | $\begin{gathered} A \\ 4.945 \end{gathered}$ |
| 200 | 2.47 | 50 | 4.94 | 2.450 | 70 | 90 | 4.900 | 2.475 | 50 | 4.950 |
| 220 | 1.74 | 50 | 4.92 | 1.744 | 80 | 90 | 4.933 | 1.750 | 31 | 4.950 |
| 311 | 1.493 | 50 | 4.952 | 1.488 | 100 | 100 | 4.935 | 1.493 | 32 | 4.950 |
| 222 | 1.431 | 17 | 4.957 | 1.426 | 40 | 80 | 4.940 | 1.429 | 9 | 4.950 |
| 400 | --- | ----- | --------- | 1.235 | 10 | 50 | 4.940 | 1.238 | 2 | 4.950 |
| 331 | 1.136 | 17 | 4.952 | 1.135 | 70 | 90 | 4.947 | 1.1359 | 10 | 4.9513 |
| 420 | 1.107 | 17 | 4.951 | 1.107 | 70 | 90 | 4.951 | 1.1069 | 7 | 4.9502 |
| 422 | ------- | ----- | --------- | 1.011 | 70 | 90 | 4.953 | 1.0105 | 6 | 4.9504 |
| 511 | ---- |  |  | 0.9534 | 70 | 90 | 4.9540 | 0.9526 | 5 | 4.9500 |
| 440 |  |  | --------- | . 877 | 10 | 50 | 4.961 | . 8752 | 1 | 4.9508 |
| 531 | ------- |  | -------- | . 8382 | 70 | 90 | 4.9589 | . 8369 | 9 | 4.9510 |
| 620 | ------- | ---- | --------- | . 8267 | 60 | 90 | 4. 9602 | . 8251 | 4 | 4.9507 |
| 533 |  |  | ------ | --------. |  |  | ------- | -------- |  |  |
| 622 |  |  |  |  |  |  |  |  |  |  |
| 444 | - |  |  |  |  |  |  |  |  |  |
| Average unit cell for last five lines $\qquad$ |  |  |  |  |  |  | ${ }^{\text {c }} 4.9577$ |  |  | 4.9506 |
|  |  |  | 4.953 |  |  | - |  | --- | - |  |

${ }^{\text {a }}$ As first published.
${ }^{b}$ On ASTM card.
${ }^{c}$ Average for 511,531 , and 620 lines.

### 2.17. Beryllium Oxide, Be0 (Hexagonal)

In addition to four patterns for beryllium oxide (bromellite) included in the ASTM file (see table 1), patterns by Zachariasen [258, 259] and by Cl aassen [47], found in the literature, are compared with an NBS pattern in table 18. Since the two Zachariasen patterns are very similar, only the first is reproduced in the table.

The sample of BeO used for the Bureau pattern was prepared by the Brush Beryllium Company. The material, No. 1743-1747, is of fluorescent grade, and was prepared at a furnace temperature of $1,150^{\circ} \mathrm{C}$. Spectrographic analysis at the Bureau laboratory indicated about 0.03 percent $\mathrm{Al},<0.01$ percent each of $\mathrm{Ca}, \mathrm{Fe}, \mathrm{Mg}$, and Si , and traces of $\mathrm{Cu}, \mathrm{Pb}$, and Sn .

For the McKeehan pattern the ASTM card carries spacings derived from the author's Bragg angle data, while for table 18, d was
obtained directly from the author's log d values, and converted from $k X$ to angstrom units. The Claassen interplanar spacings were calculated directly in angstrom units for table 18, from the Bragg angle data published. Since it is not clear from the ASTM card whether the pattern of the United Steel Companies, England,.is in $k X$ or angstroms, it was not altered. All others were converted to angstroms upon the assumption that they are published in $k X$ units. Two lines, 114 and 212, not previously observed, show up in the NBS pattern. The 004 and 104 appearing in the two Zachariasen patterns and in the United Steel Companies pattern were observed only with difficulty in the NBS pattern. In the United Steel Companies pattern the line of interplanar spacing 0.993 is indexed as a compound reflection from 104 and 113 planes. As the presence of a 113 reflection is not

Table 18. Beryllium oxide, BeO (hexagonal)

| $h k l$ | $1922$ <br> McKeehan <br> Mo, 0.7093 A |  | 1925 <br> Zachariasen <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | 1925 <br> Aminoff <br> $\mathrm{Fe}, 1.9360 \mathrm{~A}$ |  | 1926 <br> Claassen <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | 1938 <br> Hanawalt, Rinn, and Frevel Mo, 0.7093 A |  | United Steel <br> Co, 1.7902 A |  | $1953$ <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | $I$ | d | $I$ | d | I | ${ }^{\text {d }}$ | I | d | I | d | I | d | I |
|  | A |  | A |  | A |  | A |  | A |  | (a) |  | A |  |
| 100 | 2.34 | 80 | 2.34 | 100 | 2.33 | 100 | 2.40 | 81 | 2.34 | 80 | 2.34 | 100 | 2.337 | 91 |
| 002 | 2.18 | 60 | 2.20 | 60 | 2.18 | 50 | 2.20 | 58 | 2.19 | 50 | 2.19 | 80 | 2.189 | 61 |
| 101 | 2.06 | 100 | 2.07 | 100 | 2.05 | 100 | 2.14 | 100 | 2.06 | 100 | 2.06 | 100 | 2.061 | 100 |
| 102 | 1.601 | 30 | 1.60 | 50 | 1.59 | 50 | 1.60 | 24 | 1.59 | 24 | 1.60 | 60 | 1.598 | 22 |
| 110 | 1. 349 | 80 | 1.35 | 80 | 1.34 | 75 | 1.35 | 39 | 1.353 | 32 | 1.35 | 70 | 1.349 | 29 |
| 103 | 1.239 | 80 | 1.24 | 80 | 1.20 | 75 | 1.24 | 31 | 1.242 | 32 | 1.24 | 70 | 1.238 | 24 |
| 200 | ------- | ----- | 1.167 | 20 | 1.165 | 25 | 1.169 | 4 | 1.172 | 4 | 1.17 | 40 | 1. 1682 | 4 |
| 112 | 1.148 | 60 | 1.149 | 70 | 1.144 | 75 | 1.151 | 31 | 1.152 | 20 | 1.15 | 60 | 1.1482 | 16 |
| 201 | ------- | ----- | 1.129 | 20-30 | 1.123 | 25 | 1.101 | 4 | 1.132 | 4 | 1.13 | 40 | 1.1287 | 5 |
| 004 | ------- | ----- | 1.119 | 0-10 | -----.- | ----- | --.-.-. | ----- | ------- |  | 1.09 | 20 | 1.0958 | $<1$ |
| 202 |  | ----- | 1.031 | 10-20 | 1.025 | 25 | 1.026 | ----- | 1.034 | 3 | 1.03 | 40 | 1.0308 | 3 |
| 104 | ------- | - | 0.995 | 0-10 |  | ----- | 0.988 | 4 | ------- |  | 0.993 | 20 | 0.9920 | $<1$ |
| 203 | 0.910 | 20 | . 911 | 70 | ------- | ----- | . 918 | ----- | 0.917 | 8 | . 914 | 70 | . 9118 | 10 |
| 210 | . 885 | 10 | . 881 | 40 | ------- | ----- | . 886 | 13 | . 882 | 2 | -.-.-.- | ----- | . 8832 | 4 |
| 211 | . 866 | 10 | . 864 | 50 | ------- | ----- | . 866 | 7 | . 872 | 2 | ------ | ----- | . 8657 | 5 |
| 114 | ---- | -- | ------- | ----- | ------- | ----- | ------- | ----- | ------- | ------ | - | ----- | . 8498 | 2 |
| 105 | 0.820 | 20 | ------- | ----- | ------- | ----- | ------- | ----- | 0.824 | 8 | ------- | ----- | . 8199 | 14 |
| 212 | ------- | --- |  |  |  | ----- | ------ | ----- | ------- | ------ | ------ | ----- | . 8179 | 8 |
| --- | 0.780 | 10 |  | ----- |  | --.-- |  | --.-- | 0.782 | 3 |  |  |  | --- |
|  | ------- | - |  |  |  |  |  |  | . 760 | 8 |  |  |  | ----- |
| --- | 0.755 | 20 |  |  |  | ----- |  |  | -.----- | ------ | ------- | ----- | ---------- | ------- |

[^4]compatible with the structure worked out by Zachariasen, the line is indexed only 104 in table 18.

The intensity values of the Cl aassen, the Hanawalt, Rinn, and Frevel, and the NBS patterns are closely comparable, with the 101, 100 , and 002 lines appearing as the first, second, and third strongest, respectively.

Zachariasen [259] in 1926 recorded the space group determination of $\mathrm{C}_{6 \mathrm{v}}^{4}$ (C6mc) for hexagonal beryllium oxide. There are two molecules in the hexagonal unit cell. Two lattice constants found in the literature compare with that determined from the NBS data as follows:

Unit cell, in angstroms

|  |  | a | c |
| :---: | :---: | :---: | :---: |
| 1925 | Aminoff [2] | 2.69 | 4.37 |
| 1926 | Zachariasen [259] | 2.699 | 4.401 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 2.698 | 4.380 |

The density, based on the NBS lattice constant, is 3.008 at $26^{\circ} \mathrm{C}$. The material used was too finely divided for determination of the refractive index.

### 2.18. Magnesium Oxide, Mg0 (Cubic)

Four patterns for magnesium oxide (periclase) lịsted in table 19 appear in the ASTM file (see table 1). The pattern of Hansen and Brownmiller, card number 2-1395, is erroneously labelled $\mathrm{Mg}(\mathrm{OH})_{2}$ in the 1950 file, although correctly ascribed to MgO in the old file. However, it appeared in the old index as $\mathrm{Mg}(\mathrm{OH})_{2}$ and was repeated thus in the new. Two of the patterns of table 19 are combined on one ASTM card; the United Steel Companies, England, interplanar spacings parallel the
intensity measurements of Wyckoff and Armstrong. Five patterns were obtained from the literature; they are by Büssem, Schusterius, and Ungewiss [43], Frevel [74], Gerlach [79], Menzer [151], and Passerini [182].

The NBS pattern was made from a sample contributed by the Radio Corporation of America as a pure compound prepared for use in phosphor research [135]. The $\mathrm{N}!\mathrm{gO}$ was crystallized in a graphite crucible which was maintained at $1,800^{\circ} \mathrm{C}$ for three hours. An NBS spectrographic analysis shows calcium and silicon between 0.01 and 0.1 percent; aluminum, boron, chromium, iron, and nickel, between 0.001 and 0.01 percent.

Some of the patterns of table 19 were corrected to angstroms from $k X$ units. Those of Gerlach, of Passerini, of Wyckoff and Armstrong, of Büssem, Schusterius, and Ungewiss, and of Menzer were calculated directly in angstroms from the published Bragg angle data. Two errors occur in the Hansen and Brownmiller pattern; the spacing for $h k l=200$ is published as 2.01 , doubtless in error for 2.10 , as the ASTM card notes; and the spacing 1.243 is superfluous to the pattern. The Hanawalt, Rinn, and Frevel, and the Frevel patterns show two $K a_{2}$ lines which are not listed in table 19, where only $K \alpha_{1}$ lines are tabulated. In the table the complete pattern of Wyckoff and Armstrong is given, of which only the intensity values for the first eight of 16 lines appear on the ASTM card. The two strongest lines are given in almost every case as 200 and 220 , but the third strongest is not universally agreed upon. Three patterns (including the two of most recent date) and the NBS pattern agree that 420 is third strongest.

Table 19. Magnesium oxide, MgO (cubic)

${ }^{\text {a }}$ Published intensity values. b Intensity values as they appear on ASTM card. © Average of four lines only. d Average of three lines only.

Magnesium oxide has a face-centered cubic lattice [101], space group $\mathrm{O}_{\mathrm{h}}^{5}(\mathrm{Fm} 3 \mathrm{~m})$, and four molecules in the unit cell. Unit cell values are tabulated below for comparison. They are all given in angstrom units, the three published ones converted from kX units, at $25^{\circ} \mathrm{C}$. The coefficient of expansion $14.45 \times 10^{-6}$ [42] was used.

Unit cell in angstroms at $25^{\circ} \mathrm{C}$

| 1935 | Büssem, Bluth, and Grochtmann [42 | 4.211 |
| :---: | :---: | :---: |
| 1936 | Straumanis and Ievinš [216] | 4.2115 |
| 1944 | Freve1 [74] | 4.214 |
| 1953 | Swanson and Tatge | 4.213 |

The density calculated from the NBS lattice constant is 3.581 at $25^{\circ} \mathrm{C}$. The NBS sample shows an index of refraction of $n=1.732$.

### 2.19. Silicon Dioxide (Low or a-cristobalite), $\mathrm{SiO}_{2}$ (Tetragonal)

Seven ASTM patterns (see table 1) for a-cristobalite are represented by nine original patterns (some are combined on the cards) in table 20. These are compared with an additional pattern from the literature, by Jay [117], and one produced at the NBS. An eighth card (number 956 of the original set) is mistakenly referred to in the original ASTM index [1] as the a form. This card, which is not itself designated a or $\beta$, is represented in the new edition by a pattern labelled correctly " $\beta$-Cristobalite" (card number 1-0430).

The NBS sample was obtained from the Radio Corporation of America Laboratories, Princeton, N. J. It was purified in connection with the RCA Phosphor project [135], and had been heated for two hours at $1,420^{\circ} \mathrm{C}$; a
trace of tridymite showed up in the $X$-ray diagram.

All the patterns of the table were changed' from $k X$ to angstrom units except that of Thilo, which was calculated directly in angstroms from Bragg angle data given. Hith regard to intensity measurements, the three strongest lines are the same as those of the NBS pattern-101, 200, 102-al though some of the patterns show 111 equal in intensity to 102, except for the British Museum pattern, in which 102 appears stronger than 200. The intensity values of most of the patterns were published as visual estimates, but are given in the table in the numerical conversion shown on the ASTM cards; the measured intensities of Barth and of Hanawalt, Rinn, and Frevel are converted to a base of 100 for the strongest line as on the ASTM cards.

Early workers considered alpha cristobalite cubic or nearly so. Nieuwenkamp [163] established the tetragonal structure of the mineral, showing that it has the space group $D_{4}^{4}\left(\mathrm{P}_{4} 2_{1}\right)$, and the enantiomorphous form $D_{4}^{8}$ $\left(P 4_{3} 2_{1}\right)$. There are four molecules in the unit cell. The following table compares lattice constants from his data with those later determined by Jay [117] and those based on the NBS pattern, all in angstrom units.

Unit cell, in angstroms

|  |  | a | c |
| :---: | :---: | :---: | :---: |
| 1935 | Nieuwenkamp [163] | 4.97 | 6.93 |
| 1944 | Jay [117] ( $22^{\circ} \mathrm{C}$ ) | 4.9715 | 6.9193 |
| 1953 | Swanson and Tatge ( $27^{\circ} \mathrm{C}$ ) | 4.973 | 6.95 |

The density was calculated from the NBS unit cell as 2.32 at $27^{\circ} \mathrm{C}$. The indices of refraction were determined as $\epsilon=1.484$ and $\omega=1.486$.

Table 20. Silicon dioxide (low or $\alpha$-cristobalite), $\mathrm{SiO}_{2}$ (tetragonal)

(Continuea)

Table 20. Silicon dioxicie (low or a-cristobalite), $\mathrm{SiO}_{2}$ (tetragonal)-Con.


### 2.20. Silicon Dioxide (High or $\beta$-Cristobalite), $\mathrm{SiO}_{2}$ (Cubic)

Al though an NBS pattern was not made for high or $\beta$-cristobalite, the published patterns were reviewed in order to select the most suitable one for retention in the ASTM file. Three cards in the latest edition of the ASTM file of X-ray diffraction patterns record two patterns for high or $\beta$-cristobalite; the two patterns are combined on the third card (see table 1). Two references are given to Wyckoff $[254,255]$ - one on the simple card, one on the combined card; they were published the same year, the one in German a translation of the one in English, and the patterns

Table 21. Silicon dioxide (high or f-cristobalite), $\mathrm{SiO}_{2}$ (cubic)

| hkl | $\begin{aligned} & 1925 \\ & \text { Wyckoff } \\ & 0.7093 \mathrm{~A}, \\ & 290^{\circ} \mathrm{C} \end{aligned}$ |  |  | $1932$ <br> Barth and Posnjak <br> Mo, 0.7093 A , $500^{\circ} \mathrm{C}$ |  |  | Combined ${ }^{\text {a }}$ <br> Mo, 0.7093 A |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | $a$ | a | I | a | a | I | $a$ |
|  | A |  | A | A |  | A | A |  | A |
| 111 | 4.137 | 100 | 7.165 | 4.15 | 100 | 7.19 | 4.14 | 100 | 7.17 |
| 211 | ----- | -- | ----- | 2.92 | 5 | 7.15 | 2.92 | 5 | 7.15 |
| 220 | 2.524 | 45 | 7.139 | 2.53 | 80 | 7.16 | 2.53 | 90 | 7.16 |
| 311 | ----- | -- |  | 2.17 | 10 | 7.20 | 2.17 | 10 | 7.20 |
| 222 | 2.070 | 13 | 7.171 | 2.07 | 30 | 7.17 | 2.07 | 30 | 7.17 |
| 320 |  | -- |  | 1.99 | 5 | 7.18 | 1.99 | 5 | 7.18 |
| 400 | 1.779 | tr. | 7.116 | 1.793 | 5 | 7.172 | 1.79 | 10 | 7.16 |
| 411 | ----- | -- |  | 1.688 | 5 | 7.162 | 1.69 | 5 | 7.17 |
| 331 | 1.637 | 35 | 7.136 | 1.639 | 60 | 7.144 | 1.639 | 70 | 7.144 |
| 422 | 1.455 | 30 | 7.128 | 1.469 | 50 | 7.148 | 1.457 | 60 | 7.138 |
| 511 | 1.372 | 10 | 7.129 | 1.379 | 20 | 7.165 | 1.376 | 20 | 7.150 |
| 440 | 1.261 | 15 | 7.133 | 1.265 | 30 | 7.156 | 1.263 | 30 | 7.145 |
| 531 | 1.203 | 28 | 7.117 | 1.209 | 30 | 7.153 | 1.206 | 50 | 7.135 |
| 620 | 1.125 | 10 | 7.115 | 1.130 | 20 | 7.147 | 1.127 | 20 | 7.128 |
| 533 | 1.085 | tr. | 7.115 | 1.089 | 5 | 7.141 | 1.087 | 10 | 7.128 |
| 444 | 1.031 | tr. | 7.143 | 1.029 | 5 | 7.129 | 1.030 | 10 | 7.136 |
| 711 | 0.993 | 5 | 7.091 | 1.000 | 10 | 7.141 | 0.997 | 10 | 7.120 |
| 642 | . 949 | 7 | 7.102 | 0.956 | 10 | 7.154 | . 953 | 20 | 7.131 |
| 731 | . 924 | 4 | 7.097 | . 929 | 10 | 7.136 | . 927 | 10 | 7.120 |
| 822 | . 838 | 3 | 7.110 |  | -- |  | . 838 | 5 | 7.110 |
|  | age un 1 for <br> e line |  | 7.109 |  | -- | 7.140 |  | - | 7.123 |

[^5]are identical. Two literature references to Barth and Posnjak [8], one on a simple, one on a combined card, are identical, although the combined card erroneously lists the junior author first. The older edition of the ASTM file labelled the Wyckoff pattern simply "cristobalite" and listed it in the accompanying index as " $\alpha$-cristobalite." In the 1950 edition the card is correctly labelled and indexed. The combined pattern apparently represents an average of the two published patterns, with the addition of lines given by one or the other.

Some of the data on the ASTM cards is confusing. The new card for the Hyckoff pattern states that the material is stable "over $275^{\circ}$ " but does not indicate that the pattern was prepared at $290^{\circ} \mathrm{C}$. The Barth and Posnjak card does not mention temperature. The combined card gives the Barth and Posnjak temperature correctly as $500^{\circ}$, the Myckoff temperature erroneously as $430^{\circ}$, and states cryptically on the card " $\mathrm{SiO}_{2}$ at about $450^{\circ}$," which is evidently meant for a rough average of the preceding values. An error occurs in the listing of intensity measurements on the Wyckoff card; the third line should read "13" rather than " 7 " (on the old card " 0.125 " rather than " 0.07 ") .

For table 21 the spacings of the ASTM patterns were reduced to angstrom units on the basis of the wavelength used for molybdenum radiation. Since, in the temperature range indicated, the coefficient of expansion is of the order $8 \times 10^{-6}$ [42], the difference in the two sets of spacings due to temperature is very little. The intensities of the three patterns correlate well; in each case the three strongest lines are represented by 111, 220, and 331.

The space group of $\beta$-cristobalite, which belongs to the cubic system, is $\mathrm{T}^{4}(\mathrm{P} 2,13)$ [42]. There are eight molecules in the unit cell. Published unit cell measurements, converted to angstrom units at $500^{\circ} \mathrm{C}$ (using the coefficient of expansion noted above) compare as follows:

Unit cell in angstroms at $500^{\circ} \mathrm{C}$

| 1929 | Wyckoff [254] | 7.127 |
| :---: | :---: | :---: |
| 1932 | Barth and Posnjak [8] | 7.16 |
| 1935 | Büssem, Bluth, and Grochtmann [42] | 7.1282 |

As noted in table 2l, however, an average of the last five lines of the Barth and Posnjak pattern at $500^{\circ} \mathrm{C}$ yields 7.140 A for the lattice constant, which is closer to the value of other workers. Because of its greater completeness, as it shows several low angle lines not given in the earlier pattern, it is recommended that the Barth and Posnjak pattern be selected as the standard ASTM pattern.

### 2.21. Calcium Oxide, CaO (Cubic)

The file of X-ray diffraction patterns of the ASTM includes three cards for calcium oxide (see table l). One of these is a composite of lines from three sources, of which one was previously unpublished. The four previously published patterns are compared in table 22 with two additional patterns found
in the literature by Gerlach [77], and by Natta and Passerini [157], and a pattern prepared at the NBS.

The NBS sample was obtained as calcium carbonate from the J. T. Baker Chemical Co., No. 121647, and calcined in a platinum crucible at $925^{\circ} \mathrm{C}$ for l hour. The following chemical analysis (in percent) was provided by the chemical laboratory of the NBS: insoluble in HCl and $\mathrm{NH}_{4} \mathrm{OH} p p t, 0.01$; chloride, $<0.005$; sulfate, 0.037 ; alkalis (as $\mathrm{SO}_{4}$ ), 0.011 ; barium, <0.l; heavy metals ( $\mathrm{Pb}, \mathrm{etc}$.$) ,$ $0.001 ; \mathrm{Fe},<0.003 ; \mathrm{MgO}$ and alkalis, 0.21 . The J. T. Baker Chemical Company specified the barium content as 0.005 percent and the iron as 0.001 percent. In preparing the pattern a petrolatum mount minimized hydration.

The interplanar spacings are given in angstroms in table 22. The Gerlach [77] pattern was calculated in angstroms from published Bragg angle data. The Harrington [89] and the Brownmiller and Bogue [40] spacings were converted to angstrom units in accordance with the wavelength given for the radiation

Table 22. Calcıum oxide, CaO (cubic)

| $h k l$ | $\begin{gathered} 1922 \\ \text { Gerlach } \\ 1.5405 \mathrm{~A} \end{gathered}$ |  |  | 1927 <br> Harrington <br> Mo, 0.7093 A |  |  | $1929$ <br> Natta and Passerini $\mathrm{Cu}, \quad 1.5405 \mathrm{~A}$ |  |  | $1930$ <br> Brownmiller and Bogue $\text { Mo, } 0.7093 \mathrm{~A}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | ${ }^{\text {d }}$ | I | $a$. | ${ }^{\text {d }}$ | I | $a$ | d | I | $a$ | $d$ | I | $a$ |
| 111 | $\begin{gathered} { }_{2}^{A} \end{gathered}$ | ms | $\begin{gathered} A \\ 4.801 \end{gathered}$ | $\begin{gathered} A \\ 2.77 \end{gathered}$ | 70 | $\begin{gathered} A \\ 4.80 \end{gathered}$ | $\begin{gathered} A \\ 2.626 \end{gathered}$ | w | $\begin{gathered} A \\ 4.548 \end{gathered}$ | $\begin{gathered} A \\ 2.754 \end{gathered}$ | m | $\begin{gathered} A \\ 4.770 \end{gathered}$ |
| 200 | 2.398 | s | 4.796 | 2.40 | 100 | 4.80 | 2.372 | ms | 4.744 | 2. 381 | ss | 4.762 |
| 220 | 1.689 | s | 4.777 | 1.698 | 100 | 4.803 | 1.683 | s | 4.760 | 1.688 | s | 4.774 |
| 311 | 1.438 | m* | 4.769 | 1.448 | 80 | 4.802 | 1.440 | ms | 4.776 | 1.439 | m | 4.773 |
| 222 | 1.379 | mw | 4.777 | 1.387 | 80 | 4.805 | 1.381 | ms | 4.784 | 1.380 | m | 4.780 |
| 400 | 1.193 | w | 4.772 | 1.200 | 60 | 4.800 | 1.193 | mw | 4.772 | ------- | ------ |  |
| 331 | 1.095 | w | 4.773 | 1.100 | 60 | 4.795 | 1.096 | mw | 4.777 | ------ | ------ | -- |
| 420 | 1.082 | s | 4.839 | 1.073 | 80 | 4.799 | 1.073 | s | 4.799 | 1.071 | m | 4.790 |
| 422 | 0.9802 | ms | 4.802 | 0.978 | 70 | 4.791 | 0.980 | s | 4.801 | 0.976 | m | 4.781 |
| 511 | . 9133 | ms | 4.746 | . 922 | 50 | 4.791 | . 926 | mw | 4.812 | ------- | ------ |  |
| 440 | . 8454 | m | 4.782 | . 847 | 30 | 4.791 | . 847 | m | 4.791 | ----- | ---- | --- |
| 531 | . 8110 | s | 4.798 | . 810 | 50 | 4.792 | . 84 | ------ | . | ---------- | -------- | ----- |
| 600 | . 8003 | s | 4.802 | . 798 | 40 | 4.788 | ------- | ------ | ------- | ----.--- | ------ | -------- |
| 620 | -------- |  | ------- | . 756 | 40 | 4.781 | ------- | ------- | ------- | -------- | ------ | ---- |
| 533 |  |  |  |  |  | -------- |  |  |  |  |  |  |
| 622 | -------- |  | ------- | 0.722 | 40 | 4.789 | ------- | ------ |  | ------- | ------ | --- |
| 444 |  |  |  | . 692 | 10 | 4.794 | ------- | ------ | ------- | ------- | ------ |  |
| 711 | -------- |  |  | . 671 | 30 | 4.792 |  |  |  |  |  |  |
| 640 |  |  |  | . 665 | 20 | ${ }^{\text {a }} 4.795$ |  |  |  |  |  |  |
| Average unit cell for last five lines $\qquad$ |  |  |  |  |  |  |  |  |  |  |  |  |
|  |  |  | 4.786 |  | ---- | ${ }^{\text {b }} 4.779$ | ------- | ------ | 4.796 |  |  | 4.780 |

[^6][^7](Continued)

Table 22. Calcium oxide, CaO (cubic)-Con.

| $h k l$ | Hanawa | $\begin{array}{r} 1938 \\ \text { qinn, } \\ 0.70 \end{array}$ | Frevel |  | Co, |  |  |  | $1953$ son an $.5405$ | ge <br> $7^{\circ} \mathrm{C}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | $I$ | $a$ | d | $I^{\text {c }}$ | $I^{\text {d }}$ | a | a | $I$ | $a$ |
| 111 | $\begin{gathered} A \\ 2.77 \end{gathered}$ | 40 | $\begin{gathered} h \\ 4.80 \end{gathered}$ | $\begin{array}{r} A \\ 2.77 \end{array}$ | vw | 20 | $\begin{gathered} A \\ 4.80 \end{gathered}$ | $\begin{gathered} A \\ 2.778 \end{gathered}$ | 34 | $\begin{gathered} A \\ 4.815 \end{gathered}$ |
| 200 | 2.39 | 100 | 4.78 | 2.39 | s | 80 | 4.78 | 2.405 | 100 | 4.810 |
| 220 | 1.69 | 63 | 4.78 | 1.69 | vs | 100 | 4.78 | 1.701 | 45 | 4.811 |
| 311 | 1.448 | 20 | 4.802 | 1.443 | w | 40 | 4.786 | 1.451 | 10 | 4.812 |
| 222 | 1.385 | 20 | 4.798 | 1.381 | w | 40 | 4.784 | 1.390 | 5 | 4.815 |
| 400 | 1.202 | 10 | 4.808 | 1.197 | vw | 20 | 4.788 | 1.203 | 4 | 4.812 |
| 331 | 1.102 | 7 | 4.804 | 1.099 | vw | 20 | 4.790 | 1. 1036 | 4 | 4.8105 |
| 420 | 1.073 | 27 | 4.799 | 1.072 | m | 60 | 4.794 | 1.0755 | 9 | 4.8098 |
| 422 | 0.981 | 13 | 4.806 | 0.9794 | m | 70 | 4.798 | 0.9819 | 9 | 4.8103 |
| 511 | . 924 | 3 | 4.810 | . 9240 | vw | 20 | 4.810 | . 9258 | 3 | 4.8106 |
| 440 | . 849 | 3 | 4.803 | -..--.....- |  |  | -.-.-. | . 8504 | 4 | 4.8106 |
| 531 | . 812 | 3 | 4.804 | --------- |  |  | ----- | . 8131 | 5 | 4.8104 |
| 600 | . 802 | 6 | 4.812 | ---1-0-0-1 |  |  | - | . 8018 | 6 | 4.8108 |
| 620 | . 761 | 2 | 4.813 |  |  |  |  |  |  |  |
| 533 | . 732 | 1 | 4.800 | -...... | --- | --- | -.... | - | --. |  |
| 622 | . 724 | 2 | 4.802 | ---------- |  | -- | ---- | --- | --. | --..- |
| 444 | -- |  | ------- | ------ |  |  | - |  | --. | --- |
| 711 | 0.672 | 1 | 4.799 |  |  |  |  |  |  |  |
| 640 | . 667 | 1 | 4.810 | ---------- |  | -- | --- |  |  |  |
| Average unit cell for last fivelines.---------------------- |  |  |  |  |  |  |  |  |  |  |
|  |  |  | 4.805 | -----.---- |  |  | 4.796 | --- | ----- | 4.8105 |

${ }^{\text {c }}$ Published.
d ASTM card.
used. The remaining patterns were converted from $k X$ units. The intensity measurements of Hanawalt, Rinn, and Frevel, and of the NBS show the three strongest lines as 200, 220, and 111 in the order given.

Calcium oxide has a face-centered cubic lattice, and the space group $\mathrm{O}_{\mathrm{h}}^{5}(\mathrm{Fm} 3 \mathrm{~m})$ [61]. There are four molecules in the unit cell. Several lattice constants are listed below for comparison.

Unit cell in angstroms

|  | United Steel, England. | 4.8082 |
| :---: | :---: | :---: |
| 1942 | Huber and Wagener [97] | 4.811 |
| 1953 | Swanson and Tatge ( $27^{\circ} \mathrm{C}$ ) | 4.8105 |

The density calculated from the NBS lattice constant is 3.345 at $27^{\circ} \mathrm{C}$. The index of refraction was not determined on the NBS sample; it is given as $n_{D}=1.837$ by Winchell [250].

### 2.22. Titanium Dioxide (Rutile), $\mathrm{TiO}_{2}$ (Tetragonal)

The three patterns of the ASTM diffraction pattern file (see table 1) are compared in table 23 with an earlier pattern found in the literature, Vegard [234], and with one prepared at the NBS. One of the ASTM patterns lists five references as sources, only three of which were published; of these only two (by Kerr, and by Weiser and Milligan) appear in table 23. The third, ascribed to Boldyrev [19] (who compiled it from a Russian published source [133]), was made from a nattural mineral from the Ural mountains and, possibly because of impurities, is so unlike the other patterns in the table that it was not included.

Material for the NBS pattern was obtained from the National Lead Company, Sample No. MP 559. Spectrographic analysis at the NBS shows no impurity greater than 0.001 percent.

The sample, chiefly anatase, was heated for two ently, in angstroms. The three strongest hours at $1,000^{\circ} \mathrm{C}$ and cooled slowly to obtain lines are shown by the Vegard and the Hanawalt, the rutile phase.

The interplanar spacings of all patterns were converted from $k X$ to angstrom units except those of the Vegard pattern, which were calculated in angstroms from the published Bragg angle data, and the pattern of the United Steel Companies, which is given, appar-

Rinn, and Frevel patterns as 211, 110, and 101; the Kerr, the Weiser and Milligan, and the NBS patterns show them to be 110,211 , and 101 .

Rutile, which belongs to the tetragonal system, has a space group determined as $D_{4}^{14}$ ( $\mathrm{P} 4 / \mathrm{mnm}$ ) by Huggins [98]. Recent unit cell determinations, converted from $k X$ to angstrom

Table 23. Titanium dioxide (rutile), $\mathrm{TiO}_{2}$

| $h k l$ | $1926$ <br> Vegard <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | $1932$ <br> Kerr |  | $\begin{gathered} 1934 \\ \text { Weiser and } \\ \text { Milligan } \end{gathered}$ |  | 1938 <br> Hanawalt, Rinn, and Frevel Mo, 0.7093 A |  | United Steel |  | 1953 <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I | a | I | 1 | I | d | I | a | I | a | I |
|  | A |  | A |  | A |  | A |  | A |  | A |  |
| 110 | 3.292 | 50 | 3.28 | 100 | 3.25 | 100 | 3.25 | 80 | 3.25 | 85 | 3.245 | 100 |
| 101 | 2.510 | 30 | 2.51 | 50 | 2.48 | 90 | 2.49 | 60 | 2.49 | 70 | 2.489 | 41 |
| 200 | 2.327 | 5 | 2.32 | 5 | 2.29 | 10 | 2.29 | 4 | 2.30 | 50 | 2.297 | 7 |
| 111 | 2.212 | 20 | 2.269 | 10 | 2.18 | 40 | 2.19 | 30 | 2.19 | 60 | 2.188 | 22 |
| 210 | 2.046 | 10 | 2.179 | 30 | 2.04 | 20 | 2.05 | 12 | 2.05 | 50 | 2.054 | 9 |
| 211 | 1.708 | 100 | 1.703 | 100 | 1.688 | 100 | 1.69 | 100 | 1.69 | 100 | 1.687 | 50 |
| 220 | 1.649 | 40 | 1.643 | 30 | 1.620 | 30 | 1.62 | 30 | 1.62 | 70 | 1.624 | 16 |
| 002 | 1.499 | 10 | 1.503 | 20 | 1.481 | 20 | 1. 487 | 20 | 1.48 | 60 | 1.480 | 8 |
| 310 | 1.472 | 10 | 1.473 | 20 | 1.450 | 20 | 1.451 | 20 | 1.45 | 60 | 1.453 | 6 |
| 301 | \} 1.377 | 70 | 1.368 | 60 | 1.353 | 80 | 1.357 | 30 | $\{1.36$ | 85 | 1.360 | 16 |
| 112 |  |  |  |  |  |  |  |  | 11.35 | 70 | 1.347 | 7 |
| 311 |  |  |  |  |  |  | ------- |  | 1.30 | 20 | 1.305 | 1 |
| 202 | ------- | ---- | 1.262 | 5 | 1.242 | 10 | 1.247 | 4 | 1.24 | 30 | 1.243 | 3 |
| 212 | ------- |  | ------- |  | ------- | ------- | ------- | ---- | 1.20 | 20 | 1.200 | 1 |
| 321 | 1.184 | 10 | 1.181 | 10 | 1.169 | 10 | 1.172 | 8 | 1.17 | 60 | 1.1700 | 4 |
| 400 | 1.164 | 10 | 1.162 | 5 | 1.146 | 10 | 1.149 | 4 | 1.15 | 50 | 1.1485 | 4 |
| 410 | 1. 107 | 25 | ----- | ------ | ------ | - | ------- | ------- | 1.11 | 20 | 1.1329 | 1 |
| 222 | 1.093 | 10 | 1.097 | 20 | 1.094 | 10 | 1.093 | 8 | 1.09 | 70 | 1.0933 | 4 |
| 330 | ------- |  |  |  |  |  | -- | ------- | 1.08 | 60 | 1.0827 | 4 |
| 411 | 1.050 | 20 | 1.048 | 20 | 1.039 | 10 | 1.042 | 8 | 1.04 | 60 | 1.0424 | 5 |
| 312 | 1.038 | 5 | -------- | ---- | -------- | -------- | ------- | ----- | 1.04 | 60 | 1.0361 | 4 |
| 420 | ------- |  | ------- | ----- | ------- | ------- | - |  | 1.03 | 60 | 1.0273 | 3 |
| 421 |  |  |  |  |  |  |  |  | 0.970 | 30 | -------- |  |
| 322 103 | \}0.976 | 10 | 0.975 | 5 |  |  | 0.966 | 4 | . 964 | 70 | 0.9642 | 2 |
| 402 | . 920 | 10 | . 912 | 5 | ------- |  | ---... | 2 | . 907 | 70 | . 9071 | 3 |
| 510 | . 913 | 10 | . 902 | 5 | ------- |  | 0.905 | 2 | . 900 | 70 | . 9007 | 3 |
| 213 | . 897 | 40 | ------- |  | ------- |  | . 892 | 8 | ---.-.- | --- | . 8892 | 5 |
| $431$ |  |  |  |  |  |  |  |  |  |  |  | 6 |
| 332 | ) .884 | 50 | 0.882 | 5 |  |  | . 877 | 4 |  |  | $\left\{\begin{array}{l}.8739\end{array}\right.$ | 5 |
| 422 | \}. 851 | 10 | . 852 | 5 |  |  | . 845 | 2 | ------- | - | . 8437 | 5 |
| 223 | ). 837 |  |  |  | --...-- |  |  |  | ------- |  |  |  |
| 303 521 | .837 .826 | 30 50 | ------- | 5 | ---------- | ----- | . 834 | 4 | ---------- | -..- | . 8290 | 5 8 |
| 432 |  |  | . 779 | 3 |  |  | ---------- |  |  |  |  |  |
|  |  |  | . 748 | 3 |  |  |  |  |  | ----- | ------- |  |

units, are compared in the following table with those of the NBS.

Unit cell in angstroms

|  |  | $a$ | c |
| :---: | :---: | :---: | :---: |
| 1942 | Schossberger [203]. | 4.598 | 2.960 |
| 1946 | Frevel, Rinn, and Anderson [75].- | 4.59 | 2.96 |
| ---- | United Steel | 4.5928 | 2.9582 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 4.594 | 2.958 |

In accordance with the NBS lattice constant the density is 4.250 at $26^{\circ} \mathrm{C}$. The indices of refraction are very high; Schröder [204] measured them at $25^{\circ} \mathrm{C}$ as $\epsilon_{D}=2.8893$ and $\omega_{D}=2.6124$.

### 2.23. Titanium Dioxide (Anatase), $\mathrm{TiO}_{2}$ (Tetragonal) The first of the four ASTM cards listed in table 1 has no X-ray data on it except a

Table 24. Titanium dioxide (anatase), $\mathrm{TiO}_{2}$

| $h k l$ | $\begin{gathered} 1926 \\ \text { Vegard } \\ \mathrm{Cu}, 1.5405 \mathrm{~A} \end{gathered}$ |  | $1934$ <br> Weiser and Milligan Mo, 0.7093 A |  | 1938 <br> Hanawalt, Rinn, and Frevel <br> Mo, 0.7093 A |  | United Steel |  | 1953 <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ}-27^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I | 2 | I | d | I | d | I | a | I |
| 101 | A 3.58 | 100 | $\begin{gathered} A \\ 3.50 \end{gathered}$ | 100 | $\begin{gathered} A \\ 3.53 \end{gathered}$ | 100 | $\begin{gathered} A \\ 3.515 \end{gathered}$ | 100 | A 3.51 | 100 |
| 103 |  |  |  |  |  | -- | 2.430 | 10 | 2.435 | 9 |
| 004 | 2.401 | 35 | 2.38 | 50 | 2.38 | 24 | 2.377 | 50 | 2.379 | 22 |
| 112 |  | -------- |  |  |  | ------ | 2.338 | 10 | 2.336 | 9 |
| 200 | 1.911 | 70 | 1. 887 | 80 | 1.88 | 40 | 1.891 | 90 | 1.891 | 33 |
| 105 | 1.717 | 30 | 1.698 | 60 | 1.70 | 28 | 1.699 | 70 | 1.699 | 21 |
| 211 | 1.681 | 50 | 1.658 | 60 | 1.66 | 24 | 1.665 | 70 | 1.665 | 19 |
| 213 | - | - | - | ------- | -- | ------ | - | --- | 1.494 | 4 |
| 204 | 1.499 | 60 | 1.480 | 50 | 1.483 | 24 | 1.450 | 70 | 1.480 | 13 |
| 116 | 1.382 | 20 | 1.361 | 20 | 1.365 | 8 | 1.364 | 60 | 1.367 | 5 |
| 220 | 1.351 | 30 | 1.337 | 20 | 1.338 | 8 | 1.338 | 60 | 1.337 | 5 |
| 215 | 1.275 | 50 | 1.267 | 40 | 1.265 | 11 | 1.264 | 70 | 1.264 | 10 |
| 301 |  |  |  | ------- | --------- | ------- | 1.250 | 20 | 1.250 | 3 |
| 303 |  |  |  | ------- | - | ------- | 1.165 | 60 | 1.171 | 2 |
| 312 | 1.173 | 50 | 1.163 | 30 | 1.166 | 6 | 1.160 | 10 | 1.1609 | 3 |
| 118 | ---------- | ------- | --------- | ------- | --------- |  | 1.056 | 10 | 1.0598 | 1 |
| 217 | --------- | ------- | --------- | ------- | --------- | ------- | 1.0509 | 50 | 1.0510 | 1 |
| 321 | 1.055 | 30 | 1.046 | 20 | 1.047 | 3 | 1.0428 | 50 | 1.0433 | 3 |
| 226 | ------- | -- | - | ------- | --------- | ------- | 1.0176 | 50 | 1.0173 | 2 |
| 109 | 1.026 | 10 | 1.017 | 20 | --------- | ------- | 1.0063 | 20 | 1.0065 | 2 |
| 323 | --------- | ------- | - | ------- | - | -- | 0.9959 | 10 | 0.9964 | 1 |
| 316 | 0.961 | 20 | 0.953 | 10 | 0.952 | 2 | . 9547 | 70 | . 9550 | 4 |
| 400 | ------ | ------- |  | ------ | ------- | ------- | . 9456 | 60 | . 9461 | 3 |
| 325 | 0.922 | 35 | 0.915 | 10 | 0.915 | 2 | . 9186 | 70 | . 9189 | 2 |
| 411 |  | - | - |  | -- | ------- | . 9132 | 70 | . 9135 | 1 |
| 219 | 0.902 | 50 | 0.894 | 10 | 0.896 | 2 | -------- | --- | . 8960 | 3 |
| 228 | - | ------- | ------- | ------ | ---------- | ------- | -------- | ----- | . 8894 | 1 |
| 332 | 0.883 | 50 | 0.878 | 10 | --------- | ------- | ---------- | ------ | . 8794 | 2 |
| 318 | . 852 | 40 | . 845 | 10 | ---------- | ------- | ---------- | ----- | . 8464 | 2 |
| 327 | --------- | -------- | --------- | ------- | --------- | ------- | --------- | ---- | . 8311 | 1 |
| 415 | 0.832 | 40 | 0.826 | 10 | --------- | ------- | --------- | ----- | . 8268 | 3 |
| 309 | . 814 | 20 | . 808 | 10 | --------- |  | -------- | ----- | . 8100 | 1 |
| 424 | . 800 | 50 | . 797 | 10 |  |  |  |  | . 7990 | 3 |
|  |  |  | . 742 | 10 | --------- | ------- | -------- | - |  | ---- |
|  | --------- | ------- | . 703 | 10 | -- | ------- | - | ----- | ---- | ------- |
|  |  |  | . 669 | 10 | --------- |  |  |  | -------- |  |

highly inaccurate value for the axial ratio, determined by Vegard in 1916 . The second card is a composite from which only the pattern of Weiser and Milligan appears in table 24 , along with the patterns of the remaining two cards, a pattern by Vegard [234] from the literature, and a pattern made at the NBS.

The NBS pattern was prepared from material supplied by the Research Laboratory of the National Lead Company, South Amboy, N.J., Sample No. MP 559. Spectrographic analysis at the NBS showed no impurity greater than 0.001 percent.

The interplanar spacings of table 24 were all recalculated to angstroms from $k X$ units except those of Vegard, which were calculated directly in angstroms from the Bragg angle data given. All patterns agree that 101 is the strongest line and 200 second strongest. Three or four almost equally intense lines, however, provide variation in the third strongest given in different patterns; this is listed as 004 in the NBS pattern. Two very weak rutile lines appearing in the NBS X-ray diagram are omitted from the pattern given in the table.

The space group of the tetragonal anatase form of titanium dioxide is $\mathrm{D}_{4 \mathrm{~h}}^{19}$ (I4/amd) according to Huggins [98] and Vegard [233]. The unit cell contains four molecules. The lattice constants obtained from the NBS pattern are compared in the table below with those of other workers:

Unit cell, angstroms

|  |  | a | c |
| :---: | :---: | :---: | :---: |
|  | United Steel | 3.783 | 9.509 |
| 1942 | Schossberger [203] | 3.784 | 9.505 |
| 1946 | Frevel, Rinn, and Anderson [75 | 3.76 | 9.45 |
| 1953 | Swanson and Tatge ( $26^{\circ}-27^{\circ} \mathrm{C}$ ) | 3.783 | 9.51 |

The density, calculated from the NBS lattice constant, is 3.899 at $26^{\circ}-27^{\circ} \mathrm{C}$. The indices of refraction could not be obtained from the NBS sample, which was too finely powdered.
2.24. Nickelous Oxide (Bunsenite), NiO (Cubic)

Three cards for nickelous oxide are included in the X-ray diffraction file of the ASTM (see table 1). One of them records no pattern but only a determination of the lattice constant (card number 3-1287). One of the patterns is a composite from four sources of which one is unpublished and not represented in table 25 . The $X$-ray patterns of the two cards, from four previously published sources, are compared in the table with those of three additional workers, Clark, Asbury, and Wick [49], Bravo [34], and Passerini [182], that were obtained from the literature, and with a pattern prepared at the NBS. An electron diffraction pattern by Darbyshire [55] is also included, for comparison.

The sample from which the NBS pattern was made was obtained from Johnson, Matthey \& Co., Ltd., and was numbered 3087. They estimated the purity at 99.99 percent. This was corroborated by spectroscopic analysis at the Bureau, which showed only faint traces of Mg , Si , and Ca .

The Levi and Tachinni, the Bravo, and the Ksanda spacings (table 25) were calculated in angstrom units from Bragg angle data. The spacings of the remaining patterns were assumed given in $k X$ units, and were converted to angstroms. The lines 200, 111, and 220 are the first, second, and third strongest index lines for the NBS and Darbyshire patterns and would be chosen in this order in selecting index lines for the Hanawalt, Rinn, and Frevel pattern, although 111 and 220 have actually the same intensity. Converting the intensity, values of the Hanawalt, Rinn, and Frevel pattern to their equivalents if copper rather than molybdenum radiation had been used, ([1] page 108 of index covering original set of cards or card no. vii of introduction to 1950 file), 111 becomes considerably stronger than 220.

The lattice of nickelous oxide is facecentered cubic [61]. It has a space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ), with four molecules to the unit cell.

In 1948 Rooksby [194] showed that at $18^{\circ} \mathrm{C}$ most of the diffraction lines are doublets or triplets, and interpreted the structural significance as a slight distortion of the gen-
erally accepted cubic lattice. He regards the lattice as rhombohedral, $a=2.9518$, $a=60^{\circ} 4.2^{\prime}$ (for a face-centered cube referred to a primitive rhombohedral lattice $\alpha=60^{\circ}$ ).

Table 25. Nickelous oxide, NiO (cubic)


The doubling of lines could not be detected on the NBS chart. The NBS unit cell determination is compared below with other published values:

## Unit cell, in angstroms



The density, calculated from the NBS value for the lattice constant, is 6.806 at $26^{\circ} \mathrm{C}$ 。 The index of refraction is very high; Ksanda in 1931 [134] gave $n_{L i}=2.73$.

### 2.25. Cupric Oxide (Tenorite), CuO (Monoclinic)

The ASTM file contains six cards for cupric oxide (see table 1). One of these (number 2-1263) contains only lattice constants and structure data. Of the cards containing patterns, one (number 2-1037) is a composite of data from four sources, only one of which is in the literature. Two others contain patterns by Tunell, Posnjak, and Ksanda, one from molybdenum radiation, the other copper, of which the lines were indexed on the basis of single crystal data. Those ASTM patterns appearing in the literature are compared in table 26 with two additional patterns, by Waldo [242], and by Billiet and Vandendriesshe [16], and with a pattern prepared at the NBS. The Harcourt and Waldo data of card number 2-1037 were doubtless communicated to the ASTM before publication elsewhere.

The sample used at the NBS was obtained from Johnson, Matthey \& Co., Ltd, and was numbered 3257. Spectrographic analysis at the NBS showed only faint traces of iron and magnesium as impurities.

All the interplanar spacings of the patterns in table 26 were converted from k $\lambda$ to angstrom units except the first one, by Niggli, accompanied by a wavelength value of 1.541 for copper radiation. The indexing of the lines in table 26 follows that worked out by Tunell, Posnjak, and Ksanda in 1935, differing only where a line due to a group of superimposed reflections is resolved in the NBS pattern.

The three strongest or index lines are, in all except the first pattern, the combined $\overline{1} 11-002,111-200$, and $\overline{2} 02$ lines. The NBS pattern records the three strongest lines as $\overline{1} 11,111$, and 002 . The $\overline{1} 11$ reflection cannot be separated entirely in the NBS powder pattern, from the 002, nor can the intensity of the 111 be measured without the influence of the 200. Integrated measurements showed the total intensity of the $002, \overline{1} 11$ doublet to be about 85 percent of the 111,200 doublet; however, because 002 and $\overline{1} 11$ reflections are closer to each other than are the 200 and 111 , their reinforced intensities are greater when measured as peak height above background. ${ }^{3}$

In 1933 Tunell, Posnjak, and Ksanda [225] assigned tenorite to the monoclinic system, with the space group $\mathrm{C}_{2 \mathrm{~h}}^{6}(\mathrm{C} 2 / \mathrm{c})$. There are four molecules to the unit cell. Converted from $k X$ to angstrom units, the 1935 set of data from Tunell, Posnjak, and Ksanda compare thus with the NBS measurements:

Uhit cell, in angstroms

| 1935 | Tunell, Posnjak, and Ksanda [226] | $\begin{gathered} a \\ 4.662 \end{gathered}$ | $\begin{gathered} b \\ 3.417 \end{gathered}$ | $\begin{gathered} c \\ 5.118 \end{gathered}$ | $\begin{gathered} \beta \\ 99^{\circ} 29^{\prime} \end{gathered}$ |
| :---: | :---: | :---: | :---: | :---: | :---: |
| 1953 | Swanson and <br> Tatge $\left(26^{\circ} \mathrm{C}\right)$ _ | 4.684 | 3.425 | 5.129 | $99^{\circ} 28^{\prime}$ |

The density, based on the NBS lattice constant, is 6.51 at $26^{\circ} \mathrm{C}$.

[^8]Table 26. Cupric oxide, CuO (monoclinic)

2.26. Germanium Dioxide, $\mathrm{GeO}_{2}$ (Hexagonal)

The two patterns for germanium dioxide in the X-ray diffraction file of the ASTM (see table l) are compared in table 27 with a pattern prepared at the NBS. The sample for the NBS pattern was obtained from Johnson, Matthey \& Co., Ltd, and was labelled number 3662. The only impurity indicated in their spectrographic analysis was a faint trace of calcium.

In table 27 the interplanar spacings of all patterns are given in angstrom units.

Table 27. Germanıum dioxide, $\mathrm{GeO}_{2}$ (hexagonal)

| $h k l$ | 1928 |  | 1938 |  | 1953 |  | 1953 |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | Zachariasen |  | Hanawalt, |  | Swanson andTatge |  | hkl | Swanson and Tatge |  |
|  | $\mathrm{Fe}, 1.9360 \mathrm{~A}$ |  | Mo, 0.7093 A |  | $\left\|\begin{array}{c} \mathrm{Cu}, 1.5405 \mathrm{~A} \\ 26^{\circ} \mathrm{C} \end{array}\right\|$ |  |  | $\begin{gathered} \mathrm{Cu}, 1.5405 \mathrm{~A} \\ 26^{\circ} \mathrm{C} \end{gathered}$ |  |
|  |  |  |  |  |  |  |  |  |  |
|  | d | I | d | I | ${ }^{\text {d }}$ | I |  | d | I |
|  | ${ }_{\text {A }}$ |  | ${ }^{\text {A }}$ |  | ${ }^{\text {A }}$ |  |  | ${ }^{\text {A }}$ |  |
| 100 | 4.33 | 20 | 4.32 | 20 | 4.32 | 21 | 005 | 1.1308 | 1 |
| 101 | 3.431 | 100 | 3.42 | 100 | 3.429 | 100 | 312 | 1.1026 | 2 |
| 110 | 2.486 | 15 | 2.49 | 14 | 2.496 | 11 | 105 | 1.0933 | 2 |
| 102 | 2.362 | 40 | 2.35 | 25 | 2.366 | 22 | 214 | 1.0683 | 3 |
| 111 | 2.278 | 15 | 2.28 | 16 | 2.283 | 13 | 401 | 1.0605 | 2 |
| 200 | 2.159 | 40 | 2.15 | 20 | 2.159 | 18 | 223 | 1.0397 | 2 |
| 201 | 2.014 | 5 | 2.00 | 2 | 2.018 | 2 | 115 | 1.0297 | 2 |
| 003 | 1.879 | 15 |  | --- | 1.884 | 8 | 402 |  |  |
| 112 | 1.868 | 30 | 1.87 | 25 | 1.870 | 14 | 304 | 1.0084 | 4 |
| 103 | 1.727 | 15 | --.- | --- | 1.726 | 4 | 321 | 0.9759 | 2 |
| 202 | 1.717 | 15 | 1.71 | 12 | 1.716 | 7 | 006 | . 9419 | 1 |
| 210 | 1.634 | 10 | 1.62 | 2 | 1.633 | 3 | 322 | . 9352 | 1 |
| 211 | 1.564 | 60 | 1.56 | 25 | 1.568 | 13 | 224 | . 9345 | 3 |
| 113 | 1.500 | 20 | 1.498 | 8 | 1.303 | 5 | 411 | . 9294 | 3 |
| 203 | 1.420 | 30 | 1.448 | 4 | 1.420 | 11 | 412 | . 8943 | <1 |
| 212 | 1.413 | 30 | 1.413 | 25 | 1.414 | 13 | 305 | . 8894 | <1 |
| 301 | 1.394 | 30 | 1.389 | 8 | 1.395 | 7 | 403 | 8814 | $<1$ |
| 104 | 1.341 | 30 | 1.342 | 10 | 1.343 | 5 | 500 | . 8636 | <1 |
|  | ----- | --- | 1.304 | 2 |  | -- | 404 | . 8579 | 1 |
| 302 | 1.281 | 25 | 1.279 | 10 | 1.283 | 4 | 501 | 41 | <1 |
| 220 | 1.246 | 5 | 1.256 | 2 | 1.247 | 1 | 330 | . 8315 | 1 |
| 213 |  |  |  |  | 1.234 | 4 | 331 | . 8223 | 1 |
| 114 | ${ }^{1.23}$ | 25 | 1.230 | 6 | 1.231 | 4 | 420 | . 8162 | 1 |
| 221 | 1.218 | 20 |  |  | 1.218 | 2 | 324 | . 8112 | 2 |
| 310 | 1.196 | 40 |  |  | 1.1976 | 4 | 421 | . 8078 | 1 |
| 204 | 1.182 | 5 |  |  |  | -- |  |  | -- |
| 311 | 1.172 | 10 |  |  | 1.1720 | 1 |  |  | -- |
| $\begin{array}{\|l\|} 303 \\ 222 \end{array}$ | \}1.142 | 10 |  |  | 1.1420 | 1 |  |  |  |

One line of the Hanawalt, Rinn and Frevel pattern is extraneous to the postulated structure; it could not be indexed. The intensity measurements of the first lines of the three patterns are closely comparable. For all patterns the first, second, and third strongest lines are the 101,100 , and 110 , respectively.

The lattice of germanium dioxide is hexagonal and was determined by Zachariasen [2.60] as $\mathrm{D}_{3}^{4}\left(\mathrm{C}_{1} 2\right)$, isomorphous with low quartz. There are three molecules in the unit cell. The new file card for the Hanawalt, Rinn, and Frevel pattern is unfortunately mislabelled "Tetragonal." From table 27 the pattern is plainly identical to the hexagonal patterns of Zachariasen and the NBS. The published pattern [85] is unaccompanied by symmetry classification. Converted to angstrom units, the Zachariasen measurements compare thus with those of the NBS pattern:

Unit cell, anestrons

| 1928 | Zachariasen $[260] \ldots$ | $a$ | $c$ |
| :--- | :--- | :---: | :---: |
| 1953 | Swanson and Tatge $\left.\left(26^{\circ} \mathrm{C}\right) \ldots-\ldots-\right)^{2}$ | 4.982 | 5.659 |

The density, using the NBS unit cell dimensions, is 4.280 . The material was very fine-grained, which made optical examination difficult. The double refraction is very weak; the average index of refraction is $n=1.67$.

### 2.27. Arsenic Trioxide, $\mathrm{As}_{2} \mathrm{O}_{3}$ (Cubic)

The ASTM file of diffraction patterns includes five cards for arsenic trioxide (see table 1); one of these (number 3-1234) does not bear a pattern, but records only a lattice constant and a space group determination. Cf the remaining four, two are for synthetic compounds, and two for naturally occurring minerals. One of the latter (2-0530) represents the monoclinic form claudetite, the other (2-0531), like the two artificial forms, represents cubic arsenolite. The natural arsenolite is from Bieber, Hesse, Germany (misspelled "Hasse" on the new file card). The

Table 28. Arsenic trioxide, $\mathrm{As}_{2} \mathrm{O}_{3}$ (cubic)

| $h k l$ | 1928 <br> Passerini <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  |  |  |  | 1931 <br> Lihl <br> $\mathrm{Fe}, 1.9360 \mathrm{~A}$ |  |  | 1938 <br> Mikheev and Dubinina <br> $\mathrm{Fe}, 1.960 \mathrm{~A}$ |  |  | 1938 <br> Hanawalt, Rinn, and Frevel Mo, 0.7093 A |  |  | 1953 <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405,26^{\circ} \mathrm{C}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
|  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
|  | $d^{\text {a }}$ | $d^{\text {b }}$ | $I^{\text {c }}$ | $I^{\text {d }}$ | $a$ | a | I | $a$ | d | I | $a$ | d | I | $a$ | a | I | $a$ |
|  | A | A |  |  | A | A |  | A | A |  | A | A |  | A | A |  | A |
| 111 | 5.975 | 6.39 | w | 40 | 10.34 |  |  |  |  | --- |  | 6.3 | 56 | 10.9 | 6.39 | 63 | 11.07 |
| 220 | 3.754 | 3.92 | ms | 70 | 10.60 |  | --- |  |  | --- |  |  | ----- | ----- | 3.92 | $<1$ | 11.09 |
| 222 | 3.111 | 3.20 | s | 80 | 10.78 | 3.191 | $s$ | 11.05 | 3.195 | 100 | 11.07 | 3.19 | 100 | 11.1 | 3.195 | 100 | 11.07 |
|  |  |  |  | -- |  |  | --. |  | 2.944 | 10 | ----- | ----- | --- |  |  | --- |  |
| 400 | 2.725 | 2.77 | m | 60 | 10.88 | 2.764 | s | 11.06 | 2.764 | 60 | 11.06 | 2.76 | 24 | 11.0 | 2.768 | 28 | 11.07 |
| 331 | 2.500 | 2.54 | m | 60 | 10.92 | 2.535 | s | 11.05 | 2.539 | 90 | 11.07 | 2.54 | 32 | 11.1 | 2.541 | 38 | 11.08 |
|  | 2.332 | 2.36 | vw | 20 | ----- |  | --- |  |  | --- |  | ----- | ----- | ----- | ------ | - | ----- |
| 422 | 2.235 | 2.26 | vw | 20 | 10.93 | 2.257 | w | 11.10 | 2.255 | 50 | 11.05 | 2.24 | 8 | 11.1 | 2.262 | 12 | 11.08 |
| 511 | 2.112 | 2.13 | w | 30 | 10.97 | 2.126 | w | 11.05 | 2.127 | 60 | 11.05 | 2.12 | 16 | 11.0 | 2.132 | 17 | 11.08 |
| 440 | 1.937 | 1.96 | ms | 70 | 10.96 | 1.971 | s | 11.15 | 1.955 | 90 | 11.06 | 1.95 | 24 | 11.0 | 1.958 | 27 | 11.08 |
| 531 | ----- | ---.- | --- | --- | ----- | ------ | --- |  | 1.875 | 40 | 11.09 | ----- | ----- | ----- | 1.873 | 6 | 11.08 |
| 442 | 1.837 | 1.85 | mw | 50 | 11.02 |  | --- |  | 1.841 | 50 | 11.05 |  | ----- | ----- | 1.846 | 5 | 11.08 |
|  | ----- |  | --- | --- | --... | ---.... | --- |  | 1.810 | 10 | ----- | ----- | -- | ----- |  |  |  |
| 622 | 1.661 | 1.67 | ms | 70 | 11.00 | 1.666 | s | 11.05 | 1.668 | 90 | 11.06 | 1.66 | 16 | 11.0 | 1.670 | 21 | 11.08 |
| 444 | 1.589 | 1.60 | mw | 50 | 11.00 | ------ | --- |  | 1.596 | 60 | 11.06 | 1.59 | 8 | 11.0 | 1.599 | 10 | 11.08 |
| 711 | 1.541 | 1.55 | ms | 70 | 11.01 | 1. 547 | --- | 11.05 | 1.550 | 90 | 11.07 | 1.54 | 16 | 11.0 | 1.551 | 22 | 11.08 |
| 642 | ----- | ----- | --- | --- | ----- | ------ | --- | ------ | 1.480 | 40 | 11.08 | ----- | ----- | ----- | 1.480 | 2 | 11.08 |
| 731 | 1.434 | 1.44 | ms | 70 | 11.02 | 1.439 | s | 11.05 | 1.442 | 90 | 11.08 | 1.441 | 8 | 11.07 | 1.442 | 12 | 11.08 |
| 800 |  |  | --- | --- | --- | ------ | --- |  | 1.383 | 10 | 11.06 |  |  |  | 1.385 | 3 | 11.08 |
| 733 | ----* | ----- | --- | --- | ----- | 1.350 | $s$ | 11.05 | 1.353 | 90 | 11.07 | --..-- | ---.- | ----- | 1.353 | 10 | 11.07 |
| 644 | 1.342 | 1.342 | ms | 70 | 11.05 | ------ | --- | ------ | ----- | --- | ----- | 1.346 | 8 | 11.10 | 1.343 | 1 | 11.07 |
| 822 | 1.300 | 1.300 | w | 40 | 11.04 | 1.302 | w | 11.04 | 1.305 | 80 | 11.07 | 1.305 | 8 | 11.06 | 1.305 | 5 | 11.07 |
| 751 |  |  | --- | --- | ----- | 1.278 | w | 11.07 | 1.277 | 60 | 11.06 |  | ----- |  | 1.278 | 3 | 11.07 |
| 662 | 1.269 | 1.269 | w | 40 | 11.06 |  | --- |  | 1.270 | 50 | 11.07 | 1.269 | 8 | 11.06 | 1.271 | 1 | 11.08 |
| 840 | ----- | ----- | --- | --- | ----- | ------ | --- |  | 1.238 | 40 | 11.07 | ----- | ----- | ----- | 1.238 | 2 | 11.07 |
| 911 | ----- | ----- | --- | --- | ----- | ------ | --- | ------ | 1.213 | 60 | 11.05 |  | ----- |  | 1.216 | 5 | 11.08 |
| 842 | 1.206 | 1.206 | ms | 70 | 11.06 | 1.2065 | m | 11.058 | 1.208 | 60 | 11.07 | 1.207 | 8 | 11.06 | 1.208 | 6 | 11.07 |
| 664 |  |  | -.- | --- |  | 1.1781 | w | 11.052 |  | --- |  |  |  |  | 1.1812 | 1 | 11.081 |
| 931 | 1.161 | 1.161 | w | 40 | 11.08 | 1.1592 | m | 11.058 | 1.161 | 50 | 11.08 |  |  |  | 1.1610 | 3 | 11.075 |
| 933 |  |  | -- | --- | ----- |  | --- | ------ | 1.113 | 25 | 11.07 |  |  |  | 1.1132 | 2 | 11.076 |
|  | 1.108 | 1.108 | vvw | 10 | ----- |  | --- |  | ----- | --- | ----- |  |  |  | ------ | -- | ------ |
| 10.2 .0 |  | ----- | --- | --- | ----- |  | --- |  | 1.086 | 13 | 11.08 |  | ----- | ----- | 1.0859 | 1 | 11.074 |
| 951 | 1.068 | 1.068 | vs | 100 | 11.05 | 1.0685 | s | 11.053 | 1.070 | 100 | 11.07 | 1.066 | 8 | 11.03 | 1.0706 | 6 | 11.074 |
| 864 | 1.030 | 1.030 | vw | 20 | 11.10 | 1.0266 | w | 11.057 | 1.029 | 25 | 11.08 | ----- | ----- |  | 1.0294 | 2 | 11.087 |
| $10 \cdot 4 \cdot 2$ |  | ----- | --- | --- | ----- | ------ | --- | ------ | ----- | --- | ----- |  | ----- |  | 1.0117 | $<1$ | 11.083 |
| $11 \cdot 1 \cdot 1$ | ----- | ----- | --- | --- | ----- | 0.9970 | m | 11.057 | ----- | --- | ----- |  | ----- | ----- | 0.9976 | 2 | 11.064 |
| $11 \cdot 3 \cdot 1$ |  | ----- | --- | -- | --- |  |  |  |  |  |  |  |  | --- | . 9676 | 4 | 11.075 |
| $10.4 \cdot 4$ | 0.965 | 0.965 | vs | 100 | 11.08 |  |  |  |  | - |  |  |  |  | . 9643 | 2 | 11.079 |
| 10.6 .0 | . 950 | . 950 | vw | 20 | 11.07 | ---- |  |  |  | --- |  |  |  |  | . 9496 | 1 | 11.074 |
| $11 \cdot 3 \cdot 3$ | . 937 | . 937 | vw | 20 | 11.08 |  | --- |  |  | --- |  |  |  |  | . 9392 | 2 | 11.073 |
| $11 \cdot 5 \cdot 1$ | . 915 | . 915 | vvw | 10 | 11.09 |  | --- | ------ | ----- | --- | ----- |  | ----- | ----- | . 9135 | 1 | 11.075 |
| $12 \cdot 2 \cdot 2$ | . 898 | . 898 | $s$ | 80 | 11.08 |  |  |  |  |  |  |  |  |  | . 6982 | 3 | 11.073 |
| $12 \cdot 4 \cdot 2$ | . 865 | . 865 | ms | 70 | 11.08 |  |  |  |  |  |  |  |  |  | . 8648 | 2 | 11.074 |
| $13 \cdot 1 \cdot 1$ | . 846 | . 846 | ms | 70 | 11.08 | ------ |  |  |  | --- |  |  |  | ----- | . 8469 | 3 | 11.075 |
| $13 \cdot 3 \cdot 1$ | -.-.-- | -- | --- | --- | ----- | ------ | --- | ------ | ----- | --- | ----- | ----- | ----- | ----- | . 8278 | 3 | 11.075 |
| $13 \cdot 3 \cdot 3$ |  |  |  |  |  |  | --- |  |  | --- |  |  |  |  | . 8098 | 2 | 11.074 |
| Average unit cell for last five lines. |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
|  |  |  |  |  | 11.08 |  | --- | 11.055 | ----- | --- | 11.08 |  |  | 11.07 |  |  | 11.074 |

[^9]
## card.

claudetite pattern is of little value to the file inasmuch as its locality is not given and it contains a large number of arsenolite lines. The reference "RI" appearing on the ASTM card for the two patterns of naturally occurring minerals indicates a compilation of Boldyrev [19] which gives the original published source [153] of the patterns. In table 28 the natural arsenolite pattern is listed along with the remaining two ASTM patterns, a fourth from the literature, Lihle [140], and one by the NBS.

The NBS pattern was made from a sample obtained from the Mallinckrodt Chemical Works, and numbered 906487. Their spectrographic analysis indicated the following impurities in amouncs of 0.001 to 0.01 percent: $\mathrm{Ca}, \mathrm{Fe}$, $\mathrm{Mg}, \mathrm{Pb}, \mathrm{Sb}$, and Si . The material was recrystallized by sublimation before using.

The Passerini and Lihl spacings were calculated for table 28 in angstroms from published Bragg angle data; the Mikheev and Dubinina, and the Hanawalt, Rinn, and Frevel spacings were converted from $k X$ units. On the ASTM card the first fourteen spacings of the Passerini pattern are not those originally published, but were recalculated on the basis of a lattice constant determined from the last or high angle lines of the pattern. The original intensity measurements were converted to numerical designations for the ASTM card. Both the published and the ASTM patterns are given in table 28. The Passerini pattern and that of Mikheev and Dubinina both include lines not permitted by the postulated $\mathrm{O}_{\mathrm{h}}^{7}$ space group. The intensity values of the Hanawalt, Rinn, and Frevel and the NBS patterns agree fairly closely. Both show 222, 111, and 331 as the first, second, and third strongest lines.

Arsenic trioxide was determined by Bozorth in 1923 [24] as having the diamond structure on the basis of line spectra from 100,110 , and 111 , and Laue photographs. Eight $\mathrm{As}_{4} \mathrm{O}_{6}$ units are tetrahedrally arranged in a unit cell having the space group $\mathrm{O}_{\mathrm{h}}^{7}$ (Fd3m). Two recent measurements of the lattice constant are compared below with that of the NBS. All
are in angstroms at $25^{\circ} \mathrm{C}$, converted by means of the coefficient of expansion $37.0 \times 10^{-6}$ [217].
lint cell in angstroms at $25^{\circ} \mathrm{C}$

| 1936 | Straumanis and Ieviņš [216] | 11.0724 |
| :---: | :---: | :---: |
| 1939 | Straumanis, Ievinss, and Karlsons [217]. | 11.07441 |
| 1953 | Swanson and Ta | 11.074 |

The density, based on the NBS lattice constant, is 3.8654 at $25^{\circ} \mathrm{C}$ 。 The index of refraction determined for the NBS sample is $n=1.748$.

### 2.28. Selenium Dioxide, $\mathrm{SeO}_{2}$ (Tetragonal)

No patterns were found for selenium dioxide (selenolite) in the ASTM file or in the literature. The one given in table 29 was prepared at the NBS from specially purified material supplied by the Mallinckrodt Chemical Works. Spectrographic analysis at the NBS showed no impurities greater than 0.001 percent. The material is very hygroscopic and, although the sample was mixed with petrolatum, a few weak lines from the monohydrate appeared in the diagram which were omitted from the pattern in table 29.

The lines of the pattern are indexed in accordance with the structure and unit-cell dimensions determined by McCullough [145] in 1937. Although crystals of $\mathrm{SeO}_{2}$ generally have been described as monoclinic (Waitkins and Clark [24l]), the NBS pattern agrees with the structure determination of McCullough, showing a tetragonal structure. McCullough gives the probable space group as $\mathrm{D}_{4 \mathrm{~h}}^{13}$ (P4/mbc), or $\mathrm{C}_{4 \mathrm{r}}^{8}$ (C4cb), with eight molecules in the unit cell. The lattice constants derived by McCullough, which were used by Frevel, Rinn, and Anderson [75] in 1946, compare with the NBS determinations as follows, after conversion to angstrom units:

Unit cell, in angstroms

|  |  | $a$ | $c$ |
| :---: | :---: | :---: | :---: |
| 1937 | McCullough [145] | 8.370 | 5.061 |
| 1951 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 8.35 | 5.08 |

The density is 4.16 at $26^{\circ} \mathrm{C}$, based on the NBS lattice constant. The material proved too unstable for a determination of the indices of refraction.

Table 29. Selenium dıoxide, $\mathrm{SeO}_{2}$ (tetragonal)

| $h k l$ | $1951$ <br> Swanson and Tatge $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  | $h k l$ | $1951$ <br> Swanson and Tatge $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I |  | d | I |
|  | A |  |  |  |  |
| 110 | 5. 92 | 13 | 322 | 1.711 | 25 |
| 200 | 4.17 | 85 | 500 | 1.673 | 9 |
| 210 | 3.73 | 100 | 510 | 1.640 | 9 |
| 201 | 3.227 | 11 | 431 | 1.588 | 3 |
| 211 | 3.009 | 88 | 511 | 1.559 | 14 |
| 220 | 2.998 | 38 | 332 | 1.556 | 10 |
| 300 | 2.789 | 2 | 213 | 1.538 | 4 |
| 310 | 2.640 | 14 | 422 | 1.503 | 3 |
| 002 | 2.533 | 2 | 440 | 1.478 | 4 |
| 311 | 2.343 | 3 | 530 | 1.437 | 4 |
| 112 | 2.320 | 15 | 313 | 1.421 | 5 |
| 202 | 2.252 | 14 | 432 | 1.394 | 8 |
| 321 | 2.105 | 6 | 512 | 1.379 | 20 |
| 400 | 2.090 | 14 | 522 | ) 1.324 | 9 |
| 330 | 1.973 | 10 | 620 | \} 1.324 |  |
| 401 | 1.933 | 17 | 540 | 1.305 | 4 |
| 411 | 1.895 | 14 | 413 | 1.292 | 3 |
| 420 | 1.871 | 14 | 621 | 1.278 | 12 |
| 312 | 1.831 | 17 | 004 | 1.264 | 15 |
| 421 | 1.755 | 13 | 612 | 1.209 | 13 |

### 2.29. Stannic Oxide, $\mathrm{SnO}_{2}$ (Tetragonal)

In addition to the seven patterns for stannic oxide (cassiterite) appearing on ASTM cards (see table 1), an eighth was found in the literature, Natta and Passerini [158]. These are compared with an NBS pattern for which a sample of tin oxide was obtained from Johnson, Matthey \& Co., Ltd; the sample was
numbered 2763. The report on the spectrographic examination which accompanied the sample shows no lines for impurities stronger than faintly visible.

The interplanar spacings recorded in table 30 were all converted to angstrom units except those of the United Steel Companies pattern, which were evidently calculated in angstroms originally, and of the Natta and Passerini pattern, which were calculated directly in angstroms from the sine $\theta$ data published.

Several patterns list the 211 as the strongest liné, or at least equal in strength to the 110 and 101. The Hanawalt, Rinn, and Frevel, and the NBS patterns show the 110 strongest. In listing the first, second, and third strongest lines, four of the eight patterns, including those of Hanawalt, Rinn, and Frevel, and the NBS, would record them in the following order: 110, 101, 211.

Vegard [232] in 1916 recorded the space group determination of $\mathrm{D}_{4 \mathrm{~h}}^{14}$ for tetragonal stannic oxide. There are two molecules in the unit cell. The lattice constants derived from the NBS pattern compare as follows with earlier determinations:

## Unit cell, in angstroms

|  |  | $a$ | c |
| :---: | :---: | :---: | :---: |
| 1924 | Davey [58] | 4.728 | 3.167 |
| 1932 | Bragg and Darbyshire [33] | 4.73 | 3.18 |
| ---- | United Steel. | 4.7355 | 3.1850 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 4.738 | 3.188 |

The density calculated from the NBS dimensions of the unit cell is 6.995 .

Table 30. Stannic oxide $\mathrm{SnO}_{2}$ (tetragonal)

| $h k l$ | 1929 |  | 1932 |  | 1938 |  | 1938 |  | 1942 |  |  |  |  |  | --- |  | 1953 |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | Natta and Passerini |  | Weiser and Milligan |  | Boldyrev |  | Hanawalt, Rinn, and Frevel |  | Harcourt |  | Harcourt; Roldyrev |  | British Museum |  | United Steel |  | Swanson and Tatge |  |
|  | Fe, 1.9360 A |  | Mo, 0.7093 A |  | Mo, 0.7093 A |  | Mo, 0.7093 A |  | $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | Mo, 0.7093 A |  | $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | Co, $1.7902^{\text {a }} \mathrm{A}$ |  | $\begin{gathered} \mathrm{Cu}, \quad 1.5405 \mathrm{~A}, \\ 26^{\circ} \mathrm{C} \end{gathered}$ |  |
|  | d | $I$ | d | $I$ | d | $I$ | d | 1 | d | I | d | $I$ | d | I | d | $I$ | d | $I$ |
| 110 | A 3. 27 | mw | $\begin{gathered} A \\ 3.41 \end{gathered}$ | 100 | $\begin{gathered} A \\ 3.337 \end{gathered}$ | 50 | $\begin{gathered} A \\ \text { 3. } 35 \end{gathered}$ | 100 | A | 100 | A 3.34 | 80 | A 3.31 | 80 | A 3.36 | 80 | $\begin{gathered} A \\ \text { 3. } 351 \end{gathered}$ | 100 |
|  |  | ---- |  | --- |  | ---- |  | ---- |  | ---- | ----- | ---- | 2.92 | 20 |  | ----- |  | --- |
| 101 | 2.606 | $s$ | 2.68 | 100 | 2.636 | 50 | 2.65 | 63 | 2.63 | 100 | 2.64 | 80 | 2.63 | 80 | 2.65 | 80 | 2.644 | 81 |
| 200 |  | ---- | 2.35 | 50 | 2. 367 | 40 | 2. 36 | 18 | 2.35 | 33 | 2. 36 | 60 | 2.35 | 60 | 2. 37 | 50 | 2. 369 | 24 |
| 111 |  | ---- |  | --- | ----- | --- | -- | ---- | 2. 28 | 8 | 2. 28 | 30 | ---- | -- | 2.31 | 20 | 2. 309 | 5 |
| 210 | ----- | ---- | ------ | --- | ----- | ---- | ----- | ---- | 2.11 | 5 | 2.11 | 20 | 1.95 | 50 | ----- | ----- | 2. 120 | 2 |
| 211 | 1.754 | vs | 1.77 | 100 | 1.764 | 100 | 1.75 | 63 | 1.75 | 100 | 1.75 | 100 | 1.75 | 100 | 1.76 | 100 | 1.765 | 63 |
| 220 | ----- | ---- |  | --- | 1.675 | 70 | 1.67 | 10 | 1.668 | 33 | 1.67 | 70 | 1.67 | 60 | 1.67 | 60 | 1.675 | 63 |
| 002 | 1. 583 | mw | ------ | --- | 1. 590 | 20 | 1. 58 | 5 | 1. 58 | 17 | 1.58 | 50 | 1. 59 | 50 | 1. 59 | 50 | 1. 593 | 8 |
|  |  | ---- |  | --- | 1. 529 | 10 | ----- | - | ----- | -- | ----- | -- | 1. 57 | 40 | ----- | ----- |  | --- |
| 310 | 1. 491 | ---- |  | --- | 1. 500 | 80 | 1. 495 | 10 | 1.49 | 33 | 1. 49 | 70 | 1. 50 | 60 | 1. 50 | 60 | 1. 498 | 13 |
| 112 | 1.430 | s | 1. 43 | 70 | 1. 438 | 70 | 1. 438 | 10 | 1.43 | 33 | 1. 43 | 70 | 1. 44 | 60 | 1. 44 | 60 | 1. 439 | 17 |
| 301 | 1. 409 | $s$ |  | --- | 1. 413 | 70 | 1. 415 | 15 | 1. 408 | 33 | 1.411 | 70 | 1. 41 | 60 | 1.41 | 60 | 1. 415 | 15 |
|  |  | ---- | ------ | --- | ----- | ---- | ----- | ---- | ----- | ---- | ----- | --- | 1.35 | 20 | ----- | ----- | ------ | --- |
| 202 | ----- | ---- | 1. 32 | 20 | 1. 323 | 40 | 1. 318 | 6 | 1.318 | 17 | 1.321 | 50 | 1. 32 | 40 | 1. 32 | 50 | 1. 322 | 7 |
| 321 | 1. 211 | vs | 1. 21 | 70 | 1. 216 | 80 | 1.215 | 10 | 1.211 | 33 | 1.213 | 80 | 1. 22 | 70 | 1.21 | 60 | 1.215 | 11 |
| 400 |  | ---- |  | --- | 1. 185 | 30 | 1. 182 | 2 | 1. 181 | 8 | 1. 183 | 40 | 1. 19 | 20 | 1. 18 | 40 | 1. 184 | 3 |
| 222 | 1. 150 | s | 1. 16 | 20 | 1. 155 | 70 | 1. 152 | 6 | 1. 151 | 17 | 1. 153 | 60 | 1. 16 | 60 | 1. 15 | 60 | 1. 155 | 8 |
|  |  | --- |  | --- |  | - | ----- | ---- | ----- | ---- | 1. 141 | 30 | ----- | --- | ----- | ----- | ------ | --- |
| 330 | 1. 114 | ms |  | --- | 1. 118 | 60 | 1. 112 | 3 | 1.113 | 17 | 1. 115 | 60 | 1. 12 | 60 | 1. 12 | 50 | 1. 117 | 3 |
| 312 | 1.087 | $s$ | 1. 10 | 40 | 1.092 | 70 | 1.087 | 8 | 1.088 | 33 | 1.090 | 80 | 1.09 | 80 | 1.09 | 60 | 1.092 | 8 |
| 411 | 1.078 | s |  | --- | 1.081 | 80 | ----- | ---- | 1.077 | 33 | 1.079 | 80 | 1.08 | 80 | 1.08 | 60 | 1.081 | 8 |
| 420 | 1.058 | ms |  | --- | 1.060 | 70 | 1.059 | 3 | 1. 057 | 17 | 1.059 | 70 | 1.06 | 50 | 1.06 | 50 | 1.059 | 3 |
| --- | 1.049 | mw | 1.05 | 10 |  | --- |  | ---- | ----- | - | 1.047 | 70 | 1.05 | 40 |  | ----- |  | --- |
| 103 | 1.035 | ms | ----- | --- | 1.037 | 40 | 1.037 | 2 | 1.033 | 17 | 1.035 | 50 | 1.04 | 40 | 1.04 | 50 | 1.036 | 4 |
| --- |  |  |  | -- |  |  |  | ---- |  | ---- | ----- | -- | 1.01 | 20 |  | ----- |  |  |
| 402 |  | ---- |  | --- | ----- | ---- | 0.950 | 8 | 0.947 | 50 | 0.947 | 80 | ------ | --- | 0.951 | 80 | 0.9505 | 8 |
| 510 |  | ---- |  | --- | ----- | ---- | . 931 | 1 | . 928 | 17 | . 928 | 60 |  | --- | . 931 | 70 | . 9291 | 3 |
| 332 |  | ---- |  | --- |  | ---- | ----- | -- | . 914 | 17 | . 914 | 60 | ------ | --- | . 916 | 78 | . 8153 | 3 |
| 501 | ----- | ---- |  | --- |  | ---- | 0.907 | 4 | . 907 | 33 | . 907 | 70 |  | --- | . 910 | 80 | . 9081 | 8 |
| 422 |  | ---- |  | --- |  | ---- | . 882 | 4 | . 881 | 50 | . 881 | 80 |  | -- |  | --. | . 8819 | 7 |
| 303 | ----- | ---- |  | --- | ----- | ---- | - | - | -.--- | -- | -- | ---- | ----- | --- | ----- | ----- | . 8814 | 6 |
| 521 |  |  |  |  |  | ---- | 0.847 | 2 | 0.848 | 50 | 0.848 | 80 |  | -- | ---- | --- | . 8480 | 6 |
| 440 |  | ---- | ------ | --- | ----- | ---- | ----- | ---- | . 838 | 17 | ----- | ---- |  | --- | ----- | ----- | . 8375 | 1 |
| 323 | ----- | ---- | ------ | --- | ----- | ---- | ----- | ---- | . 826 | 33 | ----- | ---- | ------ | --- | ----- | ----- | . 8261 | 4 |
| 530 |  | ---- |  | --- |  | ---- | ----- | ---- | . 813 | 8 | ----- | ---- |  | --- | ----- | ----- | . 8125 | 2 |
| 441 |  |  |  | --- |  |  |  | ---- | . 807 | 33 |  |  |  | -- |  |  |  |  |
| 512 |  | ---- | ------ | --- |  | ---- |  | ---- |  | ---- | ----- | -- |  | --- | ----- | -- | 0.8026 | 6 |

${ }^{a}$ Weighted K $\alpha$.

### 2.30. Ceric Oxide, $\mathrm{CeO}_{2}$ (Cubic)

Two patterns for ceric oxide in the ASTM file (see table l) are compared in table 31 with a pattern prepared at the NBS. The sample used, of unknown origin, was obtained from the NBS spectrographic laboratory, and was labeled number 41-9100. It is approximately 99.99 percent pure, showing only spectrographic traces of praseodymium and copper.

The three patterns are compared in table 31. The spacings of the Hanawalt, Rinn, and Frevel pattern were converted from kX units to angstroms. The Passerini pattern was derived directly in angstroms from the Bragg angles published. As shown in table 31 by comparing the unit cell values in column $a$, the first six spacings are not in very good
agreement with the last four. For this reason they had been replaced on the ASTM card with values calculated from the smaller interplanar spacings in the last three lines of column $d$.

The intensity measurements by Passerini were published as visual estimates which were given numerical values for the ASTM card. The first, second, and third strongest lines are 111, 220, and 311, respectively.

Ceric oxide has the fluorite structure, a face-centered cubic lattice, space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ), and four molecules to the unit cell [80]. Several determinations of the unit cell have been made but temperature data have not been published with them. McCullough used angstrom units. Assuming that the other

Table 31. Ceric oxide, $\mathrm{CeO}_{2}$ (cubic)

| $h k l$ |  |  |  | Hana | $\begin{aligned} & 1938 \\ & \text { Rinn, } \\ & 0.709 \end{aligned}$ |  |  | $1953$ $5405$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | $a$ | d | I | a | d | I | $a$ |
|  | A |  | A | A |  | A | A |  | A |
| 111 | 3.083 | 70 | 5.340 | 3.12 | 100 | 5. 40 | 3.124 | 100 | 5.411 |
| 200 | 2.678 | 40 | 5.356 | 2.70 | 25 | 5. 40 | 2.706 | 29 | 5.412 |
| 220 | 1.907 | 100 | 5.394 | 1.90 | 80 | 5.37 | 1.913 | 51 | 5.411 |
| 311 | 1.627 | 100 | 5.396 | 1.62 | 60 | 5.37 | 1.632 | 44 | 5.413 |
| 222 | 1.559 | 50 | 5.401 | 1.55 | 10 | 5.37 | 1. 562 | 5 | 5.411 |
| 400 | 1.350 | 40 | 5.400 | 1.350 | 10 | 5.400 | 1.353 | 5 | 5.412 |
| 331 | 1.242 | 80 | 5.414 | 1.239 | 25 | 5.401 | 1.241 | 15 | 5.409 |
| 420 | 1.212 | 80 | 5.420 | 1.209 | 16 | 5.407 | 1.210 | 6 | 5.411 |
| 422 | 1.107 | 100 | 5.423 | 1.103 | 20 | 5.404 | 1. 1044 | 12 | 5.4104 |
| 511 | 1.044 | 100 | 5.425 | 1.039 | 18 | 5. 399 | 1.0412 | 9 | 5.4102 |
| 440 | - |  | --- | 0.956 | 4 | 5.408 | 0.9565 | 5 | 5.4108 |
| 531 | ---- | --- | ---- | . 914 | 14 | 5.407 | . 9146 | 13 | 5.4108 |
| 600 | ----- | ---- | ----- | . 901 | 2 | 5.406 | . 9018 | 7 | 5.4108 |
| 620 | - |  |  | . 855 | 4 | 5.407 | . 8556 | 7 | 5.4113 |
| 533 |  |  |  |  |  |  | . 8251 | 6 | 5.4105 |
| 622 |  |  |  | 0.818 | 2 | 5.426 | . 8158 | 5 | 5.4114 |
| 711 |  |  |  | . 758 | 4 | 5.413 |  |  |  |
| 642 |  |  |  | . 723 | 4 | 5.410 |  |  |  |
| 731 | - |  |  | . 704 | 2 | 5. 407 |  |  |  |
| Average unit cell for last five lines $\qquad$ |  |  | 5.416 | ------- |  | 5.413 | - | --- | 5.4110 |

workers used $k X$ units, the following table makes a comparison of their values with the NBS determination, in angstroms:

Lhit cell, angstroms

| 1923 | Goldschmidt and Thomassen [80]. | 5.42 |
| :---: | :---: | :---: |
| 1925 | Goldschmidt, Ulrich, and Barth [81] | 5.413 |
| 1930 | Passerini [ 183] | 5.426 |
| 1939 | Zintl and Croatto [ 262 | 5.407 |
| 1950 | McCullough [146] | 5.411 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 5.4110 |

The density calculated from the NBS lattice constant is 7.215 at $26^{\circ} \mathrm{C}$.

### 2.31. Thorium Oxide, $\mathrm{ThO}_{2}$ (Cubic)

Two patterns for thorium oxide (thorianite) from the ASTM file (see table 1) are supplemented by three from the literature,

Van Arkel [229], Levi and Reina [138], and Burgers and Van Liempt [45], and compared in table 32 with a pattern recently prepared at the NBS. The NBS pattern was made by the use of material obtained from the Lindsay Light and Chemical Company of West Chicago, who stated a purity of 99.99 percent.

Interplanar spacings for the first three patterns of table 32 were obtained directly in angstrom units from the published Bragg angle data. At the time the Passerini pattern was transferred to the ASTM card only the last four of the interplanar spacings were copied from his data; the first six were recalculated on the basis of the unit cell derived from the remaining lines. In table 32 the original values are given instead of the ASTM values for all lines, after con-

Table 32. Thorium oxide, $\mathrm{ThO}_{2}$ (cubic)

| hkl | 1924 <br> Van Arkel <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | 1927 <br> Levi and Reina $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | 1930 <br> Burgers and Van Liempt $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  | 1930 <br> Passerini <br> $\mathrm{Fe}, 1.9360 \mathrm{~A}$ |  |  |  | 1938 <br> Hanawalt, Rinn, and Frevel <br> Mo, 0.7093 A |  |  | 1953 <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | ${ }^{\text {d }}$ | $a$ | d | $a$ | d | $a$ | d | $I^{\text {a }}$ | $I^{\text {b }}$ | $a$ | d | I | $a$ | d | I | $a$ |
|  | A | A | A | A | A | A | A |  |  | A | A |  | A | A |  | A |
| 111 | 3.14 | 5.44 | -...-- | ---.. | 3.24 | 5.61 | 3. 166 | ms | 70 | 5.485 | 3.23 | 100 | 5.59 | 3.234 | 100 | 5.602 |
| 200 | 2.73 | 5.46 | --.... | --.-- | 2.80 | 5.60 | 2.764 | w | 40 | 5.528 | 2.81 | 38 | 5.62 | 2.800 | 35 | 5.600 |
| 220 | 1.94 | 5.49 | 1.930 | 5.459 | 1.98 | 5.60 | 1.960 | vs | 100 | 5.542 | 1.97 | 75 | 5.57 | 1.980 | 58 | 5.600 |
| 311 | 1.66 | 5.51 | 1.658 | 5.499 | 1.69 | 5.61 | 1.675 | vs | 100 | 5.555 | 1.68 | 88 | 5.57 | 1.689 | 64 | 5.602 |
| 222 | 1.59 | 5.51 | ----- | ----- | 1.61 | 5.58 | 1.609 | ${ }^{\text {mw }}$ | 50 | 5.572 | ----- | -...- | ----- | 1.616 | 11 | 5.598 |
| 400 | 1.38 | 5.52 | 1.392 | 5.568 | 1.40 | 5.60 | 1.393 | w | 40 | 5.573 | 1.402 | 13 | 5.608 | 1.400 | 8 | 5. 600 |
| 331 | 1.27 | 5.54 | 1.272 | 5.545 | 1.30 | 5.67 | 1.282 | s | 80 | 5.589 | 1.283 | 38 | 5.592 | 1.284 | 26 | 5.597 |
| 420 | 1.24 | 5.55 | 1.242 | 5.554 | 1.25 | 5.59 | 1.250 | s | 80 | 5.592 | 1.248 | 25 | 5.581 | 1.252 | 17 | 5.599 |
| 422 | 1.132 | 5.546 | 1.133 | 5.551 | 1.14 | 5.58 | 1.141 | vs | 100 | 5.591 | 1.142 | 38 | 5.595 | 1.1432 | 20 | 5.6005 |
| 511 | 1.070 | 5.560 | 1.071 | 5.565 | 1.076 | 5.591 | 1.077 | vs | 100 | 5.597 | 1.076 | 38 | 5.591 | 1.0779 | 19 | 5.6010 |
| 440 | --- | -- | 0.983 | 5.561 | 0.988 | 5.589 | ----- | ---- | ---.. |  | 0.989 | 13 | 5.595 | 0.9900 | 6 | 5.6003 |
| 531 | 0.940 | 5.561 | . 941 | 5.567 | . 945 | 5.591 | ----- | ---- |  |  | . 945 | 25 | 5.591 | . 9465 | 18 | 5.5996 |
| 600 | . 929 | 5.574 | . 931 | 5.588 | . 933 | 5.598 |  | ---- |  |  | . 933 | 25 | 5.598 | . 9333 | 8 | 5.5998 |
| $620{ }^{\circ}$ | . 884 | 5.591 | . 883 | 5.582 | . 884 | 5.591 | ----- | ---- | ----- | --.-- | ----- | ----- | ----- | . 8854 | 14 | 5.5998 |
| 533 | . 852 | 5.587 | . 851 | 5.582 | . 853 | 5.593 | ----- | ---- | ----- | ----- | ----- | ----- | ----- | . 8540 | 9 | 5.6001 |
| 622 | . 843 | 5.592 | . 843 | 5.592 | . 843 | 5.592 | ----- | --- | ----- | ----- |  | ----- | ----- | . 8441 | 9 | 5.5991 |
| 444 | . 809 | 5.605 | ----- | ----- | . 807 | 5.591 | ----- | ---- |  |  | ----- | ----- | ----- | ------ | ----- | ------ |
| 711 | . 784 | 5.600 |  |  | . 783 | 5.592 |  |  |  |  |  |  |  |  |  | ------ |
| 640 |  |  |  |  | . 780 | 5.625 |  |  |  |  |  |  |  |  |  |  |
| Average unit cell for last five lines... |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
|  |  | 5.595 |  | 5.582 |  | 5.986 |  | ---- |  | 5.588 | ---.- |  | 5.594 |  | ----- | 5.5997 |

[^10][^11]version from $k X$ to angstrom units. The interplanar spacings listed by Hanawalt, Rinn, and Frevel were likewise converted from $k X$ units to angstroms for the table. The intensity data of Hanawalt, Rinn, and Frevel agree well with those of the NBS. The three strongest lines of both patterns are 111, 311, and 220.

Thorium oxide has the fluorite structure (Goldschmidt and Thomassen [80]), a facecentered cubic lattice, space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ). The unit cell contains four molecules. W. H. Zachariasen gave 5.5859 kX units for the lattice constant at the New Haven meeting of the American Society for X-ray and Electron Diffraction in 1948. This measurement and others published since 1929 (all probably in kX units and converted to angstroms) may be compared with the NBS determination thus:

Inıt cell, angstroms

| 1929 | Ruff, Ebert, and Woitinek [197] | 5.58 |
| :---: | :---: | :---: |
| 1930 | Passerini [ 183] | 5.596 |
| 1930 | Burgers and Van. Liempt [45] | 5.601 |
| 1944 | Palache, Berman, and Frondel [180] | 5.62 |
| 1948 | Zachariasen | 5.5972 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 5.5997 |

In the new (1950) ASTM file the Hanawalt, Rinn, and Frevel card (1-0731) states the lattice constant as 5.61 , ascribed correctly to " $\mathrm{D}_{7}$," that is, Dana's Mineralogy, 7th edition [180]; the Passerini card (2-1278) gives it as 5.590, ascribed to Dana, although in this case the value is actually that of Passerini. Temperature data are not available for the comparison patterns. The density calculated from the NBS lattice constant is 9.991 at $26^{\circ} \mathrm{C}$.

### 2.32. Calcium Hydroxide, $\mathrm{Ca}(\mathrm{OH})_{2}$ (Hexagonal)

The ASTM file of X-ray diffraction patterns contains four cards for calcium hydroxide (portlandite) (see table l). The four patterns are compared in table 33 with two from the literature (Levi [136] and Natta and Passerini [156]), and with a pattern prepared at the NBS.

The NBS sample was obtained as calcium carbonate from the J. T. Baker Chemical Company; it was numbered 121647. At the NBS laboratory it was heated in a platinum crucible at $925^{\circ} \mathrm{C}$ for one hour, and water added to the resulting calcium oxide in a nitrogen atmosphere.

The following chemical analysis (in percent) was provided by the chemical laboratory of the NBS: Insoluble in HCl and $\mathrm{NH}_{4} \mathrm{OH}$ ppt, 0.01 ; chloride, <0.005; sulfate, 0.037 ; alkalis (as $\mathrm{SO}_{4}$ ), 0.011 ; barium, <0.l; heavy metals (as Pb ), 0.001 ; $\mathrm{Fe}, 0.003$; MgO and alkalis, 0.21 . The laboratory of the J. T. Baker Chemical Company specified the barium content as 0.005 percent and the iron as 0.001 percent.

For table 33 the Levi and Natta and Passerini patterns were calculated directly in angstroms from Bragg angle data given. The spacings of the ASTM patterns were all converted to angstroms from $k X$ units. The fifth pattern in the table appeared in the old ASTM file with the source for it omitted; it was tentatively ascribed to the United Steel Companies, England, in the 1950 file. The Hanawalt, Rinn, and Frevel pattern and the two of the Linited Steel Companies agree with that of the NBS in showing the three strongest lines as 101, 001, and 102.

The space group assigned to hexagonal calcium hydroxide [23] is $\mathrm{D}_{3 \mathrm{~d}}^{3}$ (C3mi); there is one molecule in the unit cell. Lattice constants of several investigators are compared in the table below. All are from the literature except the two values of the United Steel Companies, which were taken from the ASTM cards, and are in angstrom units.

Unit cell, in angstroms

|  |  | $a$ | c |
| :---: | :---: | :---: | :---: |
| 1927 | Harrington [89 | 3.587 | 5.040 |
| 1933 | Tilley [222] | 3.592 | 4.905 |
| 1935 | Bunn, Clark, and Clifford [44]..-- | 3.5916 | 4. 9061 |
| ---- | United Steel.-------------------------- | 3.588 | 4.903 |
|  | United Steel (?) | 3.584 | 4.916 |
| 1953 | Swanson and Tatge ( $27^{\circ} \mathrm{C}$ ) | 3.593 | 4.909 |

The density calculated from the NBS data is 2.241. Because of the platy nature and fine grain of the material the index of re-
fraction was determined only for the ordinary ray, $\omega=1.573$. Ashton and Wilson [3] give $\omega_{\mathrm{D}}=1.574, \epsilon_{\mathrm{D}}=1.545$.

Table 33. Calcium hydroxide, $\mathrm{Ca}(\mathrm{OH})_{2}$ (hexagonal)

${ }^{\text {a }}$ Nine lines following are omitted.
2.33. Ammonium Chloride, $\mathrm{NH}_{4} \mathrm{Cl}$ (Cubic)

Four patterns from ASTM cards (see table 1) are compared in table 34 for ammonium chloride (sal-ammoniac) with a fifth pattern
made at the NBS. The NBS sample was obtained from the J. T. Baker Chemical Company. It was tested by the NBS chemical laboratory and was found to conform with ACS standards;
it was recrystallized by sublimation before exposure to $X$ rays.

The interplanar spacings of table 34 were changed to angstrom units on the basis of the wavelengths given for the radiation used in preparing the patterns, except for the Wyckoff and Armstrong spacings, which were calculated in angstroms from Bragg angle data given.

All patterns agree upon the 110 line as the strongest. For all except the NBS pattern the second strongest line is 211, and the third strongest 100 ; for the NBS pattern these are of the same intensity. This is in part due to the use of different radiation; however, recalculation of the intensities to
a common basis by the use of the ASTM conversion scale ([1] p. 108 of index covering original set of cards, or card no. vii of introduction to 1950 file) preserves the same choice of the three strongest lines although reducing the discrepancies in intensity between patterns. Bartlett and Langmuir missed the 100 line of the pattern, and weak lines are missing from other patterns.

An error has been carried over from the old ASTM card to the new one for the Bartlett and Langmuir pattern; in column $d$ the second interplanar spacing was originally published as 2.238 rather than the 2.338 of the ASTM card. The radiation wavelength is given on

Table 34. Anmonium chloride, $\mathrm{NH}_{4} \mathrm{Cl}$ (cubic)

| $h k l$ | Bart <br> Mo, | 1921 <br> lett <br> ngmui <br> 0.70 | and | Havig Mo, | $\begin{aligned} & 1924 \\ & \text { wist, } \\ & \mathrm{d} \text { Bla } \\ & 0.700 \end{aligned}$ | Mack <br> k <br> 3 A | Hyc <br> Ar <br> Mo, | $\begin{aligned} & 1929 \\ & \text { koff } \\ & \text { mstror } \\ & 0.709 \end{aligned}$ | nd $g$ 3 A | Hanaw and Mo, | $\begin{aligned} & 1938 \\ & \text { alt, } \\ & 1 \text { Fre } \\ & 0.70 \end{aligned}$ | Rinn, el <br> 3 A |  | ted $S$ ies, 0.7 | eel <br> ngland <br> 3 A | Swans $\mathrm{Cu}, 1$. | 1953 | Tatge $26^{\circ} \mathrm{C}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I | $a$ | $a$ | $I$ | $a$ | d | I | $a$ | d | $I$ | $a$ | d | I | $a$ | d | I | $a$ |
|  | A |  | A | A |  | A | A |  | A | A |  | A | A |  | 4 | A |  | A |
| 100 |  | -- |  | 3.87 | 30 | 3.870 | 3.85 | 8 | 3.85 | 3.86 | 15 | 3.86 | 3.87 | 60 | 3.87 | 3.87 | 23 | 3.87 |
| 110 | 2.708 | 100 | 3.829 | 2.731 | 100 | 3.862 | 2.728 | 100 | 3.857 | 2.73 | 100 | 3.86 | 2.74 | 100 | 3.87 | 2.740 | 100 | 3.875 |
| 111 | 2.229 | 10 | 3.861 | 2.231 | 7 | 3.864 | 2.226 | 3 | 3.856 | 2.22 | 5 | 3.85 | 2.24 | 50 | 3.88 | 2.238 | 4 | 3.876 |
| 200 | 1.917 | 20 | 3.834 | 1.932 | 15 | 3.864 | 1.928 | 11 | 3.856 | 1.92 | 12 | 3.84 | 1.94 | 60 | 3.88 | 1.939 | 7 | 3.878 |
| 210 | 1.718 | 15 | 3.842 | 1.726 | 12 | 3.859 | 1.724 | 6 | 3.855 | 1.72 | 8 | 3.85 | 1.73 | 50 | 3.87 | 1.733 | 5 | 3.875 |
| 211 | 1.562 | 30 | 3.826 | 1.556 | 40 | 3.811 | 1.575 | 21 | 3.858 | 1.57 | 25 | 3.85 | 1.58 | 70 | 3.87 | 1.582 | 23 | 3.875 |
| 220 | 1.357 | 15 | 3.838 | 1.365 | 10 | 3.862 | 1.363 | 6 | 3.855 | 1.373 | 5 | 3.883 | 1.37 | 60 | 3.87 | 1.370 | 5 | 3.875 |
| 300 | 1.282 | 10 | 3.846 | 1.288 | 12 | 3.863 | 1.286 | 3 | 3.858 | 1.291 | 3 | 3.873 | 1.29 | 50 | 3.87 | 1.292 | 3 | 3.876 |
| 310 | 1.209 | 15 | 3.823 | 1.221 | 15 | 3.861 | 1.220 | 6 | 3.858 | 1.223 | 7 | 3.867 | 1.22 | 60 | 3.86 | 1.225 | 5 | 3.874 |
| 311 | 1.158 | 8 | 3.841 | 1.165 | 3 | 3.863 | 1.163 | 2 | 3.857 | 1.167 | 1 | 3.871 | 1.17 | 50 | 3.88 | 1.1687 | 4 | 3.8761 |
| 222 | 1.107 | 8 | 3.835 | 1.116 | 2 | 3.865 | 1.113 | 2 | 3.856 | 1.117 | 1 | 3.869 | 1.12 | 50 | 3.88 | 1.1188 | 2 | 3.8576 |
| 320 | 1.069 | 6 | 3.854 | 1.070 | 2 | 3.859 | 1.069 | 1 | 3.854 | ----- | --- |  | 1.07 | 50 | 3.86 | 1.0751 | 1 | 3.8764 |
| 321 | 1.020 | 18 | 3.816 | 1.032 | 20 | 3.860 | 1.031 | 6 | 3.858 | 1.035 | 4 | 3.873 | 1.03 | 80 | 3.85 | 1.0357 | 3 | 3.8753 |
| 400 | 0.954 | 3 | 3.816 | 0.965 | 1 | 3.858 | 0.964 | 1 | 3.856 | ----- | --- | ----- | 0.968 | 50 | 3.872 | 0.9680 | 1 | 3.8720 |
| 410 | . 931 | 6 | 3.839 | . 937 | 1 | 3.862 | . 935 | 1 | 3.855 | ----- | --- | ----- | . 939 | 70 | 3.872 | . 9400 | 2 | 3.8757 |
| 411 | . 908 | 8 | 3.852 | . 910 | 6 | 3.860 | . 909 | 2 | 3.857 | 0.914 | 1 | 3.878 | . 913 | 85 | 3.873 | . 9134 | 3 | 3.8752 |
| 331 | . 888 | 5 | 3.871 | ----- | --- | ----- | . 885 | 1 | 3.857 | ----- | --- | ----- | ----- | --- | ------ | . 8890 | 1 | 3.8751 |
| 420 | . 864 | 5 | 3.864 | 0.864 | 3 | 3.863 |  | -- |  | 0.866 | 1 | 3.873 |  |  |  | . 8667 | 2 | 3.8760 |
| 421 | . 841 | 5 | 3.854 | . 844 | 2 | 3.866 | ----- | --- | ----- | ----- | --- | ----- | ----- | --- |  | . 8457 | 3 | 3.8755 |
| 332 | . 819 | 5 | 3.841 | . 823 | 1 | 3.858 |  | -- |  |  |  |  |  |  |  | . 8263 | 4 | 3.8757 |
| 422 | . 784 | 4 | 3.841 | . 788 | 2 | 3.860 |  | -- |  |  | --- |  |  | -- |  | . 7911 | 3 | 3.8756 |
| 500 | . 769 | 3 | 3.845 | . 774 | 4 | 3.869 |  | --- |  |  | --- |  |  | --- |  | ----- | --- | ------ |
| 510 | . 752 | 7 | 3.834 | . 757 | 7 | 3.863 | ----- | --- | ----- | ----- | --- |  |  | --- |  |  | --- | ------ |
| 521 |  |  |  | . 706 | 1 | 3.866 |  |  |  |  | --- |  |  |  |  |  |  |  |
| Average unit cell for last five lines.-- |  |  | 3.843 |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
|  |  |  | 3.863 |  |  | ----- | --- | 3.857 |  | --- | 3.873 |  | --- | ${ }^{\text {a }} 3.872$ | ------ | --- | 3.8756 |

[^12]the old card as 0.708 , on the new one 0.709 , while 0.712 was actually used. On the old card for the Havighurst, Mack, and Blake pattern the interplanar spacings 1.2.2l and 0.7748 should read 1.2221 and 0.7745 ; these errors were eliminated from the new card in reducing the values to two decimal places only. The radiation wavelength was changed on the cards from 0.708 to 0.709 ; actually 0.710 was used. The Wyckoff and Armstrong reference is given incorrectly on both old and new cards; it should read "Z. Krist. 72, 319 (1929)" rather than "320 (1930)." The interplanar spacings and intensity measurements check with those of the reference given; it is not understood why reference is given also to Greenberg and Walden [82] (misspelled "Walder" on the new card) with the notation "intensities by ionization spectrometer," since the intensity data of this paper seem of little value and are apparently not recorded on the ASTM card at all.

The structure of ammonium chloride [54] is based on a simple cubic lattice, space group $\mathrm{O}_{\mathrm{h}}^{1}\left(\mathrm{Pm}_{\mathrm{m}} \mathrm{m}\right)$. There is one molecule to the unit cell. The early lattice constant determinations (Bragg [28], Vegard [235], and Bartlett and Langmuir [9]) vary considerably, and uncertainty exists as to the correction to apply to convert their units to angstroms. In the table below a value of Havighurst, Mack, and Blake [92] is compared with an electron diffraction determination of Trillat and Laloeuf [223], and the NBS lattice constant. The two former are converted from asm sumed $k X$ units to angstroms.

Unit cell, angstroms

| 1924 | Havighurst, Mack, and Blake [92].--- | 3.874 |
| :---: | :---: | :---: |
| 1948 | Trillat and Laloeuf [223] | 3.871 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 3.8756 |

The density of ammonium chloride based on the unit cell of the NBS is 1.527 at $26^{\circ} \mathrm{C}$. The index of refraction of the sample was determined as $n=1.641$.

### 2.34. Lithium Fluoride, LiF (Cubic)

Six patterns are compared in table 35; besides the three from the ASTM file (see table l), there is one from Debye and Scherrer [63], one from Bruni and Levi [41], and an NBS pattern. In their same publication Debye and Scherrer give a second pattern which closely corresponds to the first and is not reproduced in table 35.

The sample of lithium fluoride used for the NBSpattern was obtained from the Harshaw Chemical Co. Spectrographic analysis at the NBS showed (in percent): $\mathrm{Sr}, 0.01$ to 0.1 ; $\mathrm{Pb}, \mathrm{Si}, 0.001$ to $0.01 ; \mathrm{Al}, \mathrm{Ba}, \mathrm{Ca}, \mathrm{Cu}, \mathrm{Fe}$, $\mathrm{Mg}, \mathrm{Sn},<0.001$.

The interplanar spacings of the ASTM patterns in table 35 were converted from $k X$ to angstrom units. The data of Debye and Scherrer and of Rruni and Levi were published in Bragg angles, from which the interplanar spacings were derived directly in angstroms. Most of the relative intensity measurements of the various patterns bear out the choice of the three strongest lines as given in the NBS pattern; 200, 111, and 220 are the first, second, and third strongest lines, respectively.

Lithium fluoride has the well-known NaCl structure, space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ), with four molecules to the unit cell. Published lattice constants after converting from $k X$ to angstrom units and allowing for temperature differences (the coefficient of expansion was determined by Straumanis, Ieviņs and Karlsons [217] as $34.17 \times 10^{-6}$ ) compare with the NBS value thus:

Unit cell at $25^{\circ} \mathrm{C}$, angstroms

| 1937 | Moeller [155] | 4.0286 |
| :---: | :---: | :---: |
| 1939 | Straumanis, Ievinš, and Karlsons [217] | 4.02620 |
| 1940 | Hutchison and Johnston [110] | 4.0255 |
| 1953 | Swanson and Tatge---------------------- | 4.0264 |

The density based on the NBS lattice constant is 2.638 at $25^{\circ} \mathrm{C}$. The refractive index of lithium fluoride determined by Spangenberg [211] is $n_{D}=1.3915$.

Table 35. Lithium fluoride, LiF (cubic)

${ }^{\text {a }}$ Average for last three lines only.

### 2.35. Lithium Chloride, LiCl (Cubic)

In table 36 the three patterns for lithium chloride included in the ASTM diffraction pattern file (see table l) are compared with a pattern prepared at the NBS. The material for the NBS sample was obtained from the Mallinckrodt Chemical Works, labelled with their number SDD, and was accompanied by the following chemical analysis (in percent): $\mathrm{N}_{2} \mathrm{O}_{5}, 0.001 ; \mathrm{SO}_{3}, 0.01$; heavy metals, 0.005 ; $\mathrm{Fe}, 0.001$; other alkalis, $0.02 ; \mathrm{Cl}_{2} \mathrm{O}_{5}$, trace.

The interplanar spacings were converted for the table from $k X$ units to angstroms. The intensity measurements of all four patterns yield 111, 200, and 220 as the three strongest lines to be used as the ASTM index lines. It should be noted that in the flat packed sample used with Geiger counter apparatus the powder easily orients so that the 200 line is strongest. The NBS intensity measurements were made as usual on a loosely
packed sample expressly prepared to avoid orientation.

Lithium chloride crystallizes with the NaCl lattice, has the space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ), and has four molecules in the unit cell. A recent value for the lattice constant is compared with that of the NBS in the following table after conversion to angstrom units at $25^{\circ} \mathrm{C}$. The coefficient of expansion of $44.76 \times 10^{-6}$ [113] was employed in making the corrections.

Unit cell in angstroms at $25^{\circ} \mathrm{C}$


The density, calculated from the NBS lattice constant, is 2.074 at $25^{\circ} \mathrm{C}$. The index of refraction determined on the material used for the NBS pattern is $n=1.663$.

Table 36. Luthum chloride, LiCl (cubuc)


### 2.36. Sodium Fluoride, NaF (Cubic)

Three patterns for sodium fluoride (villiaumite) recorded on the ASTM file cards (see table l) are compared in table 37 with two patterns obtained from the literature and one made at the NBS. Those from the literature are by Debye and Scherrer [64] and by Wasastjerna [244]. A fourth ASTM pattern, from the Crystallographic Laboratory, Cambridge, England, had not been published prior to the ASTM compilation, and as the data were combined on the file card with those of Wyckoff and Armstrong, it is impossible to reproduce it as a separate pattern for comparison in table 37.

The NBS pattern was obtained from a sample numbered 7445 furnished by the J. T. Baker Chemical Company. The NBS chemical laboratory found that the sample complied with ACS specifications. The spectrographic labora-
tory reported the presence of silicon, 0.001 to 0.01 percent, and no other impurity greater than 0.001 percent. The sample was recrystallized by sublimation before using.

The spacings of the Debye and Scherrer and the Wyckoff and Armstrong patterns of table 37 were calculated directly in angstrom units from the published Bragg angle data. The Davey, the Hanawalt, Rinn, and Frevel, and the Wasastjerna patterns were converted from the $k X$ units of the published data. Thus, the entire table is in angstroms.

The patterns of table 37 agree in showing the 200,220 , and 222 lines as the first, second, and third strongest, respectively. The Wyckoff and Armstrong intensity measurements in their original publication differ considerably from those on either the old or new ASTM cards. They were recalculated for
table 37 directly from the published photometric measurements, which have lost much of their original precision in the versions used on the ASTM cards.

Sodium fluoride has the NaCl structure, space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ), with four molecules in the unit cell. In 1939 Straumanis, Ieviņs, and Karlsons [217] found a lattice constant of 4.62345 kX units at $25^{\circ} \mathrm{C}$. Converting to angstroms at $25^{\circ} \mathrm{C}$, this compares with the NBS value:

Unit cell at $25^{\circ} \mathrm{C}$, angstroms

| 1939 | Straumanis, Ievinš, and Karlsons [217] | 4.63279 |
| :---: | :---: | :---: |
| 1953 | Swanson and Tatge | 4.6342 |

The coefficient of expansion $36.0 \times 10^{-6}$ [217] was used. The density of sodium fluo-* ride based on the NBS cell value of the lattice constant is 2.799. The index of refraction was given by Spangenberg [211] as $n_{\mathrm{D}}=1.3258$.

Table 37. Sodium fluoride, NaF (cubic)


### 2.37. Potassium Fluoride, KF (Cubic)

Three diffraction patterns, all in the ASTM file, for potassium fluoride (see table 1) are compared with a pattern prepared at the NBS in table 38 . The NBS sample was obtained from the J. T. Baker Chemical Company, and accompanied by the following analysis (in percent): insoluble, 0.05 ; Cl, 0.005 ; HF (free acid), 0.05 ; alkali $\left(\mathrm{K}_{2} \mathrm{CO}_{3}\right), 0.1$; $\mathrm{K}_{2} \mathrm{SiF}_{6}, 0.05 ; \mathrm{SO}_{4}, 0.02 ; \mathrm{SO}_{3}, 0.005$; heavy metals (as Pb ), $0.003 ; \mathrm{Fe}, 0.001$. A spectro-
graphic analysis made at the NBS indicates approximately $0 . l$ percent sodium present, the only extraneous element recognized greater than 0.01 percent. The presence of 0.1 percent sodium fluoride in solid solution with potassium fluoride decreases the unit-cell size approximately 0.0007 A. As potassium fluoride is very deliquescent, the sample was first dried at $220^{\circ} \mathrm{C}$, then mixed with petrolatum before mounting on the $X$-ray spectrometer.

The spacings of the Davey pattern were converted to angstrom units by recalculating them to agree with a molybdenum wavelength of 0.7093 angstroms rather than the 0.712 used to obtain the published data. The spacings of the remaining ASTM patterns were converted to angstroms from $k X$ units. The Hanawalt, Rinn, and Frevel, and the NBS patterns show the 200,220 , and 111 lines as the three strongest. The 420 and 422 are strong lines also, and, due to focusing and absorption effects, appear in the other two patterns as strong or stronger than the 111 line.

Potassium fluoride has the well known face-centered cubic structure of NaCl , space group $\mathrm{O}_{\mathrm{h}}^{5}\left(\mathrm{Fm}_{\mathrm{m}} 3 \mathrm{~m}\right)$, with four molecules in the unit cell. The lattice constants found in
the literature are not in close agreement. Several, converted to angstrom units, are compared with that of the NBS in the table below:

Unit cell, in angstroms

| 1922 | Posnjak and Wyckoff [187 | 5.37 |
| :---: | :---: | :---: |
| 1929 | Broch, Oftedal, and Pabst [38 ] | 5.344 |
| 1938 | Finch and Fordham ${ }^{\text {a }}$ [68] | 5.367 |
| 1948 | Mehmel [150] | 5.34 |
| 1953 | Swanson and Tatge ( $26{ }^{\circ} \mathrm{C}$ ) | 5.347 |

${ }^{a}$ By electron diffraction.
The density, calculated from the NBS lattice constant, is 2.524 at $26^{\circ} \mathrm{C}$. The index of refraction was not determined because of the fineness of the sample; it is given by Spangenberg [211] as $n_{D}=1.361$.

Table 38. Potassium fluoride, KF (cubic)

| $h k l$ |  | 1923 <br> Davev <br> 0.70 |  |  | 1938 <br> alt, <br> Frev <br> 0.709 |  |  | llogr orator $0.709$ |  | Sw Cu | 1953 and 405 A | ge $6^{\circ} \mathrm{C}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | 2 | I | $a$ | a | I | $a$ | d | I | $a$ | d | I | $a$ |
|  | A |  | A | A |  | A | A |  | A | A |  | A |
| 111 | 3.10 | 15 | 5.37 | 3.09 | 27 | 5.35 | 3.02 | 40 | 5.23 | 3.087 | 29 | 5.347 |
| 200 | 2.68 | 100 | 5.36 | 2.67 | 100 | 5.34 | 2.64 | 100 | 5.28 | 2.671 | 100 | 5.342 |
| 220 | 1.88 | 80 | 5.317 | 1.88 | 83 | 5.32 | 1.87 | 80 | 5.29 | 1.890 | 63 | 5.346 |
| 311 | 1.603 | 10 | 5.317 | 1.60 | 10 | 5.31 | 1.599 | 20 | 5.303 | 1.612 | 10 | 5.346 |
| 222 | 1.533 | 20 | 5.310 | 1.54 | 27 | 5.33 | 1.533 | 40 | 5.310 | 1.542 | 17 | 5.342 |
| 400 | 1.328 | 10 | 5.312 | 1.336 | 8 | 5.344 | 1.330 | 40 | 5.320 | 1.337 | 8 | 5.348 |
| 331 | 1.217 | 8 | 5.305 | 1. 225 | 4 | 5.340 | 1.219 | 20 | 5.313 | 1.227 | 2 | 5.348 |
| 420 | 1. 187 | 25 | 5.308 | 1.193 | 20 | 5.335 | 1.190 | 80 | 5.322 | 1.1946 | 14 | 5.342 |
| 422 | 1.083 | 15 | 5.306 | 1.091 | 10 | 5.345 | 1.091 | 80 | 5.345 | 1.0912 | 8 | 5.346 |
| 511 | 1.020 | 5 | 5.300 | 1.029 | 1 | 5.347 |  | ----- |  | 1.0297 | 3 | 5.350 |
| 440 | 0.935 | 5 | 5.289 | 0.945 | 1 | 5.346 | 0.945 | 20 | 5.346 | 0.9452 | 3 | 5.347 |
| 531 | . 897 | 5 | 5.307 | . 903 | 1 | 5.342 | . 904 | 40 | 5.348 | . 9037 | 4 | 5.346 |
| 600 | . 884 | 8 | 5.304 | . 891 | 2 | 5.346 | - | --- |  | . 8915 | 5 | 5.349 |
| 620 | . 839 | 8 | 5. 306 | . 845 | 1 | 5.344 | ----- | ----- | ------ | . 8455 | 5 | 5.347 |
| 622 | . 800 | 8 | 5.307 |  |  |  | --- |  |  | . 8060 | 4 | 5.346 |
| 642 | . 708 | 8 | 5.298 |  |  |  |  |  |  |  |  |  |
| Average unit cell for last five lines. |  |  |  |  |  |  |  |  | 5.335 |  |  | 5.347 |
|  |  |  | 5.304 |  |  | 5.345 |  |  |  |  |  |  |

### 2.38. Potassium Chloride, KCl (Cubic)

The two patterns of potassium chloride (sylvite) in the ASTM file (see table l), supplemented by two found in the literature, Wasastjerna [244a] and Sidhu [207], are com-
pared in table 39 with a pattern made at the NBS.

The sample used by the NBS was obtained from the Mallinckrodt Chemical Works, and bore the label KYD-1. The following chemi-
cal analysis accompanied it (in percent): Ba, $0.001 ; \mathrm{Ca}, \mathrm{Mg}$, and $\mathrm{NH}_{4} \mathrm{OH} \mathrm{ppt}, 0.005$; Chlorate $\left(\mathrm{ClO}_{3}\right), 0.001$; insoluble, 0.005 ; $\mathrm{Fe}, 0.0003$; heavy metals, 0.0005 ; neutrality $\mathrm{OK} ; \mathrm{NO}_{3}, 0.003$; $\mathrm{N}, 0.001 ; \mathrm{PO}_{4}, 0.002 ; \mathrm{Na}, 0.02 ; \mathrm{SO}_{4}, 0.005$.

The conversion of the interplanar spacings of the patterns of table 39 to angstrom units was made from $k X$ units except in the case of Wasastjerna, from whose data published as $\frac{\sin \theta}{\lambda}$, values were calculated directly in angstrom units. For each pattern the three strongest lines are 200, 220, and 222.

Apparently Bragg [29] is responsible for the original structure determination; he referred to the structure as "simple cubic," that is, face-centered cubic, having the space group $\mathrm{O}_{\mathrm{h}}^{5}$ ( Fm 3 m ). Tu [224] in 1936 determined the coefficient of expansion of potassium
chloride as $3.65 \times 10^{-5}$. Four measurements of the latice constant made at specified temperatures are compared, after conversion to angstrom units at $25^{\circ} \mathrm{C}$, with the NBS value in the following table:

Unit cell in angstroms at $25^{\circ} \mathrm{C}$

| 1936 | Tu [224] | 6.29229 |
| :---: | :---: | :---: |
| 1942 | Batuecas and Fernandez-Alonso [10]_-- | 6.307 |
| 1944 | Hutchinson [111] | 6.30511 |
| 1947 | Vegard [237] | 6.289 |
| 1953 | Swanson and Tatge.------------------------ | 6.2931 |

Hutchinson, and Batuecas and FernandezAlonso did not obtain their lattice constants from X -ray measurements but from precision density determinations. The density calculated from the NBS unit cell is 1.9865 at $25^{\circ} \mathrm{C}$. An index of refraction of $n=1.490$ was obtained for the NBS sample of potassium chloride.

Table 39. Potassium chloride, KCl (cubic)

|  |  | 1923 |  |  | 1938 |  |  |  |  | 948 |  |  | 953 |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $h k l$ |  | Davey |  | Hana | Fre |  | Wasa | erna |  | dhu |  | Swans | and |  |
|  | Mo | 0.70 |  | Mo, | 0.70 |  | $\mathrm{Cu}, 1$ | 405 A |  | . 54 |  | $\mathrm{Cu}, 1$ | 05 | $25^{\circ} \mathrm{C}$ |
|  | d | $I$ | $a$ | a | $I$ | $a$ | d | $a$ | a | $I$ | $a$ | d | I | $a$ |
|  | A |  | A | A |  | A | A | A | A |  | A | A |  | A |
| 200 | 3.1 | 100 | 6.24 | 3.14 | 100 | 6.28 | 3.147 | 6.294 | 3.13 | vs | 6.26 | 3.146 | 100 | 6.292 |
| 220 | 2.21 | 67 | 6.25 | 2.21 | 60 | 6.25 | 2.224 | 6.290 | 2.21 | s | 6.25 | 2.224 | 59 | 6.290 |
| 222 | 1.812 | 20 | 6.277 | 1.81 | 14 | 6.27 | 1.817 | 6.294 | 1.81 | m | 6.27 | 1.816 | 23 | 6.291 |
| 400 | 1.567 | 7 | 6.268 | 1.57 | 6 | 6.28 | 1.573 | 6.292 | 1.57 | w | 6.28 | 1.573 | 8 | 6.292 |
| 420 | 1.403 | 17 | 6.274 | 1.404 | 12 | 6.279 | 1.407 | 6.292 | 1.41 | m | 6.31 | 1.407 | 20 | 6.292 |
| 422 | 1.281 | 10 | 6.275 | 1.283 | 6 | 6.285 | 1.285 | 6.295 | 1.28 | m | 6.27 | 1.284 | 13 | 6. 290 |
| 440 | 1.110 | 3 | 6.279 | 1.110 | 2 | 6.279 | 1.112 | 6.294 | 1.11 | w | 6.28 | 1.1126 | 2 | 6.2938 |
| 600 | 1.045 | 7 | 6.270 | 1.049 | 2 | 6.294 | 1.049 | 6.294 | 1.05 | w | 6.30 | 1.0490 | 6 | 6.2940 |
| 620 | 0.991 | 3 | 6.268 | 0.993 | 2 | 6.280 | 0.9948 | 6.2917 | 0.994 | w | 6. 287 | 0.9951 | 2 | 6.2936 |
| 622 | . 994 | 3 | 6.261 |  | --- |  | . 9485 | 6.2930 | . 950 | w | 6.302 | . 9486 | 3 | 6.2923 |
| 444 | . 905 | 3 | 6.270 |  | --- | ------- | . 9083 | 6.2929 | . 912 | vw | 6.319 | . 9083 | 1 | 6.2929 |
| 640 | . 869 | 3 | 6.266 |  | --- | ------- | . 8727 | 6.2931 | . 875 | w | 6.310 | . 8727 | 2 | 6.2931 |
| 642 | . 840 | 3 | 6.271 |  | --- |  | . 8409 | 6.2927 | . 843 | m | 6.308 | . 8410 | 6 | 6.2934 |
| Average for last five lines. $\qquad$ |  |  | 6.267 | ------- | --- | 6.283 | -------- | 6.2927 | ------- | -- | 6.305 | -------- | --- | 6.2931 |
|  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |

### 2.39. Potassium Bromide, KBr (Cubic)

The two ASTM patterns of potassium bromide (see table 1), one from the literature (Wasastjerna [243]), and a pattern prepared at the NBS are compared in table 40 . The mate-
rial for the NBS pattern was obtained from the J. T. Baker Chemical Company, numbered 111642 , and accompanied by the following analysis (in percent): insoluble, $0.001 ; \mathrm{PO}_{4}, 0.000 ; \mathrm{SO}_{4}$, 0.003 ; heavy metals, $0.0001 ; \mathrm{KOH}, 0.002 ; \mathrm{N}$,
$0.0001 ; \mathrm{Ca}, \mathrm{Mg}$, and $\mathrm{NH}_{4} \mathrm{OH}$ ppt, $0.003 ; \mathrm{Cl}, 0.1$; $\mathrm{BrO}_{3}, 0.001 ; \mathrm{Ba}, 0.002 ; \mathrm{Fe}, 0.0001$. It was checked at the NBS and found to comply with ACS reagent standards.

## After the calculation of the Wasastjerna

 spacings from the given $\frac{\sin \theta}{\lambda}$ data, the interplanar spacings of the three published patterns were converted from $k X$ to angstrom units. Two errors in entering intensity measurements on the Davey ASTM card were corrected for the 1950 file. The three lines 200,220 , and 420 are recognized as the first, second, and third strongest lines, respectively, in each of the four patterns.Potassium bromide has a face-centered cubic lattice [29], space group $0_{h}^{5}(F m 3 m)$, and four molecules to the unit cell. Three determinations of the lattice constant, given at specified temperatures, are compared below
with the NBS determination, all reduced to angstrom units at $25^{\circ} \mathrm{C}$. A recent determination of the coefficient of expansion is $40.5 \times 10^{-6}$ [52].

Lnit cell in angstroms at $25^{\circ} \mathrm{C}$

| 1926 | Ott [172] | 6.600 |
| :---: | :---: | :---: |
| 1942 | Batuecas and Fernandez-Alonso [10] | 6.616 |
| 1947 | Vegard [237] | 6.593 |
| 1953 | Swanson and Tatge | 6.6000 |

Batuecas and Fernandez-Alonso did not obtain their lattice constant from $X$-ray measurements but calculated it from a pycnometric density determination of high precision. The density, calculated from the NBS diffraction data, is 2.7533 at $25^{\circ} \mathrm{C}$. The index of refraction of the specimen used for the NBS pattern was determined as $n=1.559$.

Table 40. Potassium bromide, KBr (cubic)

| $h k l$ |  | 1923 <br> Davey <br> 0.709 |  | Hanawa | 1938 <br> Rinn, $0.7093$ | nd Frevel <br> A | Was Cu, | erna <br> 05 A |  | $1953$ $5405$ | tge $25^{\circ} \mathrm{C}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | a. | d | I | $a$ | a | $a$ | d | I | $a$ |
|  | A |  | A | A |  | A | ${ }^{\text {A }}$ | A | A |  | A |
| 111 | 3.79 | 20 | 6.56 |  | ----- |  | 3.804 | 6.589 | 3.81 | 15 | 6.60 |
| 200 | 3.28 | 100 | 6.55 | 3.30 | 100 | 6.59 | 3.296 | 6.592 | 3.300 | 100 | 6.600 |
| 220 | 2.33 | 90 | 6.57 | 2.34 | 42 | 6.60 | 2.330 | 6.590 | 2.333 | 57 | 6.599 |
| 311 | 1.961 | 15 | 6.563 | ----- | ----- | ---------- | 1.987 | 6.590 | 1. 990 | 7 | 6.600 |
| 222 | 1.899 | 50 | 6.577 | 1.89 | 10 | 6.56 | 1.903 | 6.592 | 1.905 | 16 | 6.599 |
| 400 | 1.641 | 15 | 6.565 | 1.64 | . 7 | 6.57 | 1.648 | 6.592 | 1.650 | 10 | 6.600 |
| 331 | 1.513 | 8 | 6.595 |  | ----- | --------- | 1.512 | 6.591 | 1.514 | 2 | 6.599 |
| 420 | 1. 468 | 60 | 6.565 | 1.471 | 17 | 6.578 | 1.474 | 6.592 | 1.476 | 17 | 6.601 |
| 422 | 1.346 | 30 | 6.592 | 1.346 | 7 | 6.592 | 1.345 | 6.589 | 1.347 | 8 | 6.599 |
| 511 | -------- | ----- | ----- | ------- | --- | --------- | 1.268 | 6.589 | 1.270 | 2 | 6.599 |
| 440 | 1.164 | 8 | 6.586 | 1. 166 | 3 | 6.598 | 1.1651 | 6.5909 | 1.1666 | 3 | 6.5993 |
| 531 | -------- | ----- |  |  | ----- | ---------- | 1.1141 | 6.5911 | 1.1157 | 1 | 6.6006 |
| 600 | 1.098 | 10 | 6.589 | 1.097 | 3 | 6.583 | 1.0984 | 6.5904 | 1.1000 | 5 | 6.6000 |
| 620 | 1.038 | 10 | 6.565 | 1.042 | 3 | 6.591 | 1.0422 | 6.5915 | 1.0437 | 4 | 6.6009 |
| 533 |  |  |  | ------ | ----- | --------- | 1.0052 | 5.5915 | ------- |  |  |
| 622 | 0.991 | 5 | 6.573 | --... | ----- | --------- | 0.9937 | 6.5915 | 0.9949 | 4 | 6.5994 |
| 444 | -..--- | ... | ----- |  | ----- |  | . 9514 | 6.5915 | . 9527 | 2 | 6.6002 |
| 711 | ------- | ----- | ----- |  | ----- | --------- | . 9230 | 6.5915 | . 9241 | 1 | 6.5997 |
| 640 | ------ | ---- | ----- | ----- | ----- | --------- | . 9141 | 6.5917 | . 9153 | 2 | 6.6003 |
| 642 |  |  |  |  |  |  | . 2308 | 6 5913 | . 8819 | 3 | 6.5995 |
| 731 |  |  |  |  |  |  | . 8582 | 6.5919 | . 8594 | 1 | 6.6002 |
| Average unit cell for last five lines. |  |  |  |  |  |  |  |  |  |  |  |
|  |  |  | 6.581 |  |  | 6.588 | ----- | 6.5916 |  | ---- | 6.6000 |

### 2.40. Potassium Iodide, KI (Cubic)

Three patterns for potassium iodide included in the ASTM file of powder diffraction patterns (see table l) are compared in table 41 with a pattern found in the literature, Wasastjerna [244], and one prepared at the NBS. The specimen used for the NBS pattern was obtained from B. R. Elk \& Company, sample No. E-PF-3, accompanied by the following analysis, denoting higher purity than required by ACS specifications (in percent): alkali, 0.04; $\mathrm{Ba}, 0.002 ; \mathrm{Ca}, \mathrm{Mg}$, and $\mathrm{NH}_{4} \mathrm{OH}$ ppt, $0.005 ; \mathrm{Cl}$ and $\mathrm{Br}, 0.01$; insoluble, $0.005 ; \mathrm{IO}_{3}$, 0.0003 ; $\mathrm{Fe}, 0.0003$; heavy metals (as Pb ),
$0.0005 ; \mathrm{H}_{2} \mathrm{O}, 0.20 ; \mathrm{N}, 0.002 ; \mathrm{PO}_{4}, 0.005 ; \mathrm{Na}$, $0.03 ; \mathrm{SO}_{4}, 0.01$. Annealing at $450^{\circ} \mathrm{C}$ for a half hour presumably was accompanied by elimination of the water.

The spacings of the patterns were either calculated directly in angstrom units from the Bragg angle data given or were converted from kX to angstrom units. There is general agreement that the first two strongest lines are 200 and 220, but the Davey and the Olshausen patterns show 420 as third strongest, while the Hanawalt, Rinn, and Frevel pattern agrees with that of the NBS in showing the 111 line as third strongest.

Table 41. Potassium iodide, KI (cubic)


Potassium iodide has the NaCi structure [253], face-centered cubic, with four molecules to the unit cell, and space group $O_{b}^{5}$ (Fm3m). Lattice constants of several investigators are compared as follows (Finch and Fordham obtained theirs from electron diffraction measurements):

Unit cell in angstroms

| 1922 | Clark and Duane [50] | 7.064 |
| :---: | :---: | :---: |
| 1923 | Davey [57] | 7.050 |
| 1924 | Havighurst, Mack, and Blake [92] | 7.052 |
| 1925 | Olshausen [170] | 7.040 |
| 1936 | Finch and Fordham [68] | 7.078 |
| 1953 | Swanson and Tatge ( $25^{\circ} \mathrm{C}$ ) | 7.0655 |

The density calculated from the NBS lattice constant is 3.1257 at $25^{\circ} \mathrm{C}$. The index of refraction of the sample used by the Bureau was determined as $n=1.668$.

### 2.41. Calcium Fluoride, $\mathrm{CaF}_{2}$ (Cubic)

Calcium fluoride (fluorite) is represented in table 42 by four patterns reproduced in the ASTM file (see table l), one appearing in the literature, Gerlach [78], and one prepared at the NBS.

The sample of calcium fluoride used for the NBS pattern was prepared by D. C. Stockbarger at the Massachusetts Institute of Technology. Spectrographic analysis at the NBS showed arsenic, boron, iron, magnesium, silicon, and strontium less than 0.001 percent each, and silver and copper less than 0.0001 percent.

The spacings of the Gerlach pattern were computed for table 42 in angstrom units directly from the published Bragg angle data. Those of the four remaining patterns, which appear on the ASTM cards, were converted from kX units to angstroms. Of these, only the Hanawalt, Rinn, and Frevel pattern is known to be previously published. As can readily be seen from the unit-cell calculations of the table, the precision of the Jessop-United Steel Companies and the United Steel Companies pat-
terns fully justifies the use of four decimal places in the high-angle part of the patterns, arbitrarily abbreviated in the version of the pattern given on the 1950 edition of the ASTM cards. The British Museum pattern, appearing only in the 1950 edition, is possibly abbreviated also. In performing this abbreviation, the 444 interplanar spacings of the Hanawalt, Rinn, and Frevel pattern should be given 0.79 rather than the 0.80 appearing on the card (the published value appearing on the original ASTM card is 0.789). The Gerlach pattern of 1922 is the only one to show a line for the 200 plane; this, marked very very weak, may well be in error.

All patterns give the strongest line as 220. Two of the British patterns give the 111 and 311 lines the same intensity, recording these as second and third strongest lines; the United Steel Companies pattern is completely at variance here, with 422 and 531 listed second and third strongest. The Hanawalt, Rinn, and Frevel pattern and that of the NBS list the three strongest lines as 220 , 111 , and 311.

It is difficult to get unoriented intensity measurements for $\mathrm{CaF}_{2}$. The perfect cleavage of the 111 plane caused considerable variation in the first few of several patterns made at the NBS. Only after diluting the sample with finely ground silica gel and drifting it very carefully into the specimen holder were consistent values obtained. $\mathrm{CaF}_{2}$ is one of the few materials in which the question of orientation is critical in determining the strongest line for indexing, as in most cases cleavage plane reflections are inherently the strongest. With $\mathrm{CaF}_{2}$ the planes of the 111 form, which bound the eight sides of a cleavage particle, are easily oriented to produce the strongest reflections. The 220 is the strongest reflection in an unoriented sample. Flat specimens prepared for Geigercounter equipment will without extraordinary precaution invariably show the 111 as the strongest indexing line.

The structure was determined by W. H. Bragg [30] in 1914. The lattice is face-centered lhat cell, angstroms

| 1922 | Gerlach [78] | 5.466 |
| :---: | :---: | :---: |
| 1927 | Thilo [221] | 5.55 |
| 1930 | Rumpf [198] | 5.460 |
| 1933 | Schumann [205 | 5.462 |
| 1939 | Zintl and Udgard [263].--------------- | 5.479 |
| 1953 | Swanson and Tatge ( $25^{\circ} \mathrm{C}$ ) ---------------- | 5.4626 |

cubic, the space group $\mathrm{O}_{\mathrm{h}}^{5}(\mathrm{Fm} 3 \mathrm{~m})$, with four molecules to the unit cell. Published lattice constants, supposedly in kX units, were converted to angstroms and are compared with the NBS value.

The density of calcium fluoride, in accordance with the NBS lattice constant, is 3.181 at $25^{\circ} \mathrm{C}$. The index of refraction of the sample used for the NBS pattern is $n=1.433$.

Table 42. Calcium fluoride, $\mathrm{CaF}_{2}$ (cubic)

| $h k l$ | 1922 <br> Gerlach <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  |  | 1938 <br> Hanawalt, Rinn, and Frevel Mo, 0.7093 A |  |  | Jessop United Steel <br> Mo, 0.7093 A |  |  | United Steel <br> Mo, 0.7093 A |  |  | British Museum <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  |  | $1953$ <br> Swanson and Tatge <br> Cu, $1.5405 \mathrm{~A}, 25^{\circ} \mathrm{C}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | $a$ | ${ }^{\text {d }}$ | I | $a$ | d | I | $a$ | ${ }^{\text {d }}$ | $I$ | $a$ | d | $I$ | $a$ | d | I | $a$ |
|  | A |  | A | A |  | A | A |  | A | A |  | A | A |  | A | A |  | A |
| 111 | 3.15 | m | 5.46 | 3.17 | 67 | 5.49 | 3.154 | 70 | 5.463 | 3.153 | 70 | 5.461 | 3.11 | 80 | 5.39 | 3.153 | 94 | 5.461 |
| 200 | 2.74 | vvw | 5.48 | ----- | --- |  |  | --- | ------ |  | --- | ------ | ---- | --- | ---- |  | --- |  |
| 220 | 1.94 | s | 5.49 | 1.93 | 100 | 5.46 | 1.932 | 100 | 5.465 | 1.931 | 100 | 5.462 | 1.90 | 100 | 5.37 | 1.931 | 100 | 5.462 |
| 311 | 1.65 | s | 5.47 | 1.65 | 50 | 5.47 | 1.647 | 70 | 5.462 | 1.646 | 70 | 5.459 | 1.63 | 80 | 5.41 | 1.647 | 35 | 5.462 |
| 222 | ----- | --- | ----- | ----- | --- |  |  | --- |  |  | --- | ------ | ---- | --- | ---- | 1.577 | 2 | 5.463 |
| 400 | 1.37 | m | 5.48 | 1.37 | 23 | 5.48 | 1.366 | 40 | 5.464 | 1.365 | 60 | 5.460 | 1.36 | 60 | 5.44 | 1.366 | 12 | 5.464 |
| 331 | 1.25 | ms | 5.45 | 1.259 | 23 | 5.488 | 1.254 | 50 | 5.466 | 1.253 | 60 | 5.462 | 1.25 | 60 | 5.45 | 1.253 | 10 | 5.462 |
| 422 | 1.12 | s | 5.49 | 1.119 | 30 | 5.482 | 1.1153 | 70 | 5.4639 | 1.114 | 90 | 5.457 | 1.11 | 80 | 5.44 | 1.1150 | 16 | 5.4624 |
| 511 | 1.050 | ms | 5.456 | 1.052 | 10 | 5.466 | 1.0515 | 50 | 5.4638 | 1.0510 | 70 | 5.4612 | 1.05 | 50 | 5.46 | 1.0512 | 7 | 5.4622 |
| 440 | 0.964 | ms | 5.453 | 0.970 | 6 | 5.487 | 0.9659 | 50 | 5.4640 | 0.9654 | 70 | 5.4611 | ---- | --- | ---- | 0.9657 | 5 | 5.4628 |
| 531 | . 924 | s | 5.466 | . 927 | 7 | 5.484 | . 9236 | 60 | 5.4641 | . 9231 | 90 | 5.4611 | ---- | --- | ---- | . 9233 | 7 | 5.4623 |
| 600 | ----- | --- | ----- | ----- | --- | ----- |  | --- | ------ | . 9101 | 50 | 5.4606 | ---- | --- | ---- | . 9105 | 1 | 5.4630 |
| 620 | 0.863 | ms | 5.458 | 0.868 | 5 | 5.490 |  | --- |  |  | -- |  |  | -- |  | . 8637 | 9 | 5.4625 |
| 533 | . 833 | ms | 5.462 | . 837 | 2 | 5.489 |  | --- |  |  | --- |  | ---- | --- | ---- | . 8330 | 3 | 5.4623 |
| 444 |  | --- | ----- | . 791 | 1 | 5.480 |  | --- |  |  | --- | ------ | ---- | --- | ---- | ------ | --- | ------ |
| 711 |  |  |  | . 769 | 2 | 5.491 |  | --- |  |  | --- |  | ---- | -- |  |  | --- |  |
| 642 |  |  |  | . 732 | 5 | 5.478 |  | --- |  |  | --- |  | ---- | -- | ---- |  |  |  |
| 731 |  | --- |  | . 714 | 3 | 5.484 |  | --- |  |  | --- |  | ---- | --- | ---- |  | -- |  |
| 822 |  |  |  | . 645 | 1 | 5.473 |  |  |  |  | --- |  |  |  |  |  |  |  |
| Average unit cell for last five lines.--- |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |  |
|  |  |  | 5.459 |  | --- | 5.481 |  | --- | ${ }^{1} 5.4640$ |  | --- | $1_{5.4610}$ | ---- | --- | 5.44 |  | --- | 5.4626 |

${ }^{1}$ Average for last four lines only.
2.42. Barium Fluoride, BaF, (Cubic)

The two ASTM patterns of barium fluoride (see table l) are compared in table 43 with one of two published by Broch, Oftedal, and Pabst [37], of which the more nearly complete is given in the table, along with a pattern made at the NBS.

The NBS sample was a specially purified material supplied by the Mallinckrodt Chemical Works. Their spectrographic analysis showed 0.01 to 0.1 percent of sodium and strontium.

The interplanar spacings for the Thilo and for the Broch, Oftedal, and Pabst patterns
were computed for table 43 directly in angstroms from the published Bragg angle data. The Hanawalt, Rinn, and Frevel values were converted to angstroms from $k X$ units. For the Thilo pattern the values of the unit cell for the last ten lines are within 0.005 of each other, indicating that the interplanar spacings are accurate to the third decimal place in this part of the pattern. Only the second place is recorded on the ASTM cards of both the old and new (1950) files. The first three spacings of the Thilo pattern as they appear on the ASTM card are calculated values to fit the unit cell dimension based on the high angle diffraction lines. The intensity measurements published by Thilo and those recorded on the ASTM card are both
given in the table (columns $I^{a}$ and $I^{b}$ ). The Hanawalt, Rinn, and Frevel, and the NBS measurements agree relatively well; the three strongest lines of both patterns are 111, 220, and 311 .

Barium fluoride has the fluorite structure, a face-centered cubic lattice, space group $\mathrm{O}_{\mathrm{h}}^{5}$ (Fm3m), and four molecules to the unit cell. Several published lattice constants, assumed to be in kX units, compare with the NBS value thus:

Unit cell, angstroms

| 1922 | Davey [56] | 6.21 |
| :---: | :---: | :---: |
| 1927 | Thilo [221] | 6.21 |
| 1933 | Schumann [205] | 6.199 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 6.2001 |

Table 43. Barium fluoride, $\mathrm{BaF}_{2}(\mathrm{c} u$ o vc $)$

| $h k l$ | 1927 <br> Thilo <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  |  |  | 1929 <br> Broch, Oftedal, and Pabst <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  |  | $1938$ <br> Hanawalt, Rinn, and Frevel <br> Mo, 0.7093 A |  |  | $1953$ <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | $I^{\text {a }}$ | $I^{\text {b }}$ | $a$ | d | $I$ | $a$ | d | I | a | d | I | $a$ |
|  | A |  |  | A | A |  | A | A |  | A | A |  | A |
| 111 | 3.61 | m | 70 | 6.25 | ------- | ----- |  | 3.59 | 100 | 6.22 | 3.58 | 100 | 6.20 |
| 200 | 3.12 | w | 50 | 6.24 | ------- | ----- | ------- | 3.10 | 25 | 6.20 | 3.100 | 30 | 6.200 |
| 220 | 2.206 | $s$ | 100 | 6.240 | 2.194 | $s$ | 6.205 | 2.19 | 100 | 6.19 | 2.193 | 79 | 6.203 |
| 311 | 1.875 | s | 100 | 6.219 | 1.868 | s | 6.195 | 1.86 | 80 | 6.17 | 1.870 | 51 | 6.202 |
| 222 | 1.798 | w | 50 | 6.228 | 1.789 | m | 6.197 | 1.78 | 15 | 6.17 | 1.790 | 3 | 6.201 |
| 400 | 1.553 | w | 50 | 6. 212 | ------- | ----- | ------- | 1.55 | 15 | 6.20 | 1.550 | 6 | 6. 2000 |
| 331 | 1.426 | s | 100 | 6.216 | 1.420 | s | 6.190 | 1.423 | 32 | 6.203 | 1.423 | 13 | 6.2027 |
| 420 | 1.392 | m | 70 | 6.225 | 1. 385 | m | 6.194 | 1.385 | 18 | 6.194 | 1.386 | 6 | 6.1983 |
| 422 | 1.268 | s | 100 | 6.212 | 1.264 | m | 6.192 | 1.265 | 32 | 6.197 | 1.266 | 14 | 6.2021 |
| 511 | 1.195 | s | 100 | 6.209 | 1.192 | m | 6.194 | 1.192 | 20 | 6.194 | 1. 1933 | 6 | 6.2006 |
| 440 | 1.098 | w | 50 | 6.211 | 1.094 | w | 6.198 | 1.097 | 5 | 6.206 | 1.0959 | 2 | 6.1993 |
| 531 | 1.050 | s | 100 | 6.212 | ------- |  | -------- | 1.047 | 15 | 6.194 | 1.0481 | 6 | 6.2006 |
| 600 | 1.035 | m | 70 | 6.210 |  | ----- | ------- | 1.033 | 3 | 6.198 | 1.0332 | 1 | 6.1992 |
| 620 | 0.981 | s | 100 | 6.204 | ------- | ----- | ------- | 0.980 | 6 | 6.198 | 0.9803 | 2 | 6.2000 |
| 633 | . 947 | m | 70 | 6.212. |  | ----- |  | . 946 | 3 | 6.203 | . 9455 | 1 | 6.2001 |
| 622 | . 937 | m | 70 | 6.213 | ----... | ----- | ------- | . 935 | 2 | 6.202 | . 9347 | 3 | 6.2001 |
| 444 | --- | ----- | ----- | ------- | ------- | ----- | ------- | ----- |  | ------- | . 8948 | 1 | 6.1994 |
| 711 | 0.869 | s | 100 | 6.204 |  | - | ------- | 0.868 | 3 | 6.199 | . 8682 | 4 | 6.2002 |
| 640 | . 861 | m | 70 | 6.209 |  |  |  | . 861 | 2 | 6.209 | . 8599 | 1 | 6.2008 |
| 642 | ------- | ---- | ----- | ------- | ------- | ----- |  | . 829 | 5 | 6.204 | . 8285 | 5 | 6.1999 |
| 731 | ------ | ----- | ----- | ------- | ------- |  |  |  |  | --..--- | . 8072 | 6 | 6.2002 |
| Average unit cell for last five lines. |  |  |  |  |  |  |  |  |  |  |  |  |  |
|  |  |  |  | 6.208 |  | ---.- | 6.191 | ------- | -- | 6.203 | ------- | -- | 6.2001 |

${ }^{\text {a }}$ Published.
${ }^{\mathrm{b}}$ ASTM card.

The lattice constant ( 6.20 ) given on the ASTM card for the Thilo pattern is that of Thilo and not, as designated, of Wyckoff (written "Wys" in error for "Wy ${ }_{2}$ ", Wyckoff, vol. 2). The density calculated from the NBS lattice constant is 4.886 at $26^{\circ} \mathrm{C}$. The index of refraction obtained for the sample used is $n=1.472$.
2.43. Mercurous Chloride, $\mathrm{Hg}_{2} \mathrm{Cl}_{2}$ (Tetragonal)

Four ASTM patterns of mercurous chloride (calomel) (see table l) are compared in table 44 with a pattern prepared at the NBS using a sample from the General Chemical Company, labelled No. 1891. The spectrographic analysis made at the Bureau indicated only traces

Table 44. Mercurous chloride, $\mathrm{Hg}_{2} \mathrm{Cl}_{2}$ (tetragonal)

${ }^{8}$ Published.
${ }^{\text {b }}$ ASTM card.
of Cu and Fe , and faint traces of $\mathrm{Al}, \mathrm{Mg}$, and Si; the limit of detection of the alkali elements is about 0.05 percent.

The Havighurst [90] pattern was made with molybdenum radiation for which a wavelength of 0.710 angstrom was given. The interplanar spacings were converted to the angstrom unit used here in accordance with the change in wavelength assigned to the radiation. The Hylleraas [112] spacings were published as Bragg angle measurements, and were converted to interplanar spacings directly in angstroms for table 44. The spacings of the Ruff, Ebert, and Luft [196] and the Hanawalt, Rinn, and Frevel [85] patterns were measured in $k X$ units, and were converted to angstroms. The NBS pattern resolves three lines not separated in previous patterns, resulting in a different selection of the three strongest lines. Where previous patterns had always included the combined 211-105 among the ASTM index lines, although not always in the same position, the NBS pattern shows the l10, 101, and 114 lines first, second, and third strongest, respectively.

Mercurous chloride has a tetragonal lattice, space group $\mathrm{D}_{4 \mathrm{~h}}^{17}(\mathrm{I} 4 / \mathrm{mm})$ [143], and two molecules of $\mathrm{Hg}_{2} \mathrm{Cl}_{2}$ in the unit cell. The lattice constant $a$ was determined for the NBS sample from an average of five calculations from hko planes, $c$ from an average of five calculations from $h k l$ planes with 0 or low h and $k$ values. These are compared with earlier determinations thus:

Unit cell in angstroms

|  |  |  |  |
| :---: | :---: | :---: | :---: |
| 1925 | Havighurst [90]. | 4.48 | 10.91 |
| 1926 | Mark and Steinbach [ 143 ] | 4.46 | 10.91 |
| 1946 | Frevel, Rinn, and Anderson [75] | 4.47 | 10.91 |
| 1953 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 4.478 | 10.91 |

The density from the NBS lattice constant is 7.176 at $26^{\circ} \mathrm{C}$. The indices of refraction were not measured on the NBS sample. Havighurst refers to the birefringence as the strongest known, and quotes values for the in-
dices: $\omega_{D}=1.97325$ (which is miscopied on the ASTM card as 1.97525) and $\epsilon_{\mathrm{D}}=2.6559$.

### 2.44. Mercuric Chloride, $\mathrm{HgCl}_{2}$ (Orthorhombic)

The two patterns for mercuric chloride in the X-ray diffraction file of the ASTM (see table l) are compared in table 45 with one recently prepared at the Bureau. The Hanawalt, Rinn, and Frevel pattern, omitted from the original ASTM index, is included in the 1950 index. In the original index the index lines of this pattern (the three strongest lines $4.35,3.00,2.70$ ) are mistakenly assigned to mercuric chlorate [1]. The NBS pattern was obtained from a J. T. Baker Chemical Co. sample numbered 101742. Spectrographic analysis at the NBS showed no impurity greater than 0.01 percent.

The data of Bräkken and Harang were published as a table of $h k l$ indices, $\sin ^{2} \theta$ values, and intensity values visually estimated. For table 45 the $\sin ^{2} \theta$ values were converted to interplanar spacings, using the iron radiation wavelength 1.93597 A. The spacings of Hanawalt, Rinn, and Frevel were converted from $k X$ units to angstroms. The 120 line is the strongest for all three patterns. The second strongest is the 200 , but this line is not resolved from the 031 line by Hanawalt, Rinn, and Frevel, so that their intensity measurement is a combination of the two intensities. The 011 and 111 are third and fourth strongest in the Hanawalt, Rinn, and Frevel pattern, reversed for the NBS pattern. The difference in intensity is probably too small to be significant; it is not due to the radiation used, as the conversion factor for molybdenum to copper radiation is close to 1 in this range ([1] page 108 of index covering original set of cards, or card no. vii of introduction to 1950 file).

In indexing the pattern the unit-cell dimensions were taken in the Dana convention, $\mathrm{c}<\mathrm{a}<\mathrm{b}$, although the reverse order is sometimes given. The unit cell dimensions published in 1934 by Braekken and Scholten [27] converted from $k X$ units to angstroms compare
thus with those derived from the NBS pattern:

Unit cell, angstroms

|  |  | $a$ | $b$ | $c$ |
| :--- | :--- | :---: | :---: | :--- |
| 1934 | Braekken and Scholten [27] -- | 5.975 | 12.761 | 4.334 |
| 1951 | Swanson and Tatge $\left(26^{\circ} \mathrm{C}\right) \ldots---$ | 5.96 | 12.76 | 4.32 |

The presence of hkO lines only if $k$ is even, and hOl lines only if $h+l$ is even, agrees with the generally accepted orthorhombic space group determination $D_{2 h}^{16}$ (Pmnb) for the crystal orientation used here. The density of the material, calculated from the NBS lattice constant, is 5.49 at $26^{\circ} \mathrm{C}$. The indices of refraction are higher than 1.75.

Table 45. Mercuric chloride, $\mathrm{HgCl}_{2}$ (orthorhombic)

| $h k l$ | $\begin{gathered} 1928 \\ \begin{array}{c} \text { Bräkken and } \\ \text { Harang } \end{array} \\ \mathrm{Fe}, 1.936 \mathrm{~A} \end{gathered}$ |  | 1938 <br> Hanawalt, Rinn, and Frevel <br> Mo, 0.709 A |  | $\begin{gathered} 1953 \\ \text { Swanson and } \\ \text { Tatge } \\ \mathrm{Cu}, \begin{array}{l} 1.5405 \mathrm{~A}, \\ 26^{\circ} \mathrm{C} \end{array} \end{gathered}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | $I$ | d | I | $d$ | $I$ |
|  | A |  | A |  | A |  |
| 120 | 4.34 | vs | 4.36 | 100 | 4.35 | 100 |
| 011 | 4.08 | s | 4.11 | 25 | 4.10 | 38 |
| 021 | 3.57 | * | ---- | ----- | 3.58 | 3 |
| 101 | 3.488 | w | ---- | ----- | 3.51 | 1 |
| 111 | 3.368 | s | 3.41 | 38 | 3.383 | 31 |
| 040 | 3.172 | w | 3.21 | 13 | 3.188 | 11 |
| 121 | 3.056 | vw | ----- | ----- | 3.066 | 2 |
| 031 | 3.019 | w | $\} 3.01$ |  | \{ 3.033 | 21 |
| 200 | 2.976 | s | \} 3.01 | 75 | 2.986 | 48 |
| 131 | \}2.692 |  |  |  |  |  |
| 220 | $\}^{2.692}$ | $s$ | 2.70 | 50 | 2.707 | 36 |
| 211 | 2.403 | m | 2.41 | 25 | 2.420 | 14 |
| 141 | 2.348 | w | ---- | ----- | 2.366 | 2 |
| 221 | 2.281 | vw | ------ | ----- | 2.297 | 4 |
| 051 | 2.194 | w | ------- | ----- | 2.202 | 2 |
| 002 | 2.158 | vw | 2.18 | 13 | 2.182 | 6 |
| 012 |  |  |  |  |  |  |
| 231 | 2.120 | m | 2.12 | 25 | 2.132 | 9 |
| 060 |  |  |  |  |  |  |
| 151 | 2.056 | m | 2.06 | 25 | 2.065 | 13 |
| 112 | 1.997 | m | 2.00 | 50 | 2.004 | 16 |
| 241 | 1.929 | m | 1.94 | 25 | 1.940 | 11 |
| 061 | 1.895 | w | 1.90 | 13 | 1.902 | 8 |
| 132 | 1.829 | w |  | ----- | 1.837 | 1 |
| 301 | 1.806 | vw |  | ----- | 1.810 | 1 |
| 042 | $\} 1.784$ | m | 1.79 | 13 | 1.791 | 6 |
| 311 |  |  |  |  | 1.791 | 6 |
| 251 | 1.762 | w |  | ---- | 1.769 | 4 |

Table 45. Mercuric chloride, $\mathrm{HgCl}_{2}$
(orthorhombic)-Con.

| $h k l$ | 1928 <br> Bräkken and Harang <br> Fe, 1.936 A |  | 1938Hanawalt, Rinn,and FrevelMo, 0.709 A |  | $1953$ <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$, $26^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | a | I | d | I | d | I |
|  | $A$ |  | A |  | A |  |
| 202 | 1.745 | vw |  | ---- | 1.765 | 1 |
| 071 | 1.674 | vw |  | ----- | 1.681 | <1 |
| 331 | 1.658 | w | 1.67 | 13 | 1.666 | 4 |
| 052 | 1.643 | m | ------- | ----- | 1.653 | < 1 |
| 232 | ------- |  | 1.62 | 13 | 1.619 | 3 |
| 080 | 1.589 | m | 1.59 | 13 | 1.595 | 1 |
| 341 | 1.569 | vw | ------- | ----- | 1.572 | 1 |
| 180 | 1.531 | m | 1.54 | 13 | 1.539 | 2 |
| 081 | 1.489 | w | ---.-.-- | -- | 1.496 | 1 |
| 312 |  |  | 1.455 | 13 | 1.454 | 4 |
| 013 | ------- |  | ------- | ----- | 1.431 | 1 |
| 023 | ------- |  |  |  | 1.406 | 3 |

### 2.45. Mercuric Iodide, $\mathrm{HgI}_{2}$ (Tetragonal)

The ASTM file of diffraction patterns contains three cards for mercuric iodide (see table 1) of which one (No. 3-1281) records only unit cell measurements. The patterns of the other two are compared in table 46 with a pattern prepared at the NBS.

The sample used for the NBS pattern was from Mallinckrodt Chemical Works, and was stated to be of ACS purity. Spectrographic analysis at the NBS shows a trace of iron and faint traces of calcium, chromium, magnesium, and silicon.

The interplanar spacings of the Havighurst pattern were reduced to angstroms in accordance with the wavelength given for the $X$ radiation. The spacings of the Hanawalt, Rinn, and Frevel pattern were converted to angstroms from kX units. The line 200-114-201 is not resolved in the ASTM patterns, and appears very strong-even stronger than the 200-114 and 201 combined for the NBS pattern. Thus in the Havighurst pattern this is the strongest line, and in the Hanawalt, Rinn, and Frevel pattern it is equally as strong as the 102. The NBS pattern shows the three strongest lines as 102, 101, and the combination 200-114.

Table 46. Mercuric iodide. $\mathrm{HgI}_{2}$ (tetragonal)


Table 46. Mercuric iodide, $\mathrm{HgI}_{2}$ (tetragonal)-Con.


Mercuric iodide belongs to the tetragonal system. It has a space group of $\mathrm{D}_{4 \mathrm{~h}}^{15}$ (P4/nmc) [15], with two molecules in the unit cell. The unit cell measurements derived from the NBS powder pattern are compared below with those of other workers after conversion to angstroms from kX units:

|  |  | $a$ | $c$ |
| :---: | :---: | :---: | :---: |
| 1926 | Bijvoet, Claassen, and Karssen [15] | 4.366 | 12.38 |
| 1927 | Huggins and Magill [99] | 4.35 | 12.36 |
| 1953 | Swans on and Tatge ( $26^{\circ} \mathrm{C}$ ) | 4.390 | 12.38 |

The density on the basis of the NBS determined unit cell is 6.325 at $26^{\circ} \mathrm{C}$.

### 2.46. Lead Fluochloride, PbFCl (Tetragonal)

The 1950 ASTM X-ray diffraction file includes two patterns for lead fluochloride (see table 1); one, of natural matlockite, is from the mineral type locality of Matlock, Derbyshire, England, furnished by the British Museum (Natural History), London; the other, from synthetic material, was first published in 1932 by Nieuwenkamp and Bijvoet. In 1933 Nieuwenkamp [162] compared patterns of matlockite, whose formula was then given as $\mathrm{Pb}_{2} \mathrm{OCl}_{2}$, and synthetic PbFCl , showing their identity. In table 47 the two ASTM patterns are compared with one prepared at the Bureau from material of high purity obtained from the NBS chemical laboratory. The sample had been prepared as part of a project for the precise determination of fluorine.

The data published on the Nieuwenkamp and Bijvoet pattern do not include interplanar spacings; for table 47 they were calculated directly in angstroms from the $\sin ^{2} \theta$ values listed. The interplanar spacings of the British Museum pattern, presumably in $k X$ units, were converted to angstroms. Although the interplanar spacings of the patterns check closely, the intensity measurements vary. The NBS and British Museum patterns agree that 101 is the strongest line, but the Nieuwenkamp and Bijvoet pattern shows the last line (312) strongest, with the second and third strongest in close proximity. The 002, the second strongest line of the NBS pattern, is unresolved in the others. The third and fourth strongest lines of the NBS pattern appear as second and third strongest in the British Museum pattern.

Bannister [5] in 1934 gave the structure as tetragonal, space group $D_{4 \mathrm{~h}}^{7}(\mathrm{P} 4 / \mathrm{nmm})$, and
postulated two molecules in the unit cell. A Nieuwenkamp and Bijvoet determination of the lattice constant, converted to angstroms, compares thus with the NBS value:

Unit cell angstroms

|  |  | $a$ | c |
| :---: | :---: | :---: | :---: |
| 1932 | Nieuwenkamp and Bijvoet [ 164 ] | 4.09 | 7.21 |
| 1951 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 4. 106 | 7.23 |

The density, in accordance with the NBS lattice constant, is 7.13 at $26^{\circ} \mathrm{C}$. The NBS sample was too finely powdered to determine the indices of refraction; Bannister found $\omega_{D}=2.145, \epsilon_{D}=2.006$.

Table 47. Lead fluochloride, PbFCl (tetragonal)

| $h k l$ | 1932 <br> Nieuwenkamp and Bijuoet $\text { Cr, } 2.2896 \mathrm{~A}$ |  | British Museum |  | 1953 <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$, $26^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | d | I | d | I |
|  | A |  | A |  | A |  |
| 001 | ------- | --- | 7.2 | 40 | 7.22 | 20 |
| 002 | ------ | ----- | ------ | ----- | 3.61 | 70 |
| 101 | 3.58 | 48 | 3.55 | 100 | 3.56 | 100 |
| 110 | 2.905 | 28 | 2.90 | 70 | 2.904 | 47 |
| 102 | 2.719 | 28 | 2.70 | 70 | 2.714 | 35 |
| 003 | 2.410 | 6 | 2.40 | 20 | 2.409 | 6 |
| 112 | 2.262 | 51 | 2.25 | 80 | 2.263 | 38 |
| 103 | 2.074 | 20 | 2.07 | 70 | 2.079 | 14 |
| 200 | 2.052 | 34 | 1.99 | 60 | 2.053 | 24 |
| 201 | 1.954 | 23 | 1.98 | 40 | 1.974 | 1 |
| 113 | 1.852 | 17 | 1.84 | 60 | 1.855 | 4 |
| 004 | --- | --- | - | - | 1.808 | 1 |
| 211 | 1.780 | 66 | 1.77 | 80 | 1.780 | 36 |
| 104 | 1.654 | 40 | 1.65 | 70 | 1.654 | 11 |
| 212 | 1.635 | 28 | 1.63 | 60 | 1.637 | 7 |
| 203 | 1.560 | 14 | 1.558 | 20 | 1.564 | 1 |
| 213 | 1.462 | 31 | 1.461 | 60 | 1.461 | 4 |
| 220 | ------- | ----- | ------- | ----- | 1.452 | 3 |
| 005 | 1.447 | 34 | 1.443 | 60 | 1.448 | 1 |
| 221 | 1.417 | 37 | ------- | ----- | ------ | ---- |
| 105 | 1. 363 | 23 | ------- | ----- | ----- |  |
| 222 | \} 1.344 | 66 | 1.343 | 60 | 1.346 | 3 |
| 301 |  |  |  |  |  |  |
| 310 | 1.298 | 68 | ------ | ----- | 1.299 | 4 |
| 115 | ------- | ----- |  | ----- | 1.293 | 3 |
| 214 | 1.285 | 89 |  |  | 1.289 | 5 |

TABLE 47. Lead fluochloride, PbFCl (tetragonal)-Con.

| $h k l$ | 1932 <br> Nieuwenkamp and Bijvoet $\mathrm{Cr}, 2.2896 \mathrm{~A}$ |  | British Museum |  | 1953 <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$, $26^{\circ} \mathrm{C}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | d | I | d | I |
|  | $A$ |  | $A$ |  | A |  |
| 302 311 | $\} 1.281$ | 17 | 1.276 | 70 | 1.281 | 1 |
| 223 | -- | --- | 1.240 | 20 | 1.244 | 1 |
| 312 | 1.220 | 100 | 1.222 | 60 | 1.223 | 2 |
| 006 | ------- | ----- | ------- | --- | 1.2041 | 1 |
| 303 | ------- | ----- | ------- | ----- | 1.1911 | 2 |
| 205 |  | ----- | 1.181 | 60 | 1.1826 | 2 |
| 106 | ------- | ----- | 1.156 | 60 | 1.1565 | 2 |
| 313 | - | ----- | 1.142 | 40 | 1.1443 | 1 |
| 215 | ------- | ----- | ------ | ----- | 1.1386 | 1 |
| 321 | ------- | ----- | 1.126 | 50 | 1.1254 | 1 |
| 304 |  | ----- | ------- | ----- | 1.0922 | 1 |
| 322 |  |  | 1.089 | 50 | 1.0863 | 1 |
| 323 |  |  |  | ----- | 1.0300 | 1 |
| 400 |  |  | 1.027 | 60 | 1.0265 | 1 |
| 216 |  | ----- | 1.008 | 60 | 1.0078 | 2 |
| 402 | ------- | ----- | ------- | ----- | 0.9872 | 2 |
| 117 | ------- | ----- | ------- | ----- | . 9735 | 1 |
| 330 | ------- | ----- | ------- | ----- | . 9664 | 2 |
| 324 | ------- | ----- | ------- | ----- | . 9639 | 3 |
| 412 |  |  | ------- | ----- | . 9608 | 2 |
| 207 |  | ---- |  | ---- | . 9223 | 1 |
| 413 |  |  |  |  | . 9203 | 1 |
| 420 |  |  |  |  | . 9185 | 1 |

### 2.47. Potassium Cyanide, KCN (Cubic)

The card file of diffraction patterns of the ASTM contains three cards for potassium cyanide (see table 1). Only two of these give patterns, the third [3-1299] recording only a unit cell dimension. In table 48 the two patterns are compared with one produced at the NBS. The NBS pattern was obtained from a Nallinckrodt Chemical Works sample marked Lot GNB. An analysis furnished by the chemical laboratory of the NBS follows (in percent): $\mathrm{Cl}, 0.05 ; \mathrm{PO}_{4}, 0.005 ; \mathrm{SO}_{4}$ (total S), 0.005 ; $\mathrm{Fe}, 0.03 ; \mathrm{Pb}, 0.0000 ; \mathrm{Na},<0.05$.

For table 48 the spacings of both ASTM patterns were converted from $k X$ to anystrom units. The table shows the published inter-

TABLE 48. Potassium cyanıde, KCN (cubic)


[^13]planar spacings of the Natta and Passerini pattern as well as the version given on the ASTM card, which is recalculated from a unit cell derived from the last two lines. Two of the lines of this pattern are extraneous to the NaCl structure postulated for potassium cyanide and are not indexed. The three strongest lines are the same for all patterns- 200,220 , and 311.

Potassium cyanide has the NaCl structure [159] with a disordered CN group, a facecentered cubic lattice, and four molecules to the unit cell. Unit cell measurements have not been of very high accuracy. A few, converted to angstrom units, are given below:

Unit cell in angstroms

| 1921 | Cooper [53] | 6.55 |
| :---: | :---: | :---: |
| 1922 | Bozorth [22] | 6.56 |
| 1931 | Natta and Passerini [159] | 6.51 |
| 1953 | Swanson and Tatge ( $25^{\circ} \mathrm{C}$ ) | 6.527 |

The density calculated from the NBS unit cell value is 1.555 at $25^{\circ} \mathrm{C}$. The index of
refraction determined on the NBS sample is $n=1.413$.

### 2.48. Sodium Cyanide, NaCN (Cubic)

The ASTM file contains two cards for the cubic form of sodium cyanide (see table l), the patterns of which are compared in table 49 with a pattern prepared at the NBS. The NBS sample was obtained from the J. T. Baker Chemical Company; it was numbered 121444. The chemical laboratory of the NBS reports that the material satisfies ACS standards, and gives the following analysis (in percent): $\mathrm{NaCN}, 96.2 ; \mathrm{Cl}, 0.02 ; \mathrm{FeCN}, 0.00 ; \mathrm{SO}_{4}, 0.00$; $\mathrm{S}, 0.003$; Thiocyanite, 0.02; Remainder $\mathrm{CO}_{3}$ and $\mathrm{H}_{2} \mathrm{O}$.

The spacings of the ASTM patterns were corrected from kX to angstrom units for the table. The Natta and Passerini pattern lists several reflections extraneous to the facecentered cubic structure postulated for sodium cyanide. In order to record this pattern on the ASTM card, a unit cell of 5.83 was calculated from the interplanar spacings of the

Table 49. Sod ium cyanide, NaCN (cubic)

${ }^{a}$ Published data. ${ }^{b}$ As recorded on ASTM card. ${ }^{\text {c }}$ Refers to spacings of $d^{a}$ column. ${ }^{d}$ Average for last four lines only. 79
last three planes, and from this value the remaining spacings were recalculated, including the five lines which do not belong to the NaCN pattern. In the table both the originally published and the ASTM versions of the pattern are given. The NBS Geiger-counter diagram for sodium cyanide showed extraneous lines due to sodium carbonate and the strong line of the orthorhombic form of NaCN . These were not listed in the table. All three patterns list 200 and 220 as the first and second strongest lines. The intensities of 311 and 222 are very close-the earlier patterns show 222 as the third strongest line, while the NBS pattern shows 311 third strongest.

The room temperature form of sodium cyanide is face-centered cubic and has four molecules to the unit cell-that is, NaCl structure [159] with disordered CN group. Some recent lattice constants, corrected from kX to angstrom units, compare with that determined at the NBS as follows:

Unit cell in angstroms

| 1931 | Natta and Passerini [159]. | 5.84 |
| :---: | :---: | :---: |
| 1938 | Verweel and Bijvoet [240] | 5.88 |
| 1953 | Swanson and Tatge [ $26^{\circ} \mathrm{C}$ ] | 5.893 |

The density was calculated from the NBS lattice constant as 1.591 at $26^{\circ} \mathrm{C}$. The index of refraction of the NBS material was determined as $n=1.453$.

### 2.49. Sodium Cyanide, NaCN (Orthorhombic)

Sodium cyanide has a reversible inversion point from the cubic form at room temperature to an orthorhombic form at $10^{\circ} \mathrm{C}$. A pattern was made at the NBS with the temperature maintained between $6^{\circ}$ and $7^{\circ} \mathrm{C}$. This is compared in table 50 with a pattern in the ASTM file (see table l), made at $-10^{\circ} \mathrm{C}$ by Verweel and Bijvoet [240]. The NBS sample is described in section 2.48 . on the cubic form of sodium cyanide.

The spacings in table 50 for the Verweel and Bijvoet pattern were calculated in angstrom units from published $\sin ^{2} \theta$ values. The

NBS diagram showed lines due to carbonate contamination as well as weak lines due to the presence of the cubic form, which are not given in the pattern of table 50. The three strongest lines are recorded in the NBS pattern as 110,002 , and 112.

The space group $\mathrm{C}_{2 v}^{20}$ ( Imm ) has been repeatedly assigned to the orthorhombic form of sodium cyanide on the basis of the determination

Table 50. Sodium cyanide, NaCN (orthorhombıc)

of Verweel and Bijvoet, who, however, suggest the possible alternatives of $\mathrm{D}_{2}^{8}$ (I222) or $\mathrm{D}_{2}^{25}$ (Immm). There are two molecules in the unit cell. The NBS pattern as indexed satisfies the requirements of any one of these three groups. The Verweel and Bijvoet unit cell determination compares thus with that of the NBS:

Unit cell, in angstroms

| 1938 | Verweel and Bijvoet ( $-10^{\circ} \mathrm{C}$ ) | a | $b$ | c |
| :---: | :---: | :---: | :---: | :---: |
|  | [240]. | -3.75 | 4.72 | 5.62 |
| 1953 | Swanson and Tatge ( $6^{\circ}$ to $7^{\circ} \mathrm{C}$ ) | 3.774 | 4.719 | 5.640 |

The density on the basis of the unit cell dimensions determined from the NBS pattern is 1.620 at $6^{\circ}$ to $7^{\circ} \mathrm{C}$.
2.50. Strontium Nitrate, $\operatorname{Sr}\left(\mathrm{NO}_{3}\right)_{2}$ (Cubic)

A pattern for strontium nitrate is compared in table 51 with two previously published patterns. The first, by Vegard [236] in 1922, was well indexed and misses few lines, although it is of less precision than the later patterns. The data were published as $\sin \theta$ values and estimated intensities. The former were converted to interplanar spacings in angstrom units for table 5l. The second pattern, by Hanawalt, Rinn, and Frevel, included in the diffraction pattern file of the ASTM (see table l), was converted from kX to angstrom units.

The sample for the NBS pattern was a specially purified material supplied by the Mallinckrodt Chemical Works. Their spectro-

Table 51. Strontium nitrate, $\operatorname{Sr}\left(\mathrm{NO}_{3}\right)_{2}$ (cubic)

| hkl | 1922 <br> Vegard <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}$ |  |  | $1938$ <br> Hanawalt, Rinn, and Frevel Mo, 0.7093 A |  |  | 1951 <br> Swanson and Tatge $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | ${ }_{\text {d }}$ | I | $a$ | ${ }^{\text {a }}$ | I | $a$ | d | I | $a$ |
|  | ${ }^{\text {A }}$ |  | A | A |  | ${ }^{\text {A }}$ | A |  | A |
| 111 | 4. 54 | m | 7.86 | 4.51 | 100 | 7.81 | 4.48 | 100 | 7.76 |
| 200 | 3.92 | m | 7.84 | 3.92 | 33 | 7.84 | 3.88 | 13 | 7.76 |
| 210 | 3.53 | w | 7.89 | 3.49 | 33 | 7.80 | 3.474 | 21 | 7.768 |
| 211 | 3.22 | ゅ | 7.89 | ----------- |  |  | 3.175 | 14 | 7.777 |
| 220 | 2.78 | w | 7.87 | - | - |  | 2.749 | 19 | 7.775 |
| 311 | 2.37 | s | 7.86 | 2.36 | 100 | 7.83 | 2.346 | 72 | 7.781 |
| 222 | 2.27 | $s$ | 7.85 | 2.24 | 100 | 7.76 | 2.246 | 54 | 7.780 |
| 400 | 1.96 | m | 7.84 | 1.94 | 17 | 7.76 | 1.945 | 12 | 7.780 |
| 411 |  |  |  |  |  |  | 1.836 | 2 | 7.789 |
| 331 | 1.80 | m | 7.86 | 1.78 | 17 | 7.76 | 1.785 | 16 | 7.781 |
| 420 | 1.75 | m | 7.84 | 1.75 | 17 | 7.83 | 1.740 | 12 | 7.782 |
| 422 | 1.60 | m | 7.81 | 1.58 | 17 | 7.74 | 1.589 | 10 | 7.784 |
| 333 | 1.51 | m | 7.84 | 1.50 | 17 | 7.79 | 1. 498 | 12 | 7.784 |
| 521 |  |  |  |  |  |  | 1.420 | 2 | 7.778 |
| 440 | 1.39 | m | 7.84 | 1.379 | 17 | 7.801 | 1.376 | 11 | 7.784 |
| 531 | 1.32 | s | 7.83 | 1.318 | 17 | 7.797 | 1.315 | 10 | 7.780 |
| 600 | 1.30 | w | 7.81 | ----------- | --------- | ----------- | 1.296 | 4 | 7.775 |
| 620 | 1.24 | w | 7.82 | - | -------- | ---------- | 1.231 | 2 | 7.786 |
| 533 | 1.20 | m | 7.84 | ----------- | --------- | --------- | 1.1867 | 4 | 7.782 |
| 622 | 1.176 | m | 7.80 | ----------- | --...-...- | --------- | 1.1736 | 1 | 7.785 |
| 444 | 1.128 | w | 7.82 | ----------- | --------- | --------- | 1.1235 | 2 | 7.784 |
| 711 | 1.094 | m | 7.81 | ----------- |  |  | 1.0893 | 3 | 7.779 |
| 642 | 1.045 | m | 7.82 | ---------- | -------- | -----.- | 1.0396 | 3 | 7.780 |
| 731 | 1.017 | s | 7.81 | ----------- | --------- |  | 1.0128 | 5 | 7.780 |
| 732 |  |  |  | ----------- |  |  | 0.9878 | 2 | 7.778 |
| 820 | 0.951 | s | 7.84 | -- |  |  | . 9435 | 4 | 7.780 |
| 422 | . 927 | s | 7.86 |  |  |  | . 9168 | 4 | 7.779 |

Table 51. Strontium nitrate, $\mathrm{Sr}\left(\mathrm{NO}_{3}\right)_{2}$ (cubic)-Con.

${ }^{\text {a }}$ Average of last two lines only.
graphic analysis shows $\mathrm{Ba}<0.01$ percent and $\mathrm{Na}<0.01$ percent as the only impurities greater than traces.

From the intensity measurements of the NBS pattern, the three strongest lines are the 111, 311, and 222 , consistent with the index lines of the ASTM card for the Hanawalt, Rinn, and Frevel pattern.

The lattice of strontium nitrate is simple cubic, four molecules to the unit cell. The space group according to Jaeger and Van Melle [115] is $\mathrm{T}_{\mathrm{h}}^{6}(\mathrm{~Pa} 3)$; Vegard and Bilberg [238] confirm this, but indicate the possibility of $\mathrm{T}^{4}(\mathrm{P} 2,3)$. The patterns of table 51 show hk0 only if $h$ is even, adding confirmation of the $\mathrm{T}_{\mathrm{h}}^{6}(\mathrm{~Pa} 3)$ group. Three published lattice constants are converted from $k X$ to angstrom units and compared with the NBS determination in the table below. Vegard and Fioer [239] present a coefficient of expansion between $10^{\circ}$ and $70^{\circ} \mathrm{C}$ of $2.58 \times 10^{-5}$. This was used to modify their lattice constant determination to correspond to that of the NBS made at $26^{\circ} \mathrm{C}$.

Unit cell dimensions, angstroms

| 1922 | Vegard [236] | 7.81 |
| :---: | :---: | :---: |
| 1932 | Ringdal [192] | 7.827 |
| 1942 | Vegard and Hoer ( $26^{\circ} \mathrm{C}$. [ 239$]$ _----- | 7.7818 |
| 1951 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 7.779 |

The density from the NBS lattice constant is 2.986 at $26 .^{\circ} \mathrm{C}$. The index of refraction is $n=1.587$.

### 2.51. Barium Nitrate, $\operatorname{Ba}\left(\mathrm{NO}_{3}\right)_{2}$ (Cubic)

The pattern for barium nitrate (nitrobarite) closely parallels that for strontium nitrate. Vegard [236] and Hanawalt, Rinn, and Frevel published patterns of which the latter is included in the ASTM file (see table l).

The NBS sample was specially purified material supplied by the Mallinckrodt Chemical Works. Their spectrographic analysis indicates the following impurities: $\mathrm{Al}<0.01$ percent, $\mathrm{Na}<0.01$ percent, and $\mathrm{Sr}<0.01$ percent.

In Vegard's paper the data were published as $\sin \theta$ values. For comparison with the data in table 52, they were converted to interplanar spacings in angstroms. The pattern of Hanawalt, Rinn, and Frevel was converted from $k X$ to angstrom units for this table. The three strongest lines, used as index lines for the ASTN cards, are the same for the NBS and the Ilanawalt, Rinn, and Frevel patterns: 311, 111, and 222.

The lattice of barium nitrate is simple cubic, four molecules to the unit cell. The patterns of table 52 confirm the determination of the space group $\mathrm{T}_{\mathrm{h}}^{6}(\mathrm{~Pa} 3)$ by Jaeger and Van Melle [115] and by Vegard and Bilherg [238].

Three published lattice constants are converted from $k \lambda$ to angstrom units and compared with the NBS determination in the table below. Vegard and Foer [239] present a coefficent of expansion between $10^{\circ}$ and $70^{\circ} \mathrm{C}$ of

Table 52. Barium nitrate, $\mathrm{Ba}\left(\mathrm{NO}_{3}\right)_{2}$ (cubic)

$1.75 \times 10^{-5}$. This was used to modify their lattice constant determination to correspond to that of the NBS made at $26^{\circ} \mathrm{C}$.

The density calculated from the NBS lattice constant is 3.244 at $26^{\circ} \mathrm{C}$. The index of refraction is $n=1.570$.

Unit cell, angstroms

| 1922 | Vegard [236] | 8.11 |
| :---: | :---: | :---: |
| 1932 | Ringdal [192] | 8.127 |
| 1942 | Vegard and Roer ( $26^{\circ} \mathrm{C}$ ) [239] | 8.1172 |
| 1951 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) | 8.119 |

2.52. Zinc Dorate, $\mathrm{ZnB}_{2} \mathrm{O}_{4}$ (Cubic)

No published pattern for zinc borate was found. The following pattern is offered by the NBS as an addition to the ASTM file. The sample used for the pattern of table 53 was one of the phosphor preparations of the Radio Corporation of America [135], sample XII-17, of high purity. The unit cell derived from an average of the values obtained from the last five lines is 7.4726 A , at $26^{\circ} \mathrm{C}$. The lattice derived from the powder pattern is body-centered cubic, with six molecules in the unit cell. The density based on the NBS lattice constant is 3.605 at $26^{\circ} \mathrm{C}$. The index of refraction for the sample was determined as $n=1.739$.

Table 53. Zinc borate, $\mathrm{ZnB}_{2} \mathrm{O}_{4}$ (cubic)

| $h k l$ | $1951$ <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |  | hkl | $1951$ <br> Swanson and Tatge <br> $\mathrm{Cu}, 1.5405 \mathrm{~A}, 26^{\circ} \mathrm{C}$ |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | $a$ |  | ${ }^{\text {d }}$ | I | $a$ |
|  | A |  | A |  | A |  | A |
| 110 | 5.29 | 6 | 7.48 | 710 | 1.0568 | 1 | 7.473 |
| 200 | 3.74 | 3 | 7.48 | 640 | 1.0365 | 1 | 7.474 |
| 211 | 3.048 | 100 | 7.466 | 721 | 1.0169 | 3 | 7.473 |
| 310 | 2.364 | 23 | 7.476 | 642 | 0.9991 | 1 | 7.477 |
| 222 | 2.158 | 1 | 7.476 | 730 | . 9812 | 2 | 7.4726 |
| 321 | 1.997 | 20 | 7.472 | 732 | . 9490 | 1 | 7.4724 |
| 400 | 1.869 | 13 | 7.476 | 811 | . 9198 | 3 | 7.4725 |
| 411 | 1.761 | 38 | 7.471 | 820 | . 9062 | 1 | 7.4727 |
| 420 | 1.672 | 2 | 7.477 | 653 | . 8932 | 1 | 7.4730 |
| 332 | 1.594 | 3 | 7.477 | 822 | . 8807 | 1 | 7.4730 |
| 422 | 1.526 | 25 | 7.476 | 831 | . 8687 | 3 | 7.4728 |
| 510 | 1.466 | 5 | 7.475 | 662 | . 8573 | 1 | 7.4738 |
| 521 | 1.364 | 8 | 7.471 | 752 | . 8462 | 1 | 7.4734 |
| 440 | 1.321 | 4 | 7.473 | 910 | . 8252 | 1 | 7.4725 |
| 530 | 1.282 | 3 | 7.475 | 842 | . 8153 | 1 | 7.4723 |
| 600 | 1.246 | 1 | 7.476 | 921 | . 8058 | 1 | 7.4727 |
| 611 | 1.213 | 2 | 7.477 | 664 | . 7966 | 1 | 7.4728 |
| 620 | 1.1817 | 1 | 7.474 | 930 | . 7877 | 1 | 7.4728 |
| 541 | 1.1531 | 3 | 7.473 |  |  |  |  |
| 631 | 1.1025 | 1 | 7.478 | Average for last five lines. |  |  |  |
| 444 | 1.0788 | 1 | 7.474 |  |  |  | 7.4726 |

### 2.53. Magnesium Silicate, $\mathrm{Mg}_{2} \mathrm{SiO}_{4}$ (Orthorhombic)

Two patterns for magnesium silicate (forsterite) in the ASTM file (see table l) are compared in table 54 with a pattern prepared
at the NBS. The NBS was furnished with a sample of high purity, labeled X-9, by the Radio Corporation of America. The material had been prepared in connection with a phosphor project [135] as a solid state reaction, at $1,500^{\circ} \mathrm{C}$. The large unit cell of magnesium silicate furnishes a large number of the possible planar reflections for an X -ray diagram with copper radiation. Thus, indexing becomes increasingly difficult with increasing Bragg angle. As $\theta$ increases, Cl ark's interplanar spacings diverge more and more widely from the values calculated for indexing the NBS pattern. The last 20 lines of his pattern were omitted from the table because the divergence combined with the multiplicity of possible lines makes indexing purely arbitrary. The Geiger counter intensity measurements of the NBS pattern show 112 to be the strongest line, 131 second, and 222 third, rather than the order 222, 131, and 112 estimated by Clark himself, 222, ll2, 131 given on the ASTM card for Cl ark, or $222,021,130$ on the pattern of Hanawalt, Rinn, and Frevel.

Forsterite is orthorhombic with a space group presumably the same as that specified by Bragg and Brown [32] for olivine, $\mathrm{V}_{\mathrm{h}}^{16}$, or $\mathrm{D}_{2 \mathrm{~h}}^{16}$ (Pbnm). There are four molecules in the unit cell. Although several sets of unit-cell dimensions are available for the closely related mineral olivine (iron-bearing), only one was found for forsterite. Rinne [193] in 1923 examined a natural forsterite from Vesuvius, for which he found dimensions which agree very closely with olivine measurements. Converting from $k X$ to angstrom units, his values compare with those derived from the NBS pattern thus:

Unit cell, angstroms

|  |  | a | $b$ | c |
| :---: | :---: | :---: | :---: | :---: |
| 1923 | Rinne [193] ---------------- | 4.75 | 10.28 | 6.00 |
| 1951 | Swanson and Tatge ( $26^{\circ} \mathrm{C}$ ) .....- | 4.76 | 10.20 | 5.99 |

From the NBS data the cell dimensions were calculated from spacings only of planes parallel to one or more axes. The density
calculated from the cell dimensions of the NBS determination is 3.213 at $26^{\circ} \mathrm{C}$. The material was too finely powdered to determine the indices of refraction.

Table 54. Magnesium silicate, $\mathrm{Mg}_{2} \mathrm{SiO}_{4}$ (orthorhombic)

| $h k l$ | 1938 <br> Hanawalt, Rinn, and Frevel <br> Mo, 0.7093 A |  | $\begin{gathered} 1946 \\ \text { Clark } \end{gathered}$ |  |  | $\begin{gathered} 1953 \\ \text { Swanson and } \\ \text { Tatge } \\ \mathrm{Cu}, \begin{array}{l} 1.5405 \mathrm{~A}, \\ 26^{\circ} \mathrm{C} \end{array} \end{gathered}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | d | I | d | $I^{\text {a }}$ | $I^{\text {b }}$ | d | $I$ |
|  | A |  | A |  |  | A |  |
| 020 | 5.1 | 11 | 5.1 | vvw | 10 | 5.11 | 26 |
| 021 | 3.90 | 40 | 3. 86 | ms | 70 | 3.88 | 69 |
| 101 | 3.73 | 5 | 3.71 | vw | 20 | 3.73 | 25 |
| 111 | \} 3.50 | 20 | 3.49 | vw | 20 | 3.487 | 21 |
| 121 | 3.00 | 13 | 2.98 | vw | 20 | 3.000 | 17 |
| 002 |  |  | 2.87 | vvw | 10 | ------- | --- |
| 130 | 2.78 | 40 | 2.75 | ms | 70 | 2.768 | 53 |
| 131 | 2.52 | 32 | 2.50 | s | 80 | 2.513 | 73 |
| 112 | 2.45 | 40 | 2.45 | $s$ | 90 | 2.458 | 100 |
| 041 | ------- | ------ | 2.34 | vvw | 10 | 2.348 | 9 |
| 210 |  | ------ | 2.31 | vvw | 10 | 2.316 | 9 |
| 122 | 2.26 | 40 | 2.26 | w | 40 | 2.268 | 59 |
| 140 |  | ------ | 2.24 | w | 40 | 2.250 | 33 |
| 211 | 2.15 | 11 | 2.15 | vw | 20 | 2. 161 | 15 |
| 132 | 2.02 | 2 | 2.03 | vvw | 10 | 2.034 | 5 |
| 042 | 1.95 | 2 | 1.934 | vvw | 10 | 1.945 | 4 |
| 150 | 1.88 | 3 | 1.864 | vvw | 10 | 1.878 | 5 |
| 113 | 1.81 | 3 | 1.798 | vvw | 10 | 1.811 | 2 |
| 151 |  | ------ | 1.776 | vvw | 10 | 1.792 | 3 |
| 222 | 1.74 | 100 | 1.737 | vs | 100 | 1.748 | 60 |
| 240 |  | ------ | 1.729 | vvw | 10 | -- | ----- |
| 241 | 1.67 | 10 | 1.661 | vw | 20 | 1.670 | 13 |
| 061 | 1.62 | 11 | 1.624 | vw | 20 | 1.636 | 12 |
| 133 |  |  | 1.607 | vw | 20 | 1.618 | 15 |
| 152 | 1.57 | 8 | 1.579 | vvw | 10 | 1. 589 | 2 |
| 043 | ------ | ------ | 1.560 | vvw | 10 | 1. 572 | 10 |
| 301 |  |  | 1.523 | vvw | 10 | 1.531 | 1 |
| 311 | ) |  |  |  |  |  |  |
| 213 | \} |  | 1.504 | vvw | 10 | 1.514 | 10 |
| 320 |  |  |  |  |  |  |  |
| 004 | 1.493 | 32 | 1.487 | * | 40 | 1. 497 | 27 |
| 062 |  |  | 1.471 | ms | 70 | 1.479 | 30 |
| 330 | ------- | ------ | 1.424 | vvw | 10 | 1.438 | 4 |
| 170 | 1.398 | 20 | 1.385 | vw | 20 | 1. 396 | 13 |
| 233 |  |  |  | --- | --- | 1.394 | 9 |
| 322 | 1. 353 | 28 | 1.341 | w | 40 | 1.351 | 17 |
| 134 | 1.318 | 10 | 1.305 | vw | 20 | 1. 316 | 9 |
| 332 |  | ------ | 1.285 | vvw | ${ }^{\text {c }} 10$ | 1.295 | 2 |
| 204 |  |  |  | --- | --- | 1.266 | 1 |

[^14]2.54. Magnesium Tungstate, $\mathrm{MgWO}_{4}$ (Monoclinic)

Four patterns, all from the literature, of magnesium tungstate are compared in table 55 with a pattern prepared at the NBS. Two of these are by Broch [39], one by Fonda [70], and one by Dunning and Megaw [65]. Broch supplied most of the indices. In addition to the indices his data include diffraction angles and, for the second pattern, estimated intensities. The interplanar spacings listed in table 55 were computed from his reflection angles so that they appear in angstroms. The Fonda and the Dunning and Megaw interplanar spacings were converted from presumed kX units to angstroms.

For the NBS pattern, material of exceptionally high purity was obtained from the Radio Corporation of America, marked No. 4, prepared at $1,000^{\circ} \mathrm{C}$.

There is not notable agreement among the patterns on the strongest lines, chiefly because of the large number of lines of high intensity and the fact that the intensities are only estimated except for the NBS pattern. For the pattern of the Bureau the three strongest lines are the 111,011 , and 100.

Broch's 1930 paper [39] gives the space group as $C_{2 h}^{4}(P 2 / c)$, two molecules in the unit cell. The unit-cell constants of the monoclinic magnesium tungstate crystals were given by Broch from his first pattern as $a=4.67$, $b=5.66, c=4.92, \beta=89^{\circ} 35^{\prime}$, from his second pattern as $a=4.68, b=5.66, c=4.93, \beta=89^{\circ} 40^{\prime}$. Converted from kX units to angstroms, the later values compare with those derived from the NBS pattern thus:

Unit cell, angstroms

| 1930 | Broch [39] ..-.-.--------- | $a$ | $b$ | $c$ | $\beta$ |
| :---: | :---: | :---: | :---: | :---: | :---: |
| 1951 | Swanson and Tatge $\left(26^{\circ} \mathrm{C}\right)$ | 4.69 | 5.67 | 4.94 | $89^{\circ} 40^{\prime}$ |
|  |  | 4.92 | $89^{\circ} 40^{\circ}$ |  |  |

The density calculated from the NBS lattice constant is 6.897 at $26^{\circ} \mathrm{C}$. The material was too finely powdered to determine the indices of refraction; it is known that they are higher than 1.75.

Table 55. Magnesium tungstate, $\mathrm{MghO}_{4}$ (monoclinic)


[^15]
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## 4. Cumulative Index to Volumes I and II

Aluminum, Al
Aluminum oxide, alpha (corundum), $\mathrm{Al}_{2} \mathrm{O}_{3}$.-
Anmonium bromide, $\mathrm{NH}_{4} \mathrm{Rr}-$
Ammonium chloride (sal-ammoniac), $\mathrm{NH}_{4} \mathrm{Cl}--$
Arsenic trioxide (arsenolite), $\mathrm{As}_{2} \mathrm{O}_{3}$
Barium carbonate (witherite), $\mathrm{BaCO}_{3}$
Barium fluoride, $\mathrm{BaF}_{2}$
Barium nitrate (nitrobarite), $\mathrm{Pa}\left(\mathrm{NO}_{3}\right){ }_{2} \cdots$
Beryllium oxide (bromellite), BeO-
Cadmium oxide, CdO-
Calcium carbonate (calcite), $\mathrm{CaCO}_{3}$
Cal cium fluoride (fluorite), $\mathrm{CaF}_{2}-$
Calcium hydroxide (portlandite), $\mathrm{Ca}(\mathrm{OH})_{2}$
Calcium oxide, CaO- ..... 43
Carbon (diamond), C- ..... II ..... 5
Ceric oxide, $\mathrm{CeO}_{2}$ ..... 56
Cesium chloride, CsCl ..... 44
Copper, Cu ..... 15
Cupric oxide (tenorite), CuO- ..... 49
Cuprous oxide (cuprite), $\mathrm{Cu}_{2} \mathrm{O}$ ..... 23
Gallium, Ga- ..... 9
Germanium, Ge ..... 18
Germanium dioxide, $\mathrm{GeO}_{2}$ ..... 51
Gold, Au ..... 33
Lead, Pb ..... 34
Lead bromide, $\mathrm{PbEr}_{2}$ ..... 47
Lead carbonate (cerussite), $\mathrm{PbCO}_{3}$ ..... 56
Lead chloride (cotunnite), $\mathrm{PbCl}_{2}$ ..... 45
Lead fluochloride (matlockite), FbFC. ..... 76
Lead oxide (litharge), PbO (red) ..... 30
Lead oxide (massicot), FbO (yellow) ..... 32
Lead sulfide (galena), PbS ..... 18
Lithium chloride, LiCl ..... 62
Lithium fluoride, LiF ..... 61
Magnesium, Mg. ..... 10
Magnesium aluminate (spinel), $\mathrm{MgAl}_{2} \mathrm{O}_{4} \cdots-$ ..... 35
Magnesium oxide (periclase), MgO- ..... 37
Magnesium silicate (forsterite), $\mathrm{Mg}_{2} \mathrm{SiO}_{4}$ ..... 83
Magnesium tungstate, $\mathrm{MgWO}_{4}$ ..... I 84
Mercuric chloride, $\mathrm{HgCl}_{2}$ ..... 73
Mercuric iodide, $\mathrm{HgI}_{2}$ ..... 74
Mercurous chloride (calomel), $\mathrm{Hg}_{2} \mathrm{Cl}_{2}$ ..... 72
Molybdenum, Mo ..... 20
Nickel, Ni
Vol. Fage Vol. Fage ..... 11 ..... 11 ..... II ..... II ..... II ..... II ..... I 59 ..... I 51 ..... I 51 ..... II ..... II ..... I ..... I ..... 70 ..... 70 ..... I 81 ..... I 81 ..... 36 ..... 36 ..... 27 ..... 27 ..... 51 ..... 51 ..... 69 ..... 69 ..... I 58 ..... I 58
Nickelous oxide (bunsenite), NiO-
Nickelous oxide (bunsenite), NiO- ..... 47 ..... 47
Palladium, Pd
Palladium, Pd ..... 21 ..... 21
Platinum, Pt
Platinum, Pt ..... 31 ..... 31
Potassium bromide, KBr
Potassium bromide, KBr ..... 66 ..... 66
Potassium chloride (sylvite), KCl
Potassium chloride (sylvite), KCl ..... 65 ..... 65
Potassium cyanide, KCN
Potassium cyanide, KCN ..... 77 ..... 77
Potassium fluoride, KF.
Potassium fluoride, KF. ..... 64 ..... 64
Potassium iodide, KI
Potassium iodide, KI ..... 68 ..... 68
Rhenium, Re
Rhenium, Re ..... 13 ..... 13
Selenium dioxide (selenolite), $\mathrm{SeO}_{2}$
Selenium dioxide (selenolite), $\mathrm{SeO}_{2}$ ..... 53 ..... 53
Silicon, Si
Silicon, Si ..... II ..... II
Silicon dioxide (cristobalite, low or
Silicon dioxide (cristobalite, low or alpha), $\mathrm{SiO}_{2}$ alpha), $\mathrm{SiO}_{2}$ ..... 39 ..... 39
Silicon dioxide (cristobalite, high or beta), $\mathrm{SiO}_{2}-$ ..... 42
Silver, Ag ..... 23
Sodium chloride (halite), NaCl - ..... II ..... 41
Sodium cyanide, cubic, NaCN ..... 78
Sodium cyanide, orthorhombic, NaCN ..... 79
Sodium fluorióe (villiaumite), NaF- ..... 63
Sodium sulfate (thenardite), $\mathrm{Na}_{2} \mathrm{SO}_{4}$ ..... 59
Stannic oxide (cassiterite), $\mathrm{SnO}_{2}-$ ..... 54
Strontium nitrate, $\mathrm{Sr}\left(\mathrm{NO}_{3}\right)_{2}$ ..... 80
Strontium sulfare (celestite), $\mathrm{SrSO}_{4}$ ..... II ..... 61
Tantalum, Ta- ..... 29
Tellurium, Te ..... 26
Thallium oxide, $\mathrm{T1}_{2} \mathrm{O}_{3}$ ..... 28
Thorium oxide (thorianite), $\mathrm{ThO}_{2}$ - ..... I 57
Tin, alpha, Sn- ..... II 12
Tin, beta, Sn ..... 24
Titanium dioxide (enatase), $\mathrm{TiO}_{2}-$ ..... I 46
Titanium dioxide (rutile), $\mathrm{TiO}_{2}$ ..... I 44
Tungsten, W- ..... I 28
Uranium dioxide, $\mathrm{UO}_{2}$ ..... II 33
Zinc, Zn ..... I 16
Zinc aluminate (gahnite), $\mathrm{ZnAl}_{2} \mathrm{O}_{4}$ ..... II ..... 38
Zinc borate, $\mathrm{ZnB}_{2} \mathrm{O}_{4}$
Zinc oxide (zincite), ZnO O ..... II 25
Zinc pyrosilicate hydrate (hemimorphite),
$\mathrm{Zn}_{4}(\mathrm{OHi})_{2} \mathrm{Si}_{2} \mathrm{O}_{7} \cdot \mathrm{H}_{2} \mathrm{O}$ ..... II $\quad 62$
Zinc sulfide, alpha (wurtzite), ZnS - ..... II 14
Zinc sulfide, beta
(sphalerite), ZnS ..... 16
Zirconium, Alpha, Zr ..... 11


[^0]:    ${ }^{1}$ Figures in brackets indicate the literature references at the end of this volume. They are in alphabetical order.
    ${ }^{2}$ The Joint Committee represents the American Crystallographic Society, American Society for Testing Materials, and the Institute of Physics (England).

[^1]:    ${ }^{\text {a }}$ Average of last three lines.
    ${ }^{\mathrm{b}}$ Average of last two lines.
    ${ }^{\text {c }}$ Average of three lines preceding last line.
    ${ }^{d}$ Average of four lines.

[^2]:    ${ }^{\text {a }}$ Unit not known.

[^3]:    *Average for three lines only.
    ${ }^{b}$ Average for four lines only.

[^4]:    ${ }^{\text {a }}$ Unit not known.

[^5]:    ${ }^{\text {a Myckoff pattern combined with that of Barth and Posnjak }}$ on ASTM card II-588.

[^6]:    ${ }^{\text {a }}$ Eleven additional lines omitted.

[^7]:    ${ }^{\mathrm{b}}$ Averaged from lines not shown.

[^8]:    ${ }^{3}$ Peak height intensities are considered preferable in the ASTM card file to integrated intensities because most of those using the file for routine analyses measure peak height or its equivalent.

[^9]:    ${ }^{a}$ Published. $\quad{ }^{b}$ On ASTM card; first fourteen lines recalculated (converted to angstroms). ${ }^{c}$ Published.

[^10]:    ${ }^{\text {a }}$ Publ ished.

[^11]:    ${ }^{\mathrm{b}}$ ASTM card.

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[^13]:    ${ }^{a}$ Published data. ${ }^{b}$ As recorded on ASTM card.

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